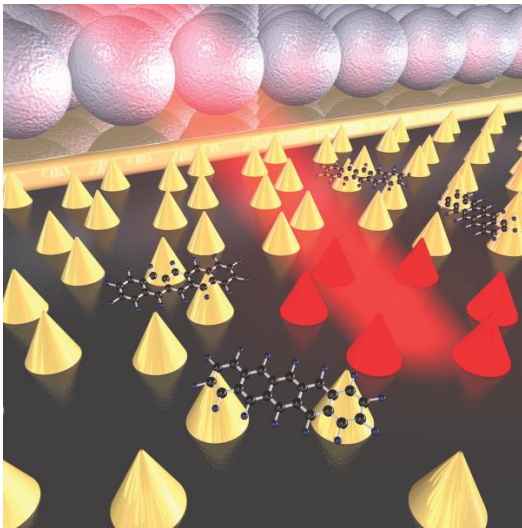


Basic module

Physics of Nanostructures

Summer term 2025

J. Meyer, D. Kölle, I. Zaluzhnyy,
Excercises: R. Löffler



Additional content

EBERHARD KARLS
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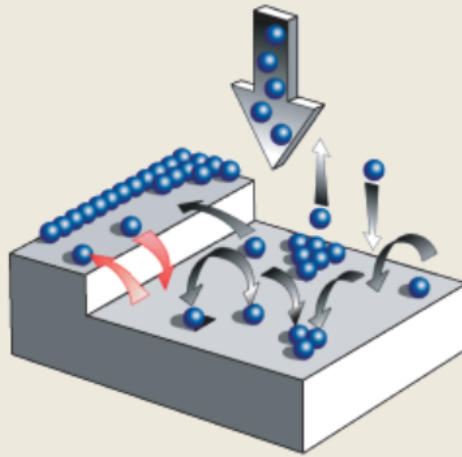
Schreiber Group

- Research Topics
- Research Methods
- People
- Publications
- Publications (sel.)
- Teaching**
- Teaching Archive
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Tübingen University

Excellence Strategy

Physik der Nanostrukturen



Teaching Material Archive

Lectures in SoSe 2025 ^{NEW}

Prof. Dr. Jannik Meyer
Prof. Dr. Dieter Kölle
Dr. Ivan Zaluzhnyy

Mon. 10.15-12.00 in D4A19 and Thu. 14.15-16.00 in N4
Eintrag im **Alma-Portal** und in **Ilias**

www.soft-matter.uni-tuebingen.de → teaching

Additional content



Schreiber Group

Research Topics

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Excellence Strategy

Equal Opportunities

Physics Home

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Computing Services

University Library

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Physics of Nanostructures

Prof. Dr. Jannik Meyer

Prof. Dr. Dieter Kölle

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Mon. 10.15-12.00 in D4A19 and Thu. 14.15-16.00 in N4

Link to **Alma-Portal**

Registration for this course in ILIAS

Materials for the lectures

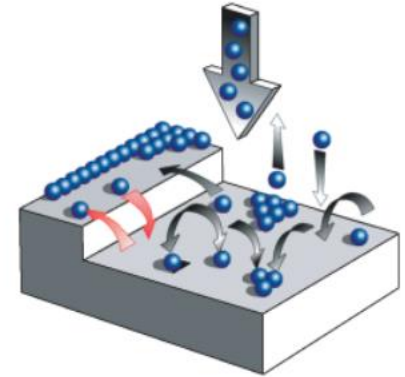
(complete list of materials is posted on ILIAS)

Preparation (Chapter 6)

- Script
- Presentation
- Lecture notes

Characterization (Chapter 7)

- Script
- Presentation
- Lecture notes



login: soft
password: soft07

Plan for the next three weeks

6. Preparation of nanostructures:

07.07, 10.07 - Theoretical introduction into thin film growth

14.07 – Overview of various experimental techniques

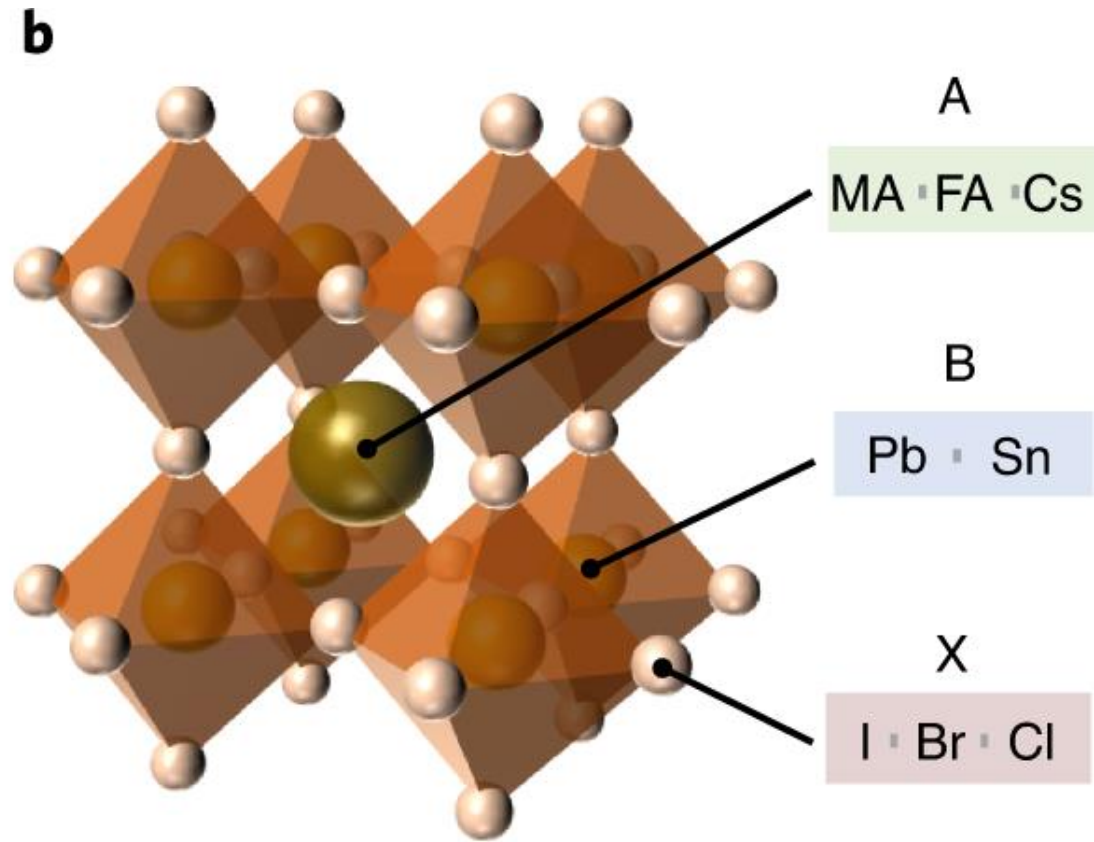
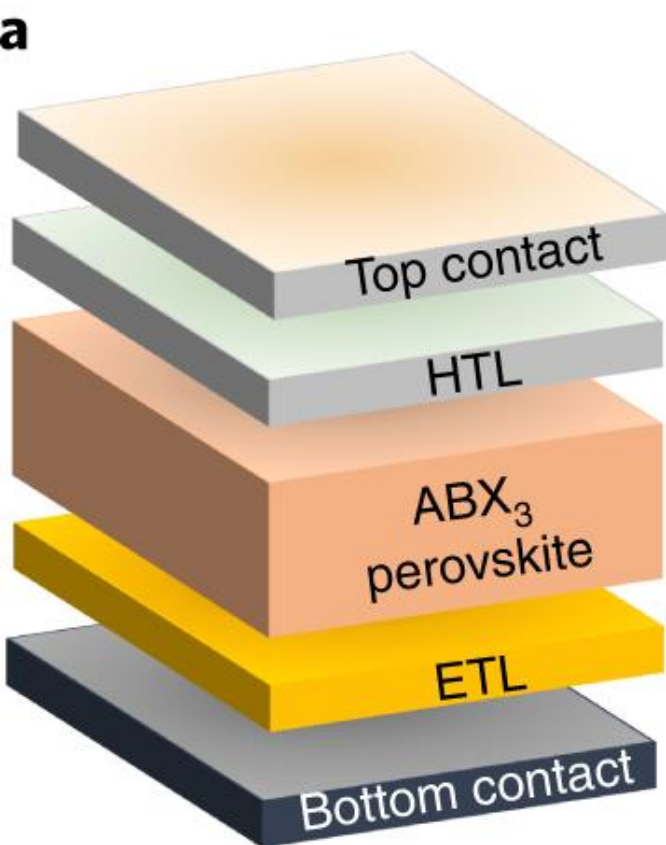
7. Characterization of nanostructures:

17.07, 21.07 – X-ray scattering from thin films

24.07 – Overview of various types of microscopy and spectroscopy

Why thin films?

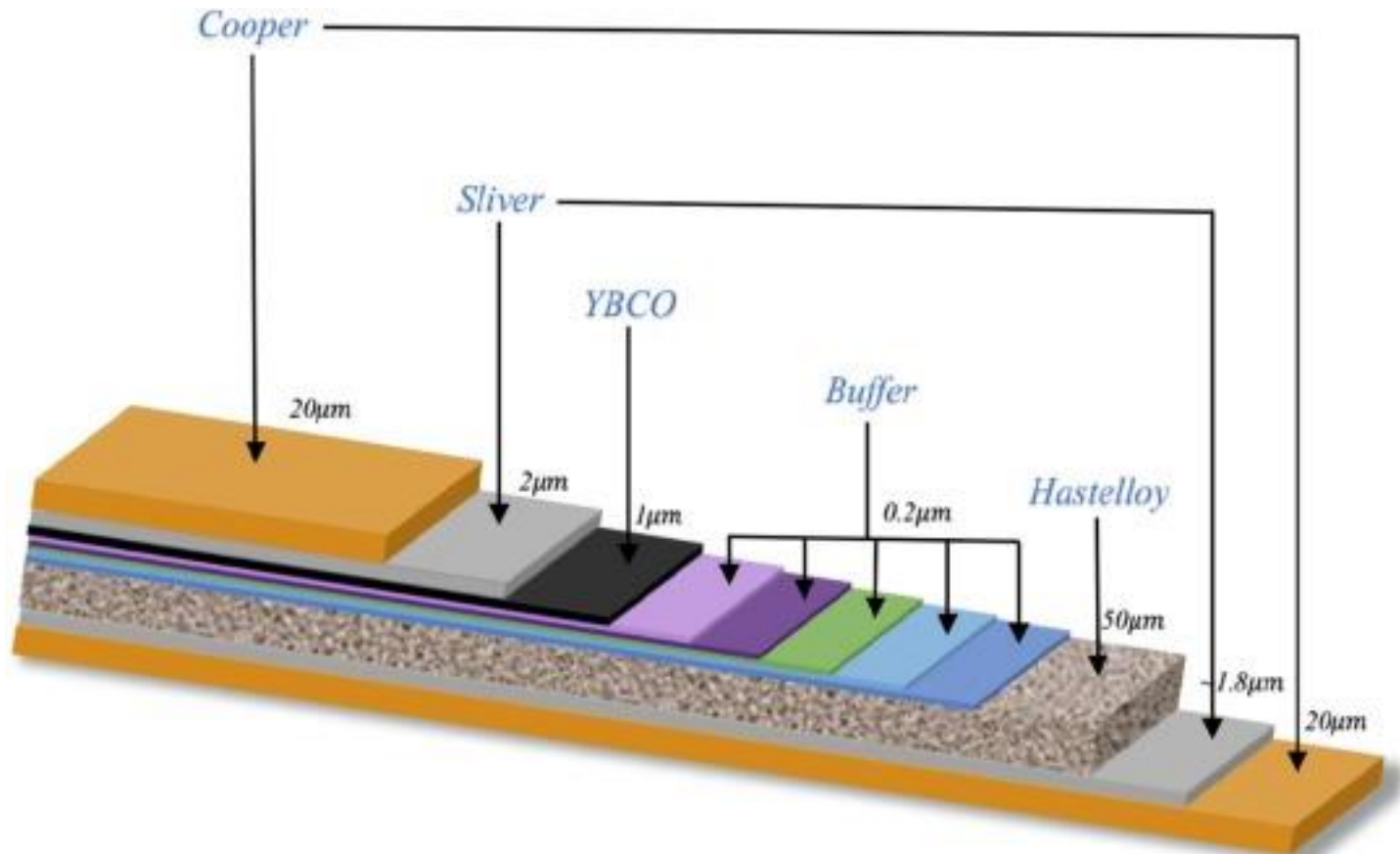
Perovskite-based solar cell



Y. Zhou, *et al.* *Nat Energy* **7**, 794–807 (2022).

Why thin films?

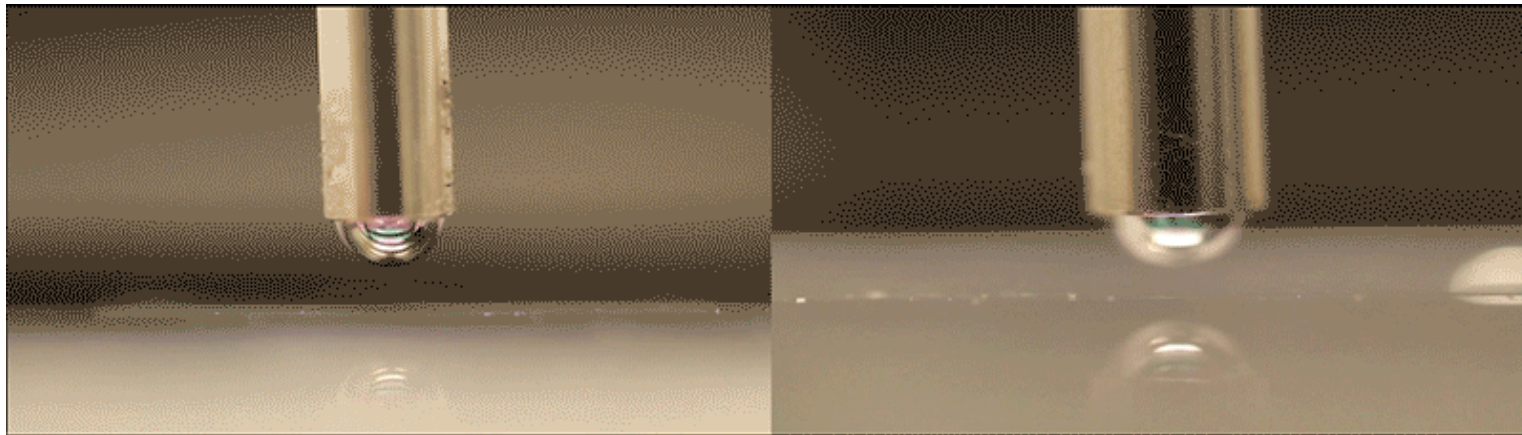
High-Temperature Superconducting tape



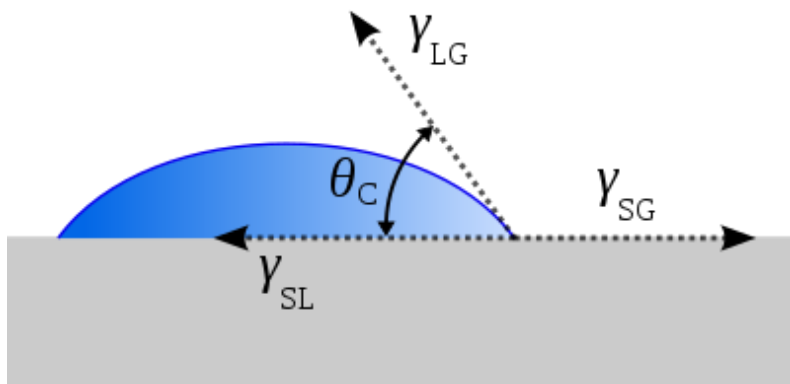
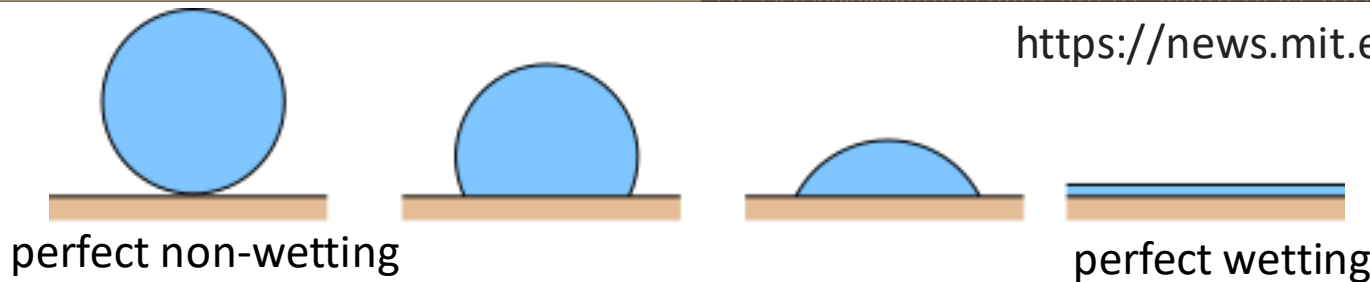
Droplet on a surface

mercury on glass

water on glass



<https://news.mit.edu/>



γ_{SL} - surface tension between surface and liquid

γ_{LG} - surface tension between liquid and gas

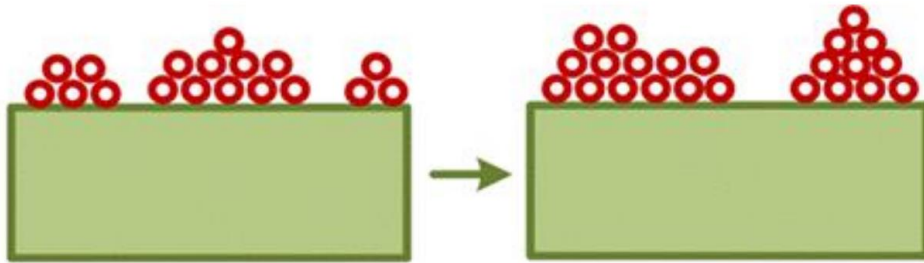
γ_{SG} - surface tension between surface and gas

θ_C - contact angle

the Young equation

$$\gamma_{SL} + \gamma_{LG} \cdot \cos \theta_C = \gamma_{SG}$$

Three categories of growth



Volmer-Weber (islands)

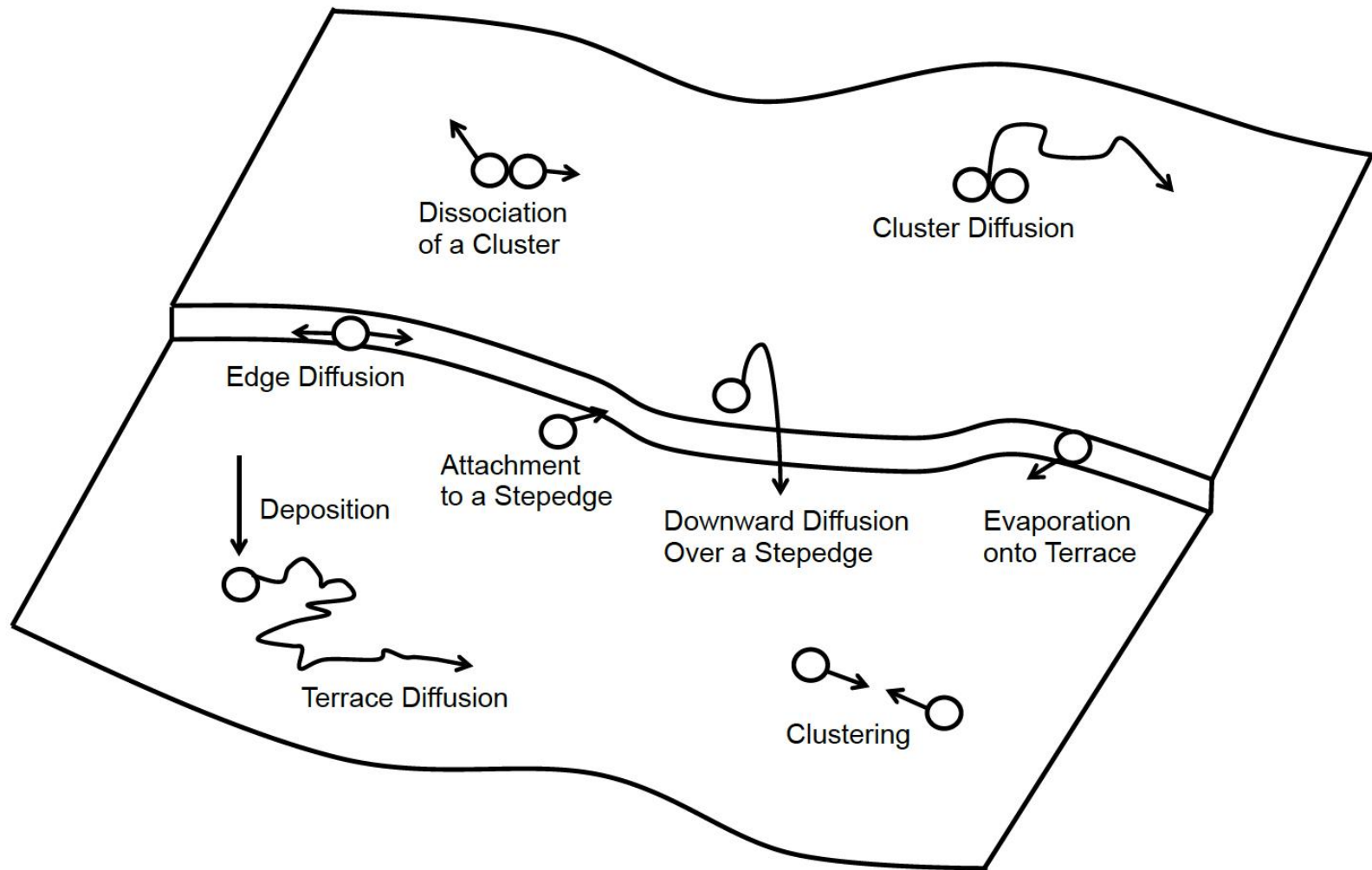


Stanski-Krastanov (first layers, then islands)

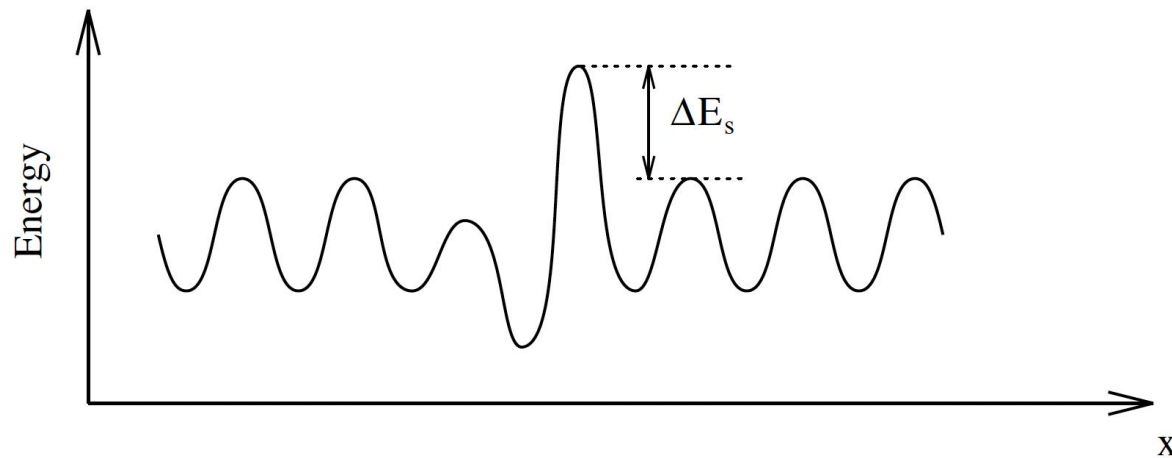
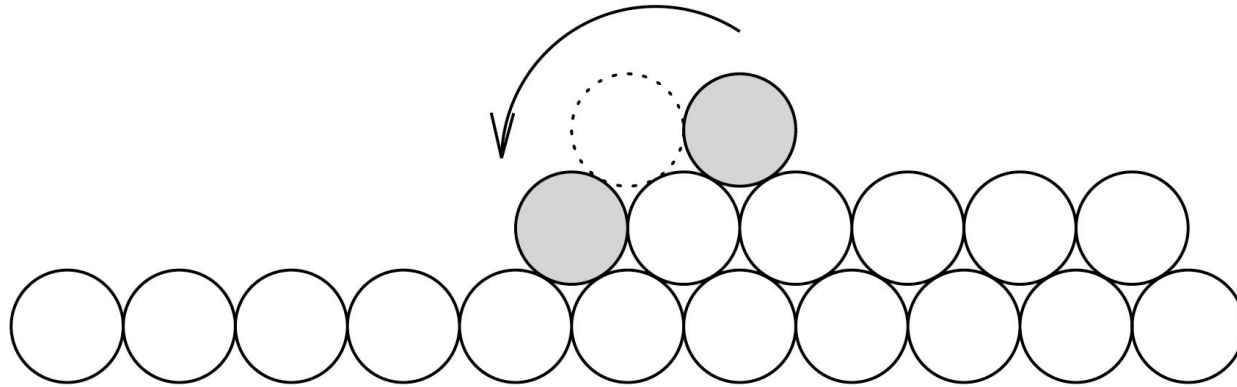


Frank-van-der-Merwe (layer-by-layer)

How to describe growth?



Ehrlich-Schwöbel barrier



Pt adatom on Pt(111) surface:

diffusion $E_D = 0.26$ eV

Ehrlich-Schwöbel barrier $\Delta E_s \approx 0.14$ eV

Adatom diffusion

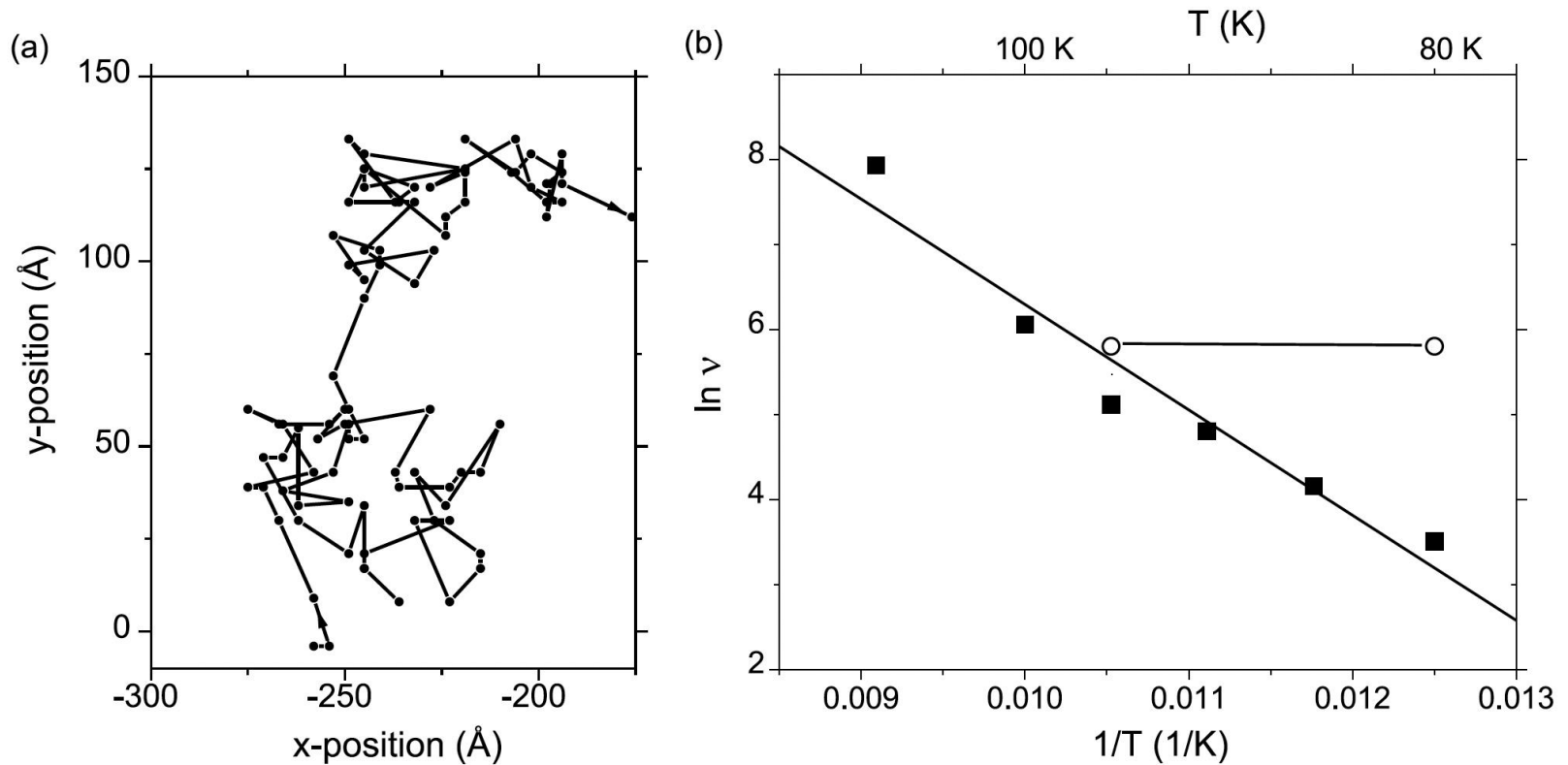
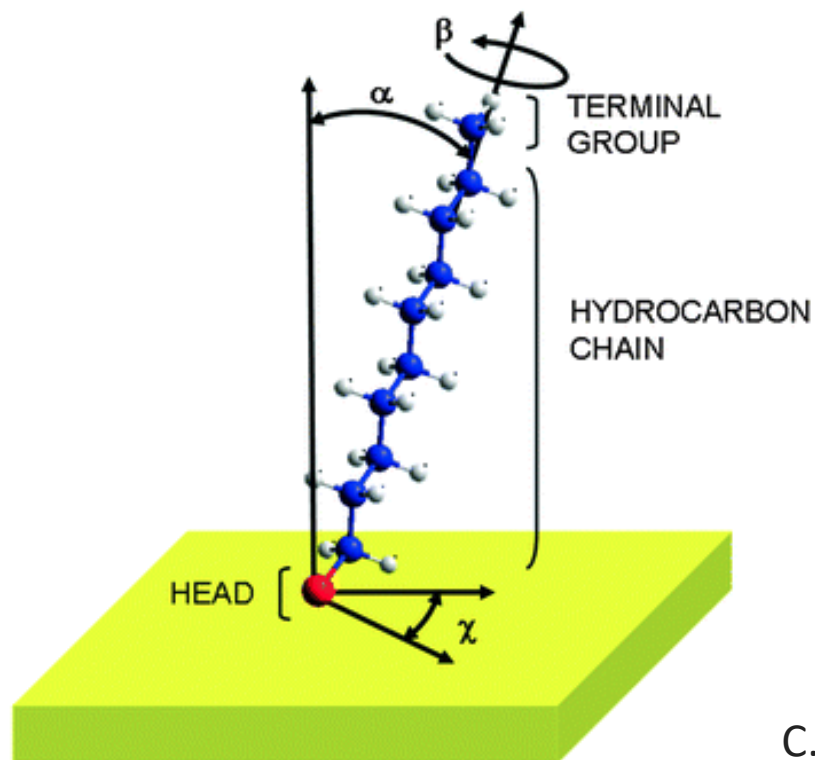
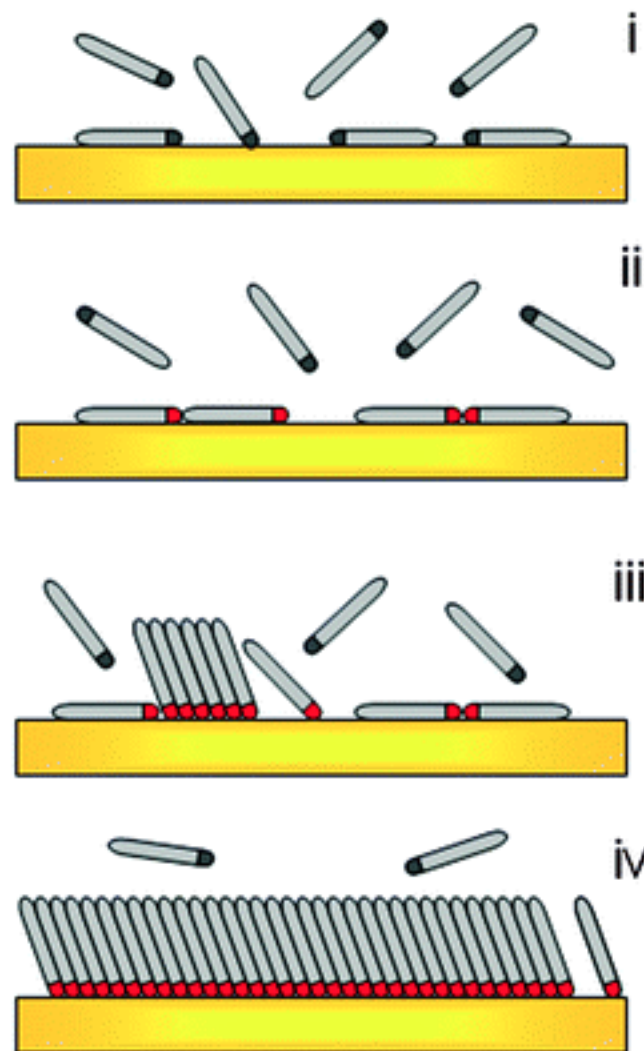
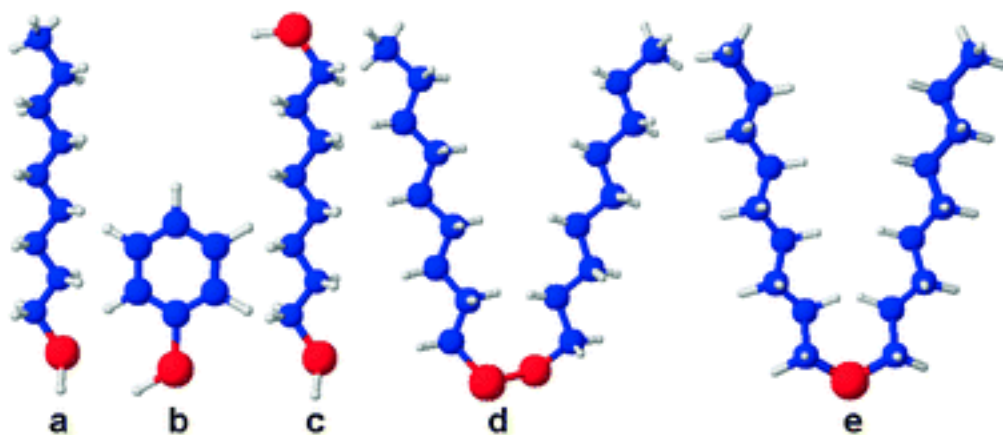


Fig. 2.3. (a) Map of consecutive adatom positions on STM topographs taken at 85 K in time intervals of 30 s. (b) Temperature dependence of the adatom hopping frequency ν determined by maps as in (a). *Full squares*: Data taken with a tunneling resistance of $R_t = 150 \text{ M}\Omega$ ($U_t = 0.03 \text{ V}$, $I_t = 0.2 \text{ nA}$). *Full line*: Linear regression to the data. *Open circles*: Data with $R_t = 15 \text{ M}\Omega$ ($U_t = 0.03 \text{ V}$, $I_t = 2 \text{ nA}$). Line to guide the eye

Thiols on gold



Linear differential equations

Inhomogeneous linear differential equation of the first order with constant coefficients:

$$a_1 \frac{dy}{dt} + a_0 y = f(t)$$

Solution of this equation:

$$y = y_{homo} + y_{inhomo}$$

General solution of the corresponding
homogeneous equation

$$a_1 \frac{dy}{dt} + a_0 y = 0$$

Look for a trial solution in the form

$$y = e^{\lambda t}$$

characteristic equation

$$a_1 \lambda + a_0 = 0$$

$$y_{homo} = C_1 e^{\lambda_1 t}$$

Any solution of the inhomogeneous
equation

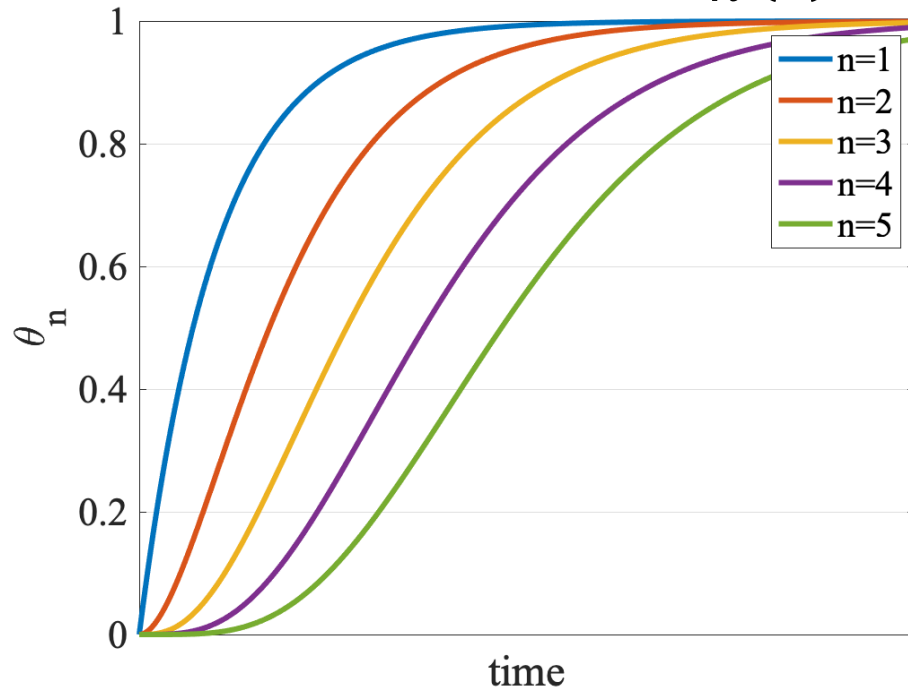
$$a_1 \frac{dy}{dt} + a_0 y = f(t)$$

Look for a trial solution in the form

$$y_{inhomo} = C_1(t) e^{\lambda_1 t}$$

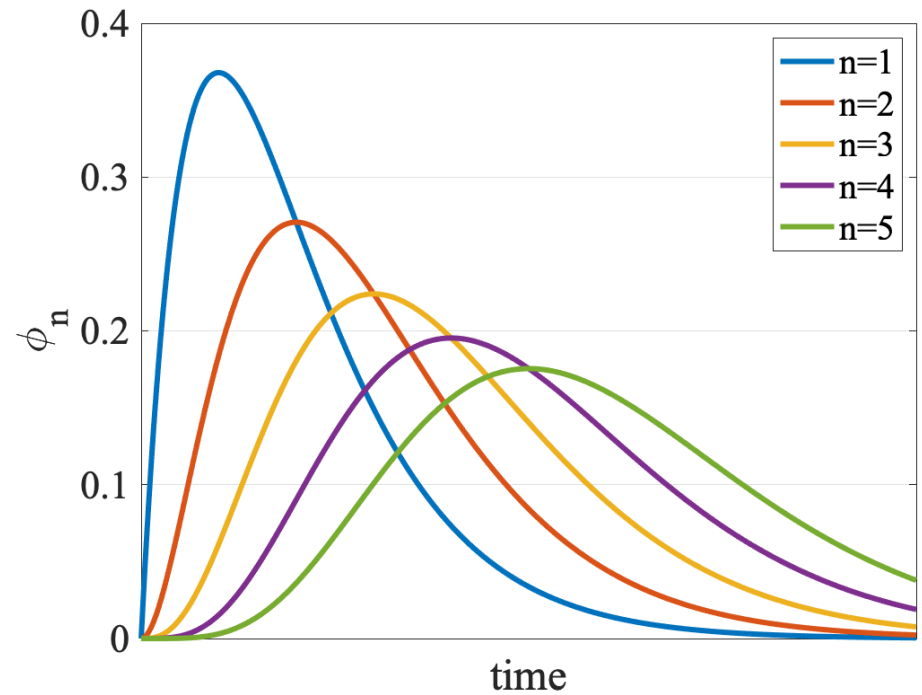
Statistical growth

Layer coverage $\theta_n(t)$

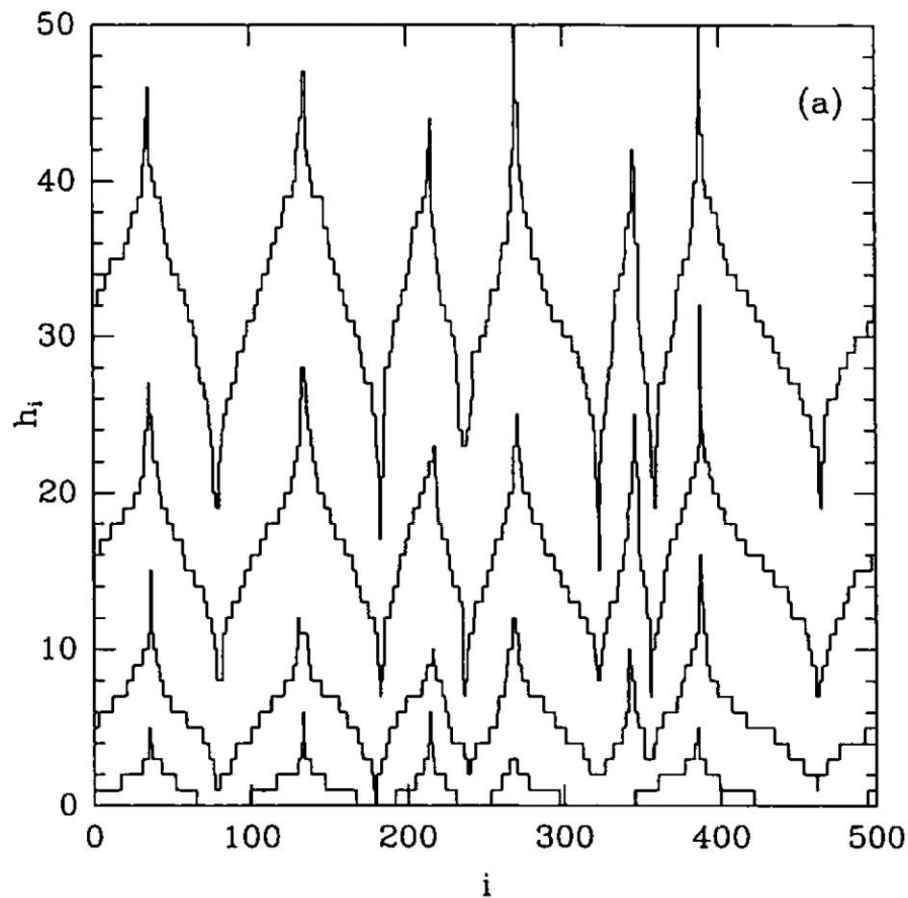


Exposed layer coverage

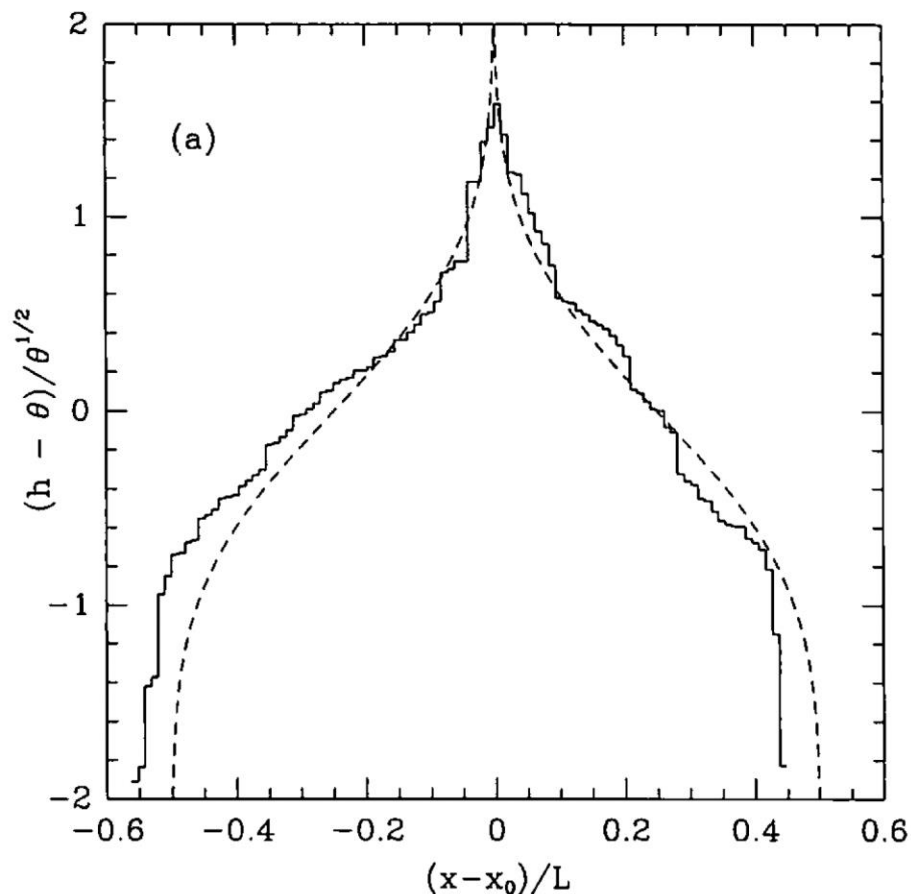
$$\varphi_n = \theta_n - \theta_{n-1}$$



Wedding cake shape



Line cut through the surface at different deposition times (simulations)



Fit of the "wedding cake" profile with the error function

Wedding cake shape

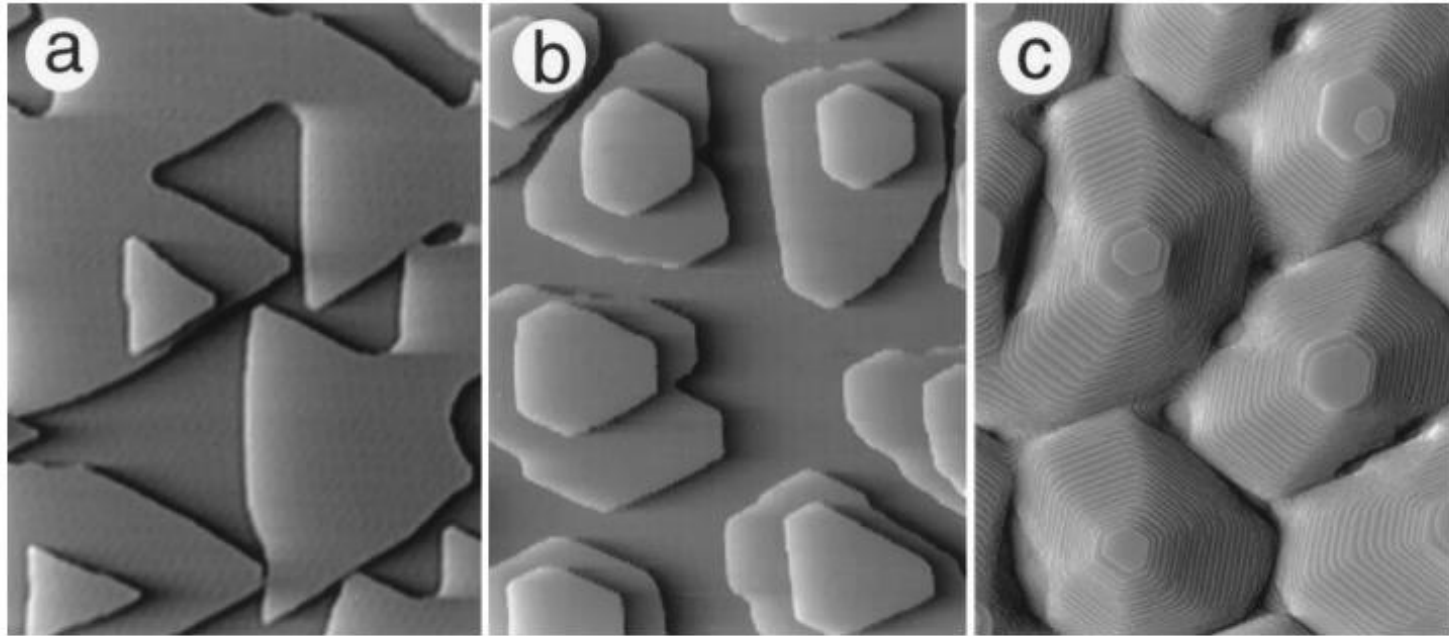


Fig. 4. STM topographs after deposition of (a) 0.78 ML, (b) 0.83 ML, and (c) 40 ML Pt at $T=440$. Deposition (a) under clean conditions and (b), (c) in the presence of 1.9×10^{-9} mbar of CO. Scan size $1610 \times 1200 \text{ \AA}^2$.

Island size and shape

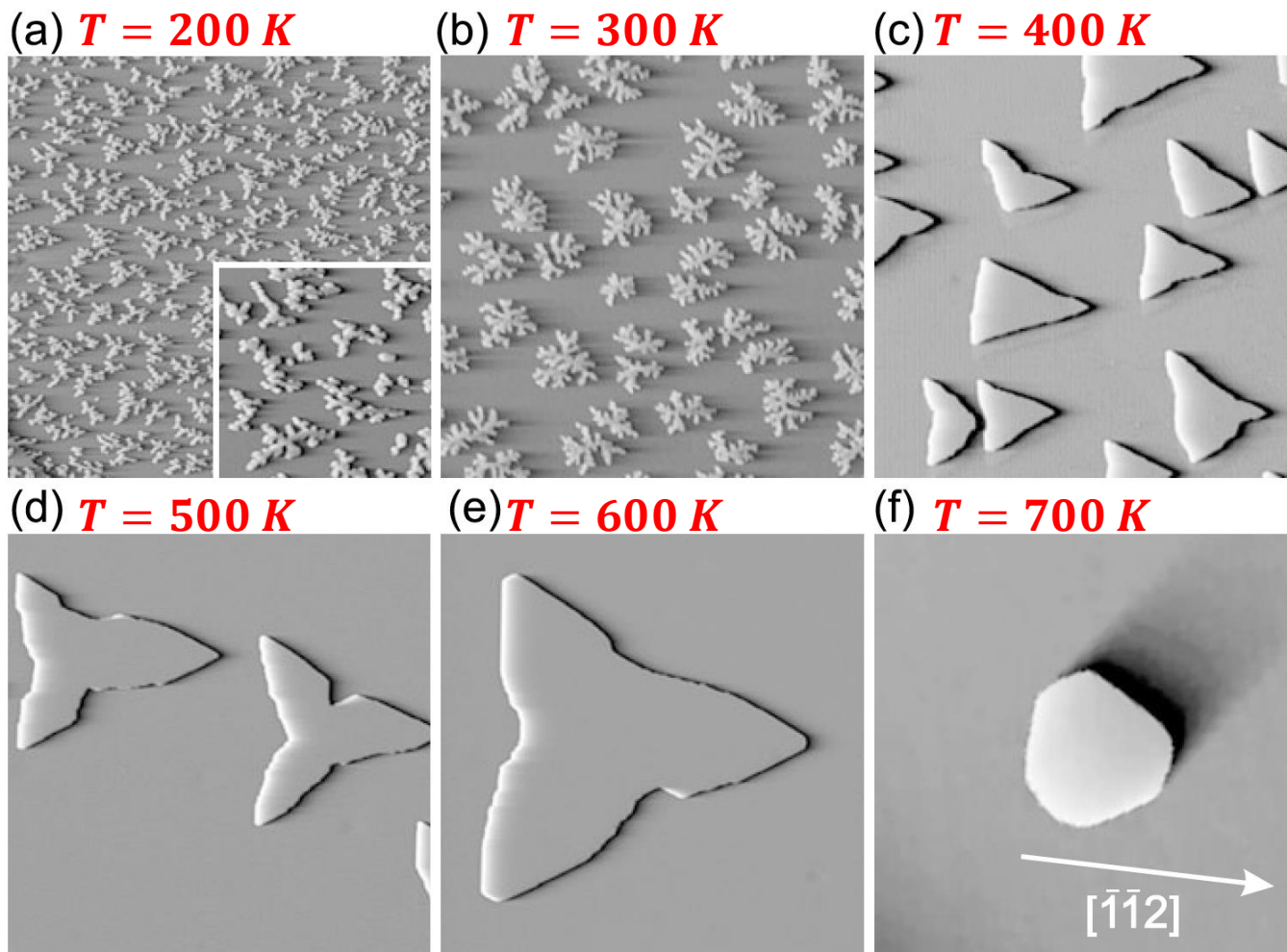
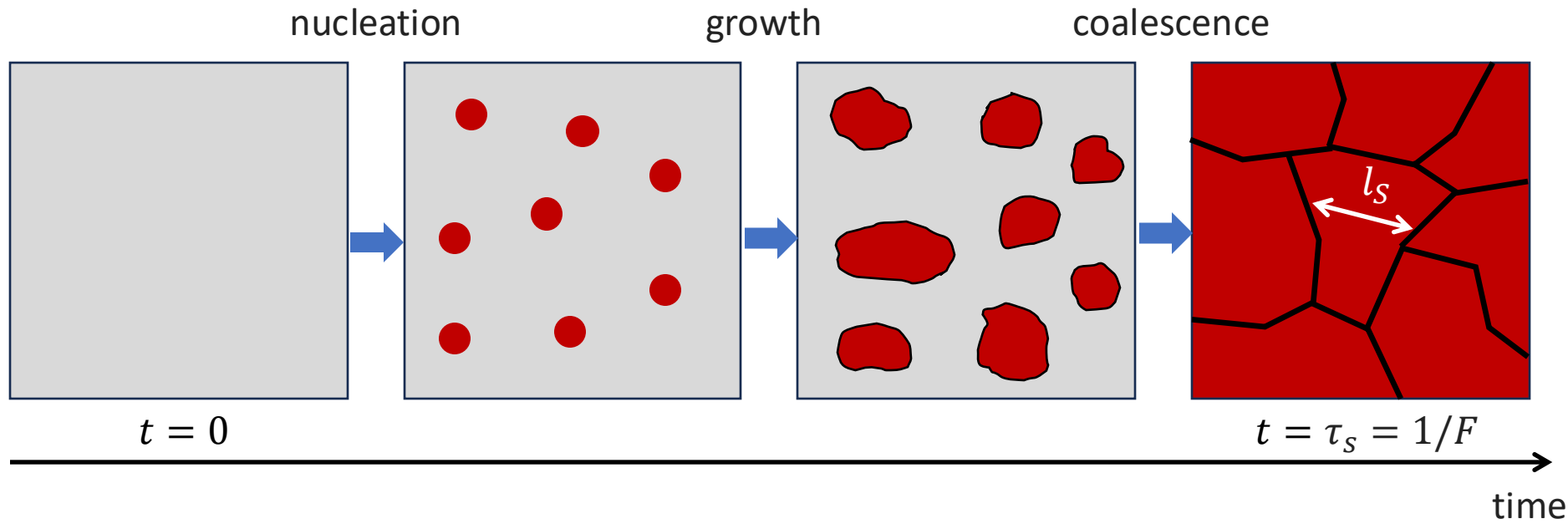


Fig. 3.1. STM topographs of adatom islands on Pt(111) after deposition of Pt at (a) 200 K, (b) 300 K, (c) 400 K, (d) 500 K, (e) 600 K and (f) 700 K. The deposited amount is 0.15 ML in (a)–(e) and 0.08 ML in (f), the deposition rate is 7×10^{-3} ML/s in (a)–(e) and 2.7×10^{-3} ML/s in (f). The topograph size is always $1560\text{ \AA} \times 1560\text{ \AA}$

Growth of 2D clusters



Deposition rate: F [ML/sec]

Time until coalescence: $\tau_s = 1/F$ [sec]

size of a cluster: l_s

area of a cluster: l_s^2

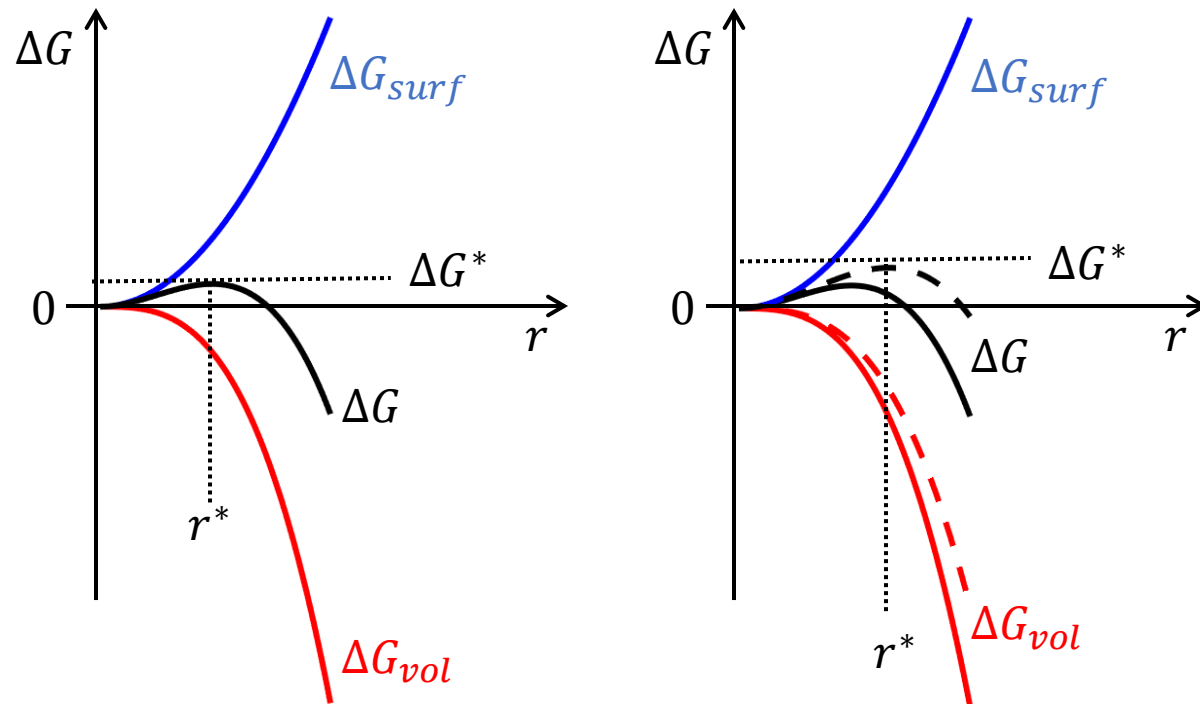
number of a clusters per unit area: $1/l_s^2$

Classical nucleation theory (3D)

The change in Gibbs free energy when a nucleus is formed:

$$\Delta G = \Delta G_{vol} + \Delta G_{surf} = \frac{4}{3}\pi r^3 \cdot \Delta g_v + 4\pi r^2 \cdot \gamma$$

$$\Delta g_v \propto -\frac{k_B T}{v} \ln S \quad (v - \text{molar volume, } S - \text{supersaturation})$$



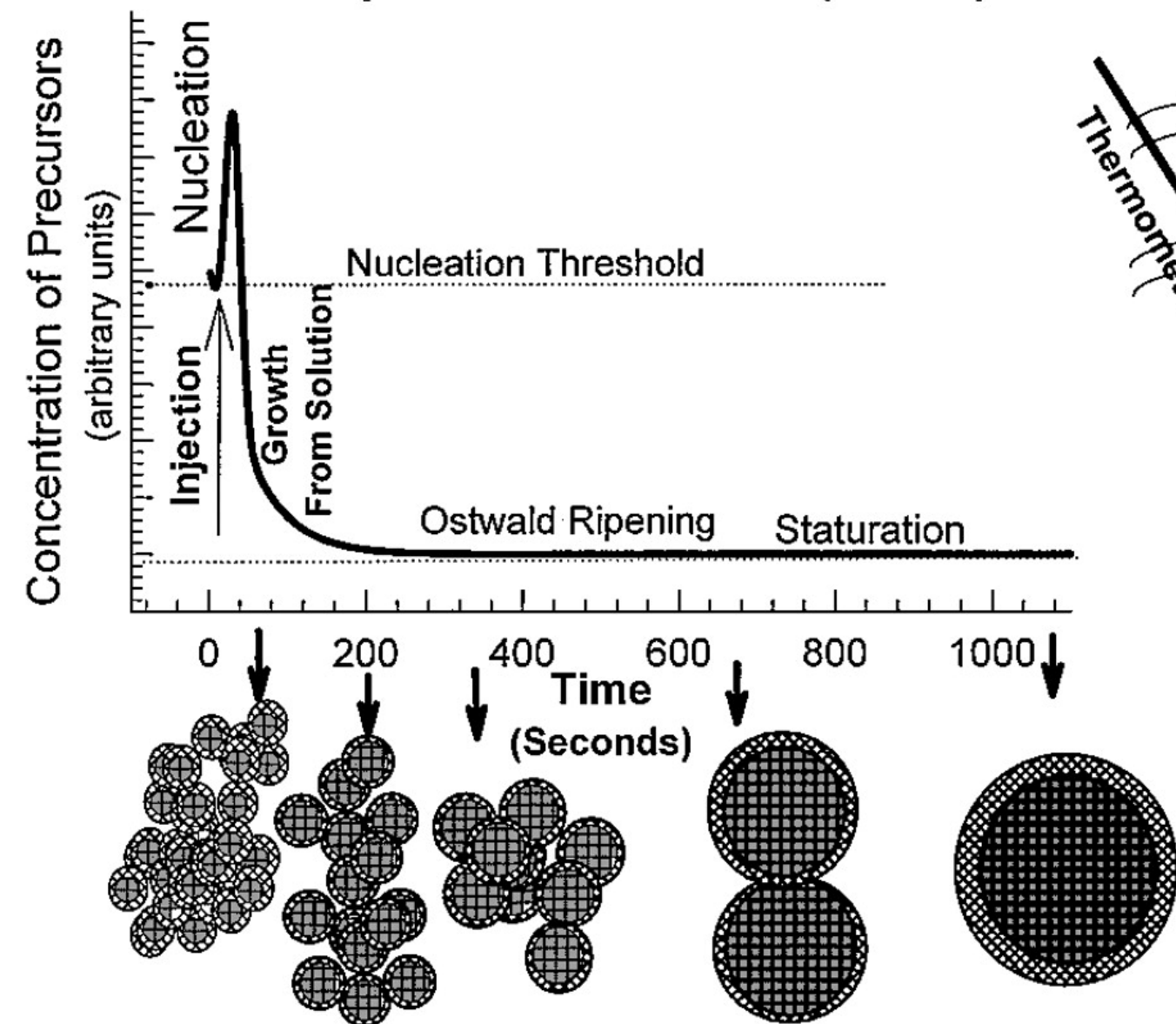
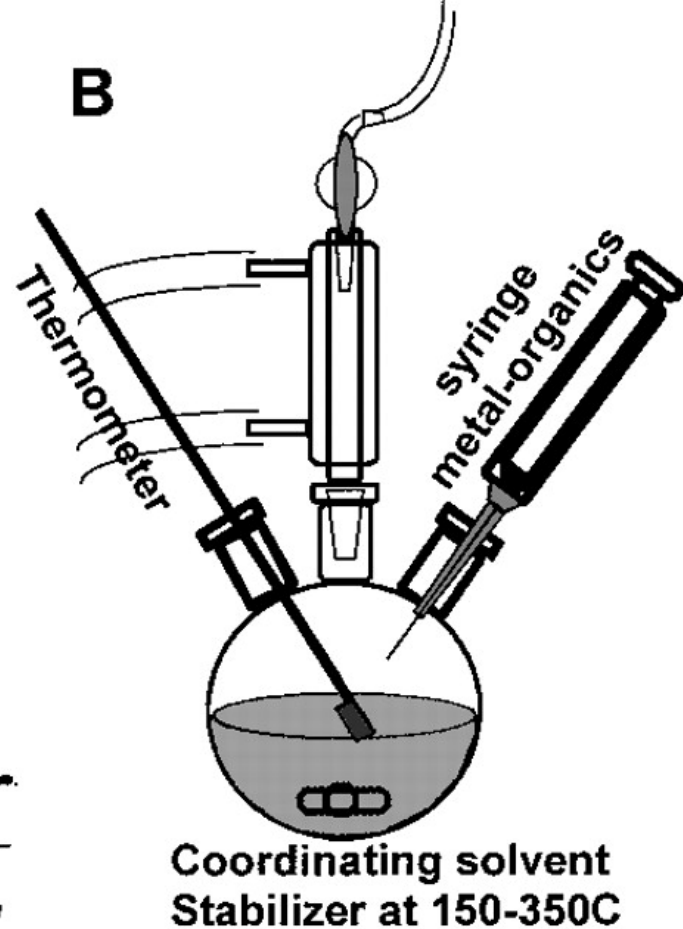
Nucleation rate:

$$\frac{dN}{dt} \propto \exp \left[-\frac{\Delta G^*}{k_B T} \right]$$

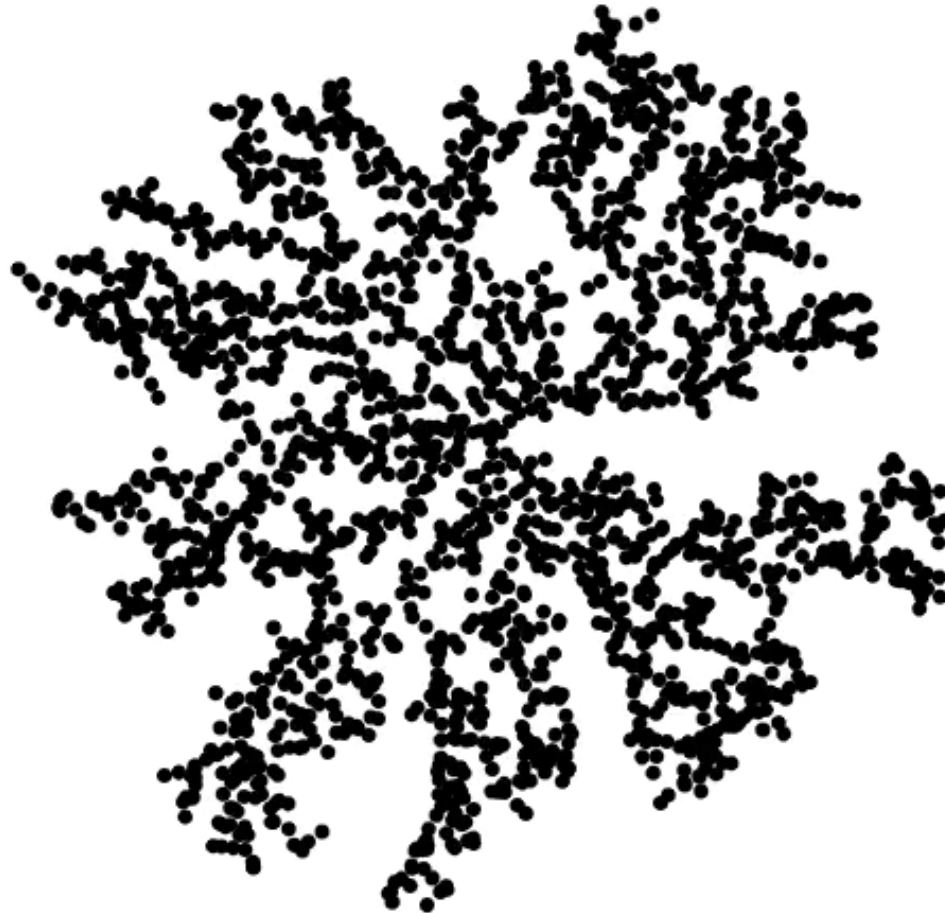
dashed lines – lower temperature (and a larger critical radius)

A

Monodisperse Colloid Growth (La Mer)

**B**

Diffusion limited aggregation



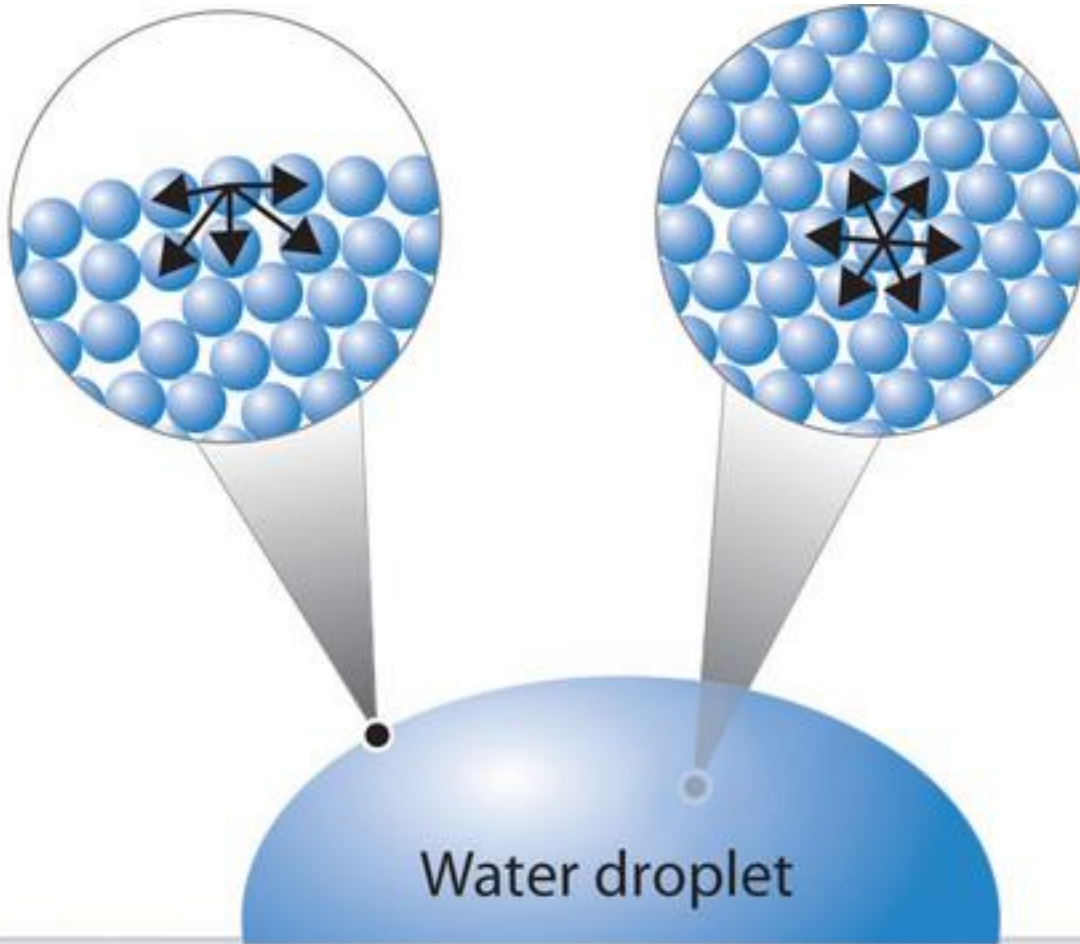
Pt adatom on Pt(111) surface:

diffusion $E_D = 0.26$ eV

step edge diffusion $E_a \geq 0.6$ eV

<https://flocc.network/model/diffusion-limited-aggregation>

Surface tension



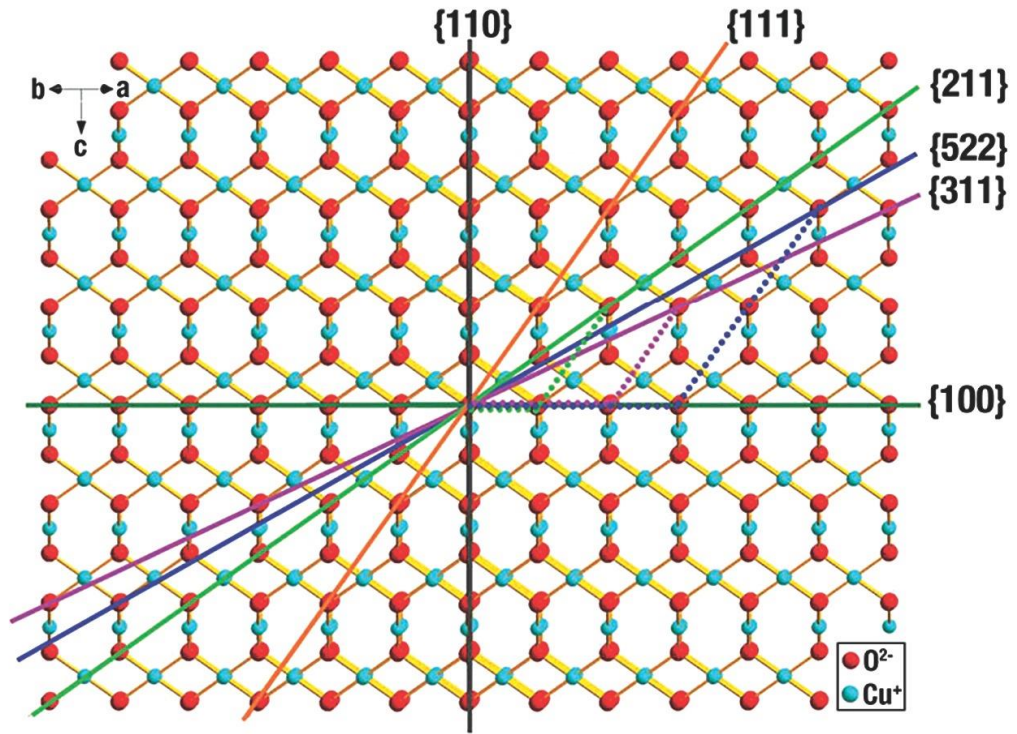
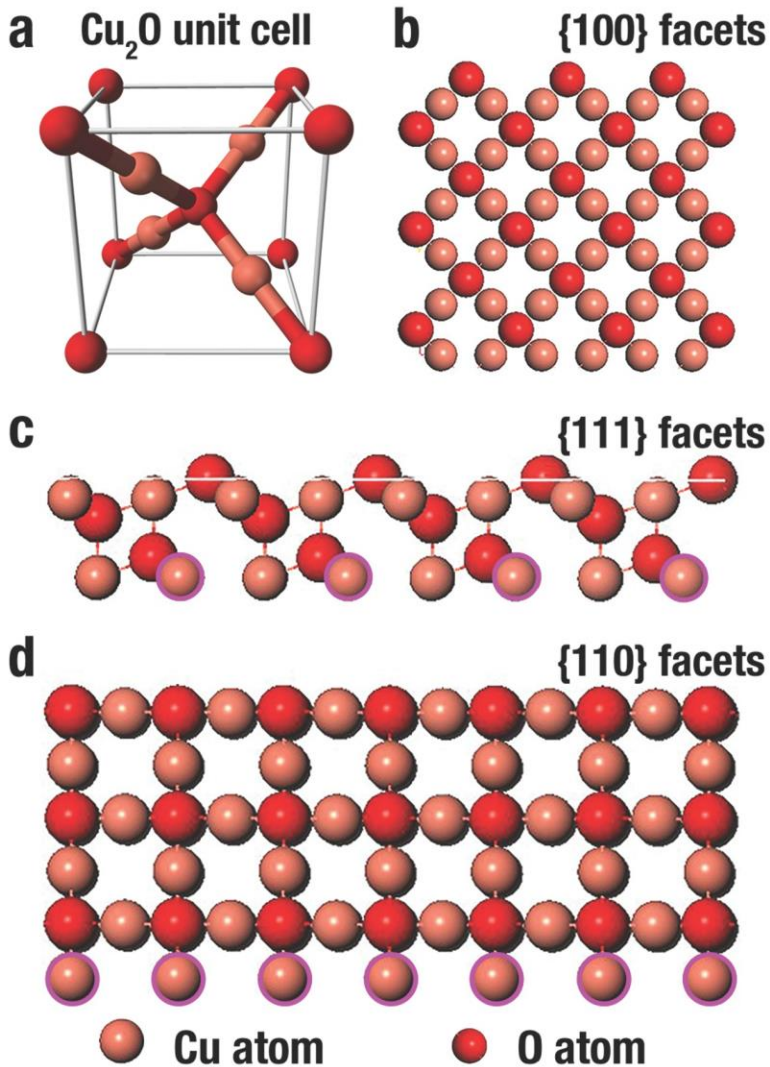
$$\gamma = \frac{\Delta E}{\Delta S}$$

excess energy per surface area;
energy that is needed to create
additional surface with area ΔS

$$\text{H}_2\text{O}: \gamma = 72 \frac{\text{mN}}{\text{m}} = 4.5 \frac{\text{meV}}{\text{\AA}^2}$$

$$\text{Hg}: \gamma = 487 \frac{\text{mN}}{\text{m}} = 30 \frac{\text{meV}}{\text{\AA}^2}$$

Equilibrium shape of an island



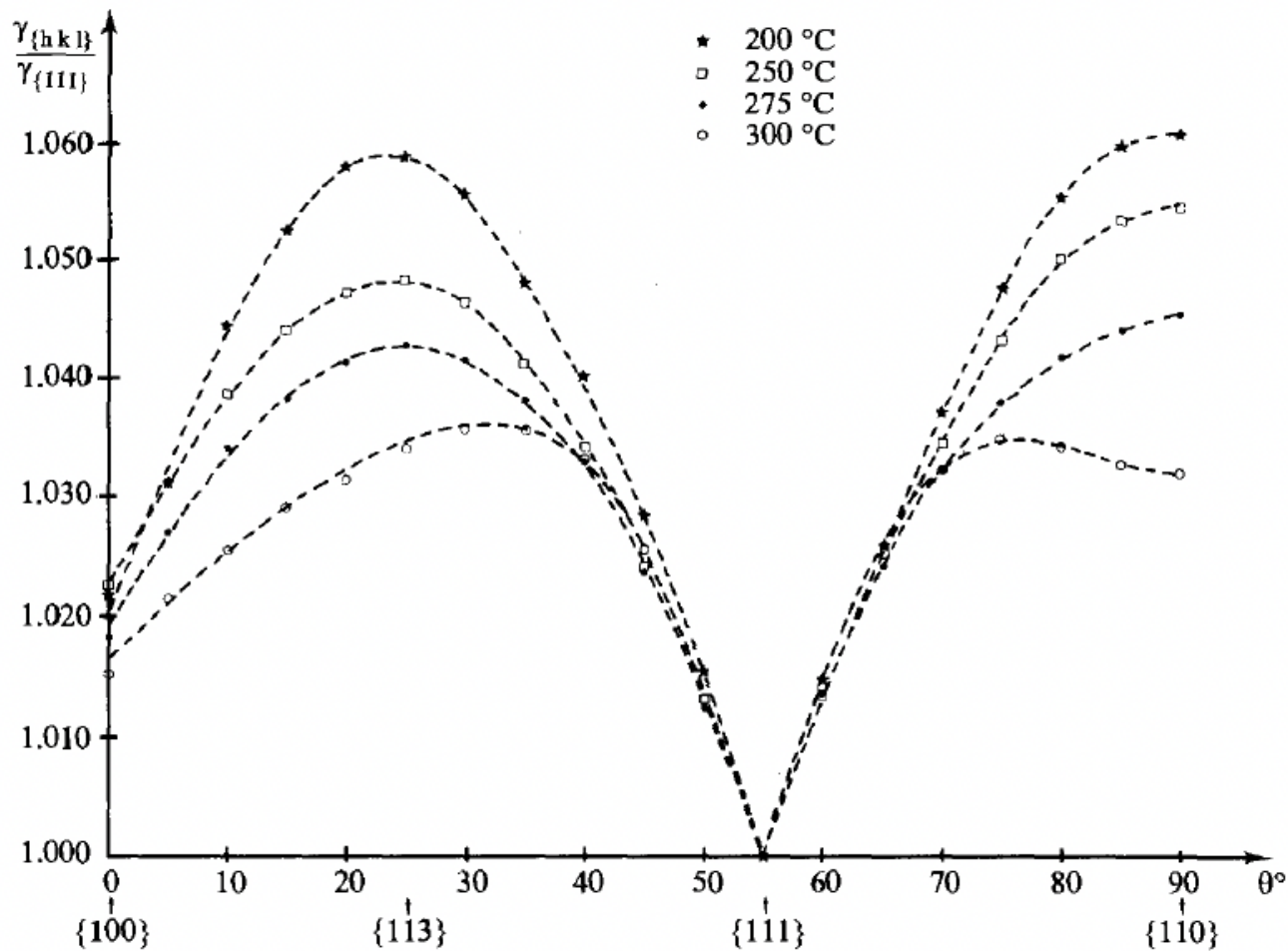
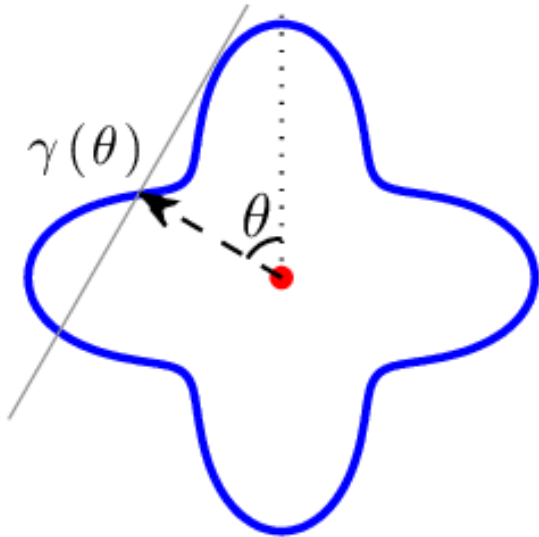
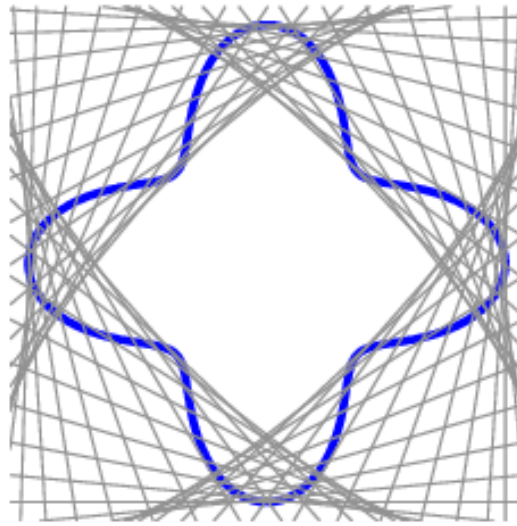


Fig. 2.2. The surface tension of indium at various temperatures, for the orientations $\{11k\}$, normalized to the value for $\{111\}$. Here, k is larger than 1 to the left and smaller than 1 to the right of the cusp. Note the weak variation (visually amplified by the choice of the normalization) and the roughening singularities (cusps) in the directions $\{100\}$ and $\{111\}$. The function ϕ has similar singularities (Heyraud & Métois 1986, with permission by the authors).

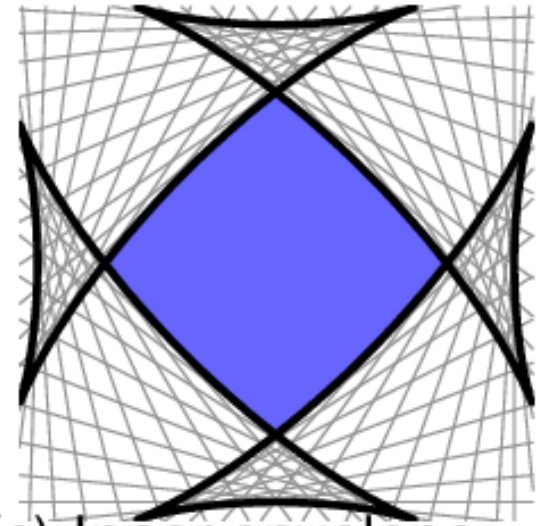
Equilibrium shape of an island (Wulff construction)



(a) γ -plot

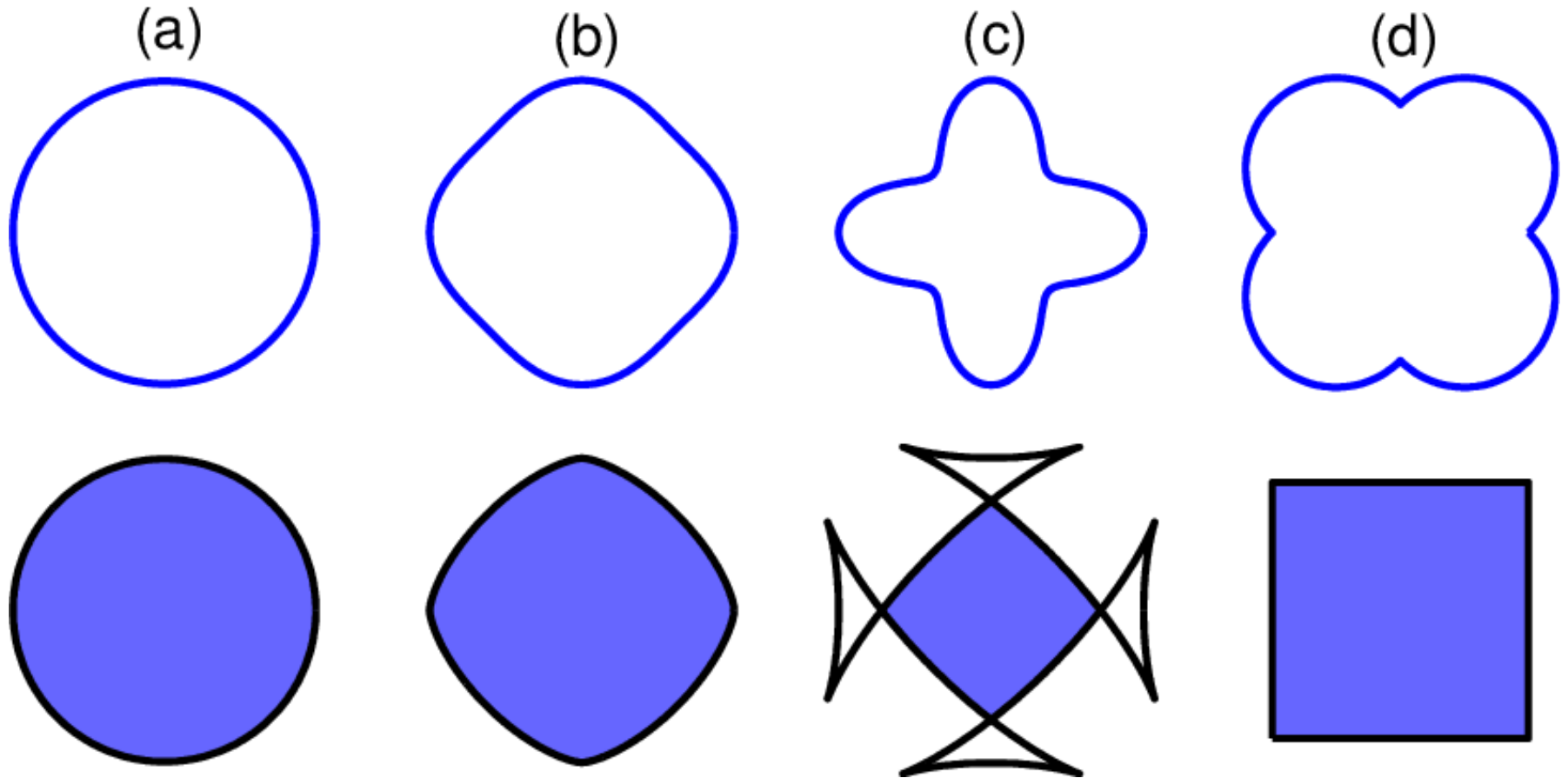


(b) Draw normals

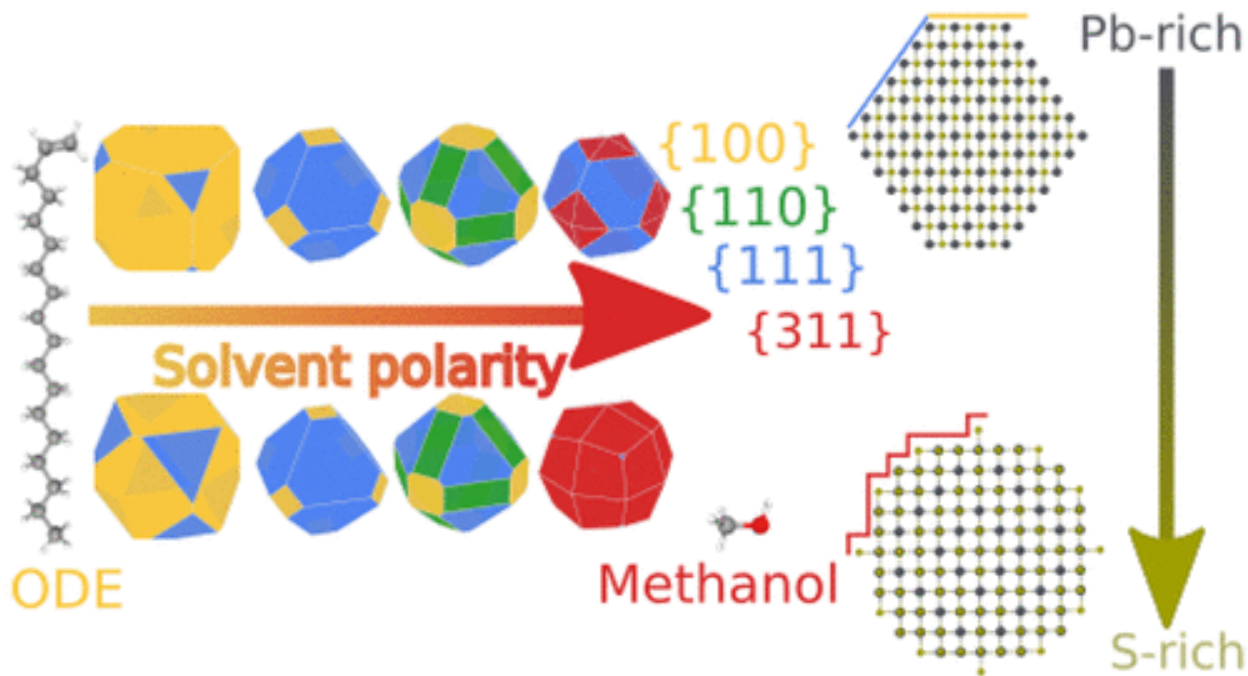


(c) Inner envelope =
equilibrium shape

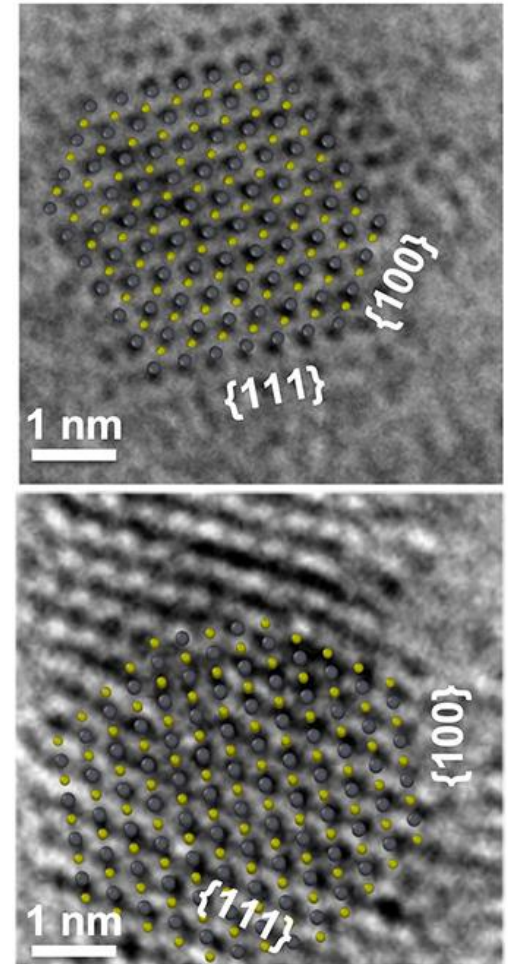
Equilibrium shape of an island (Wulff construction)



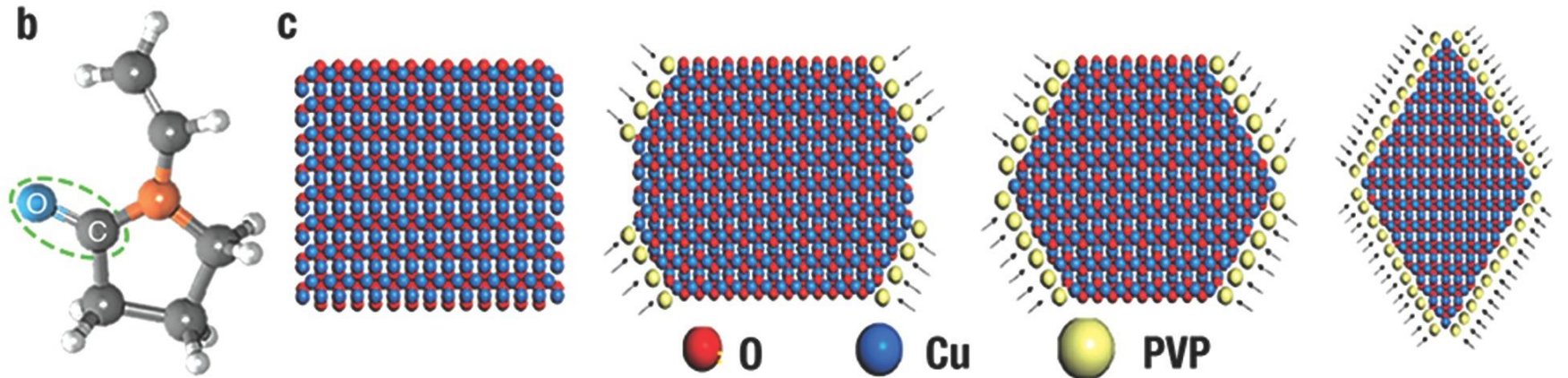
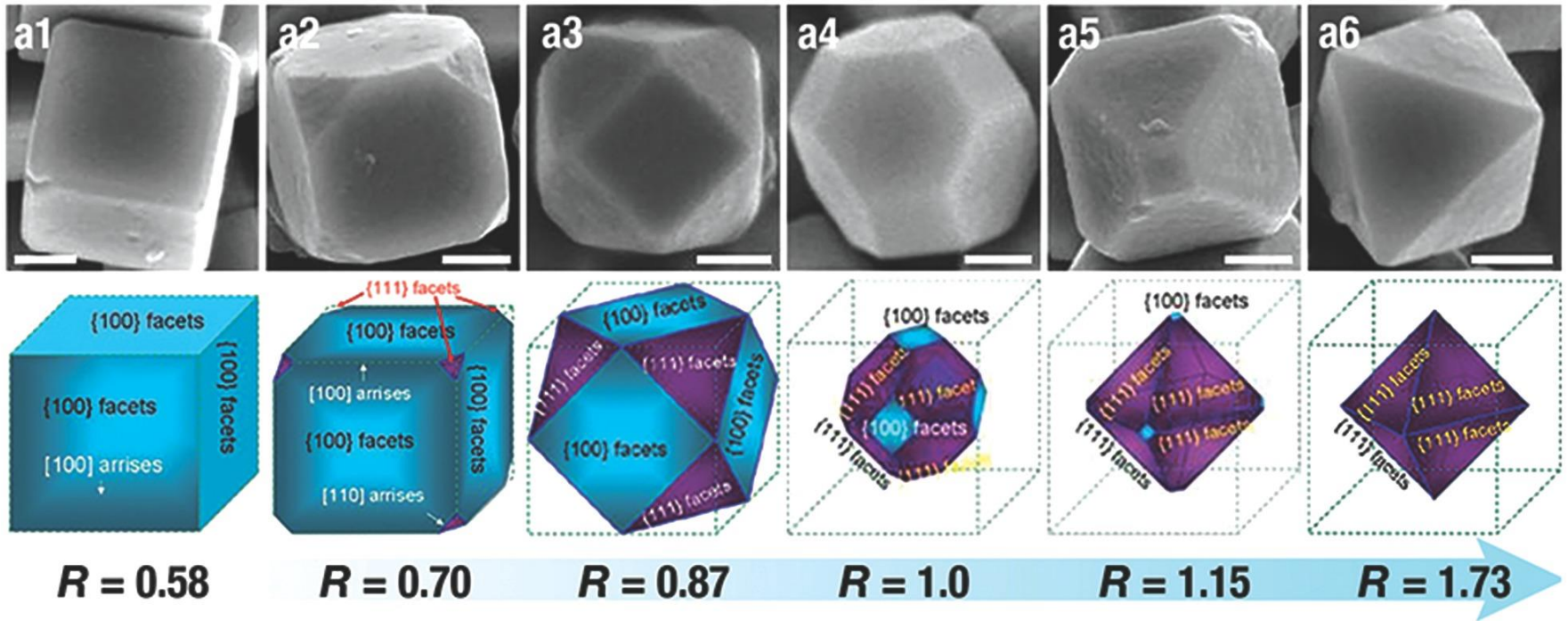
Equilibrium shape of nanoparticles



HRTEM of PbS/OA



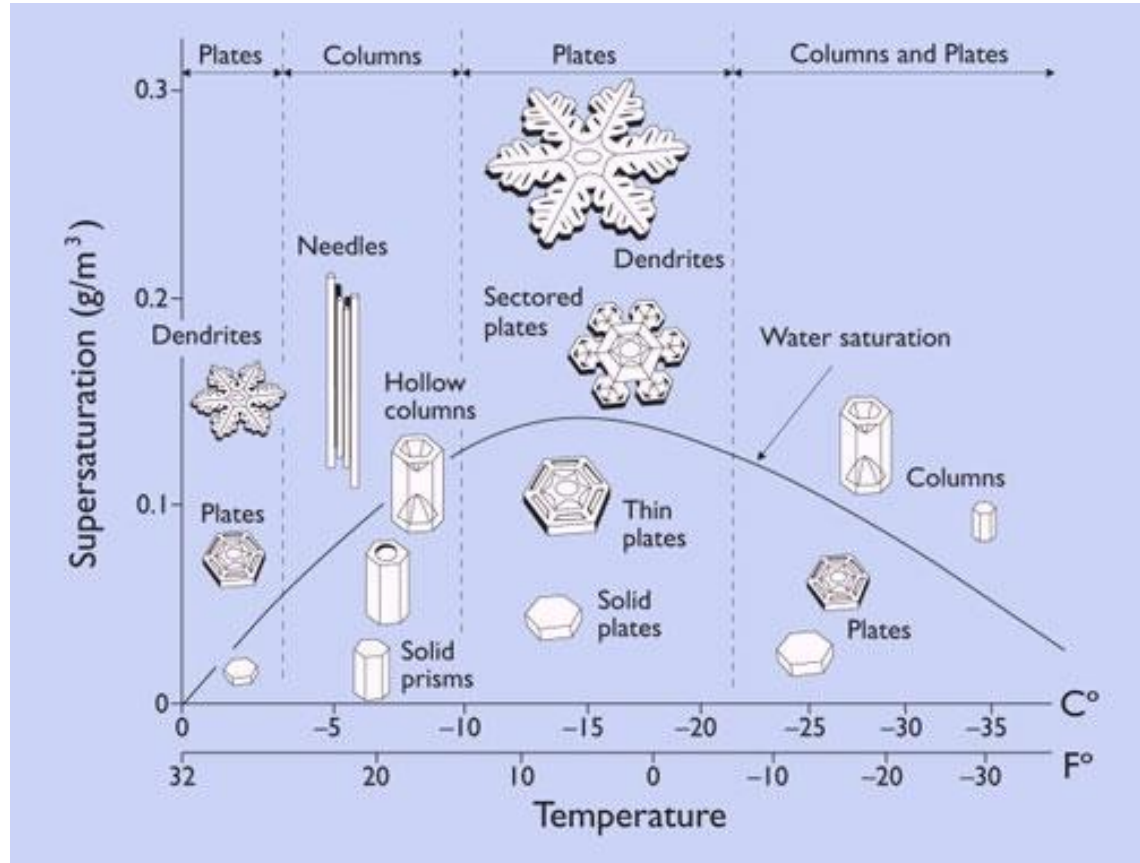
Kinetic control of the nanoparticle shape



Growth of snowflakes

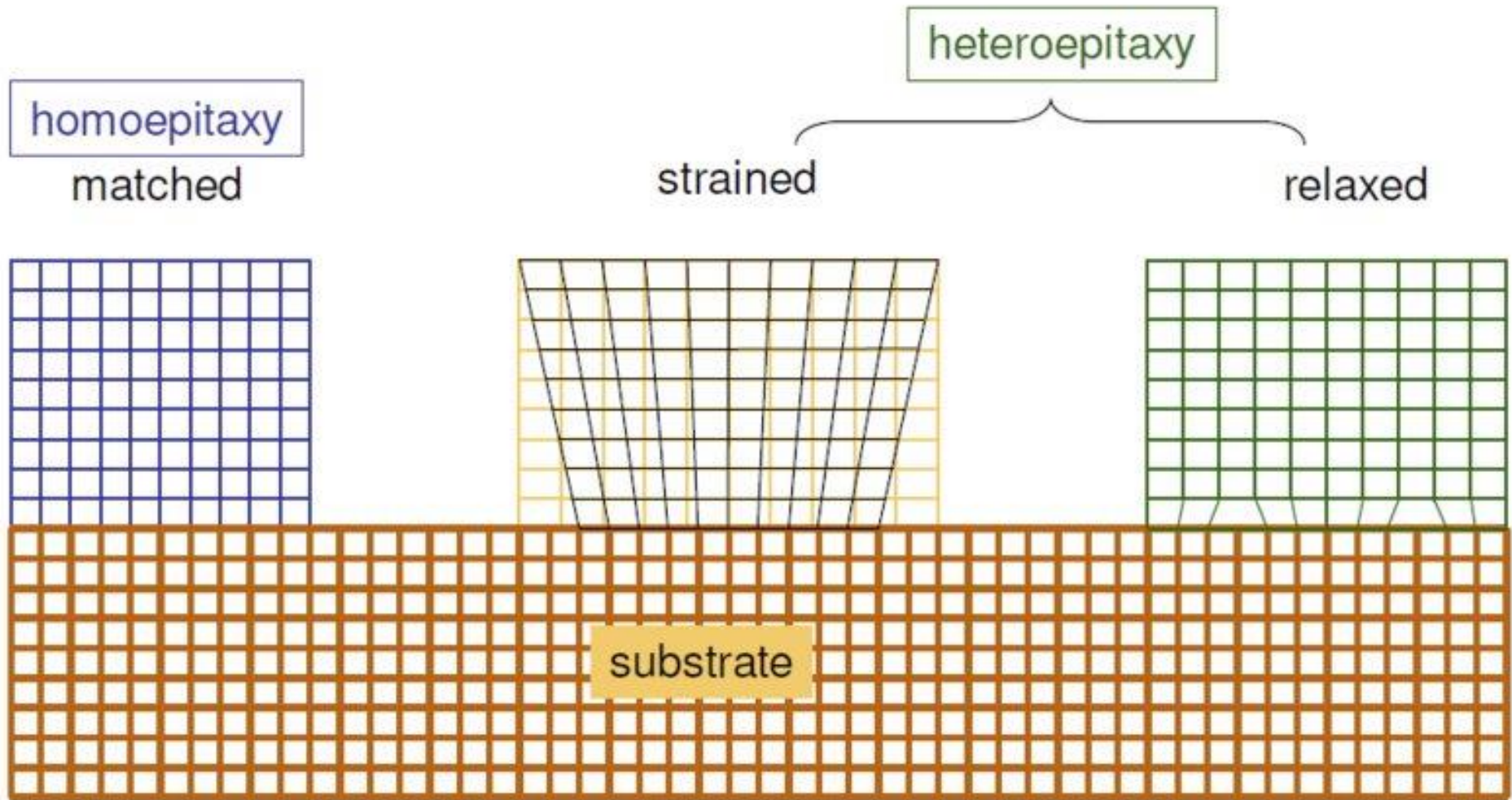


its.caltech.edu

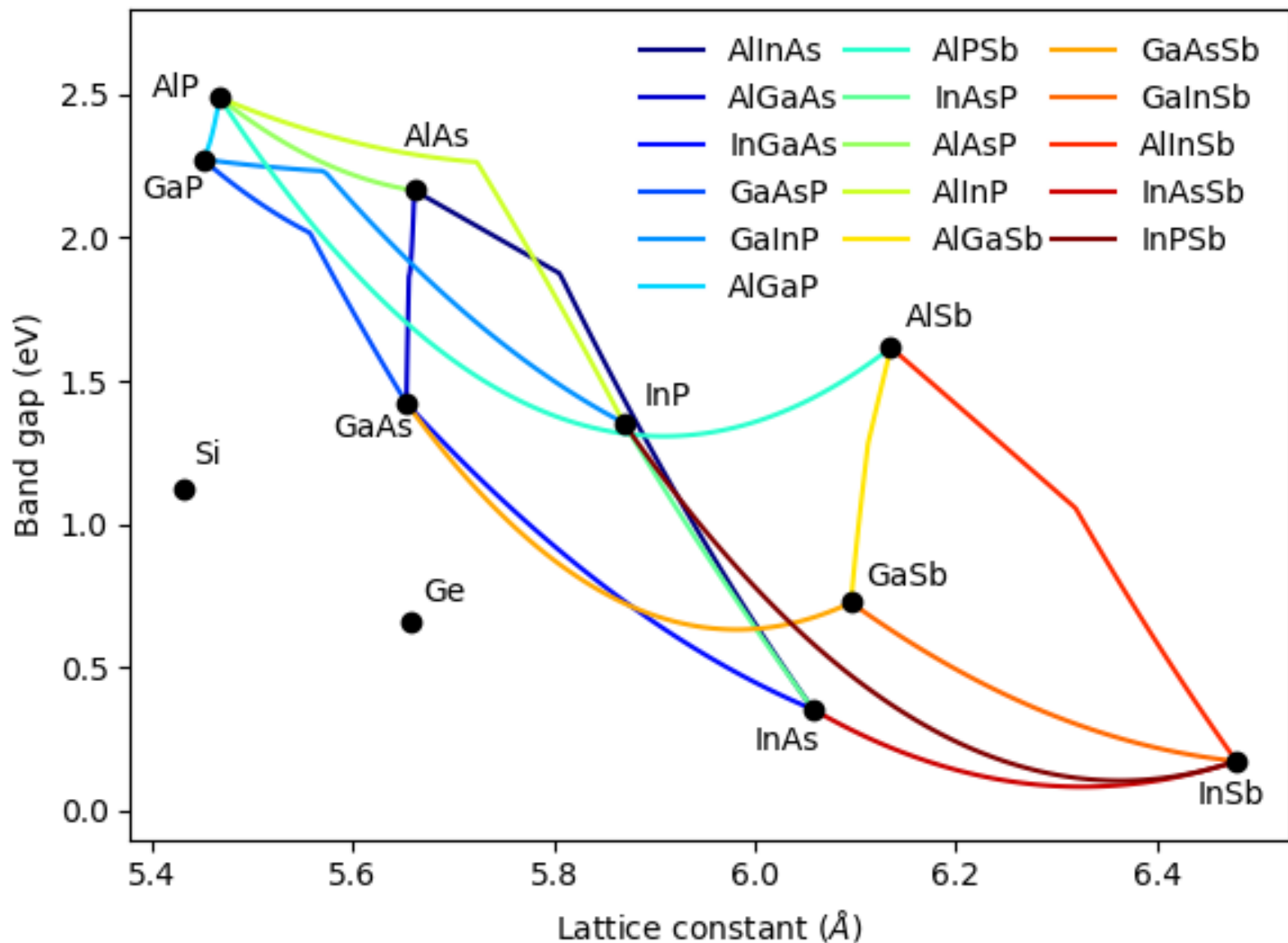


earthsky.org

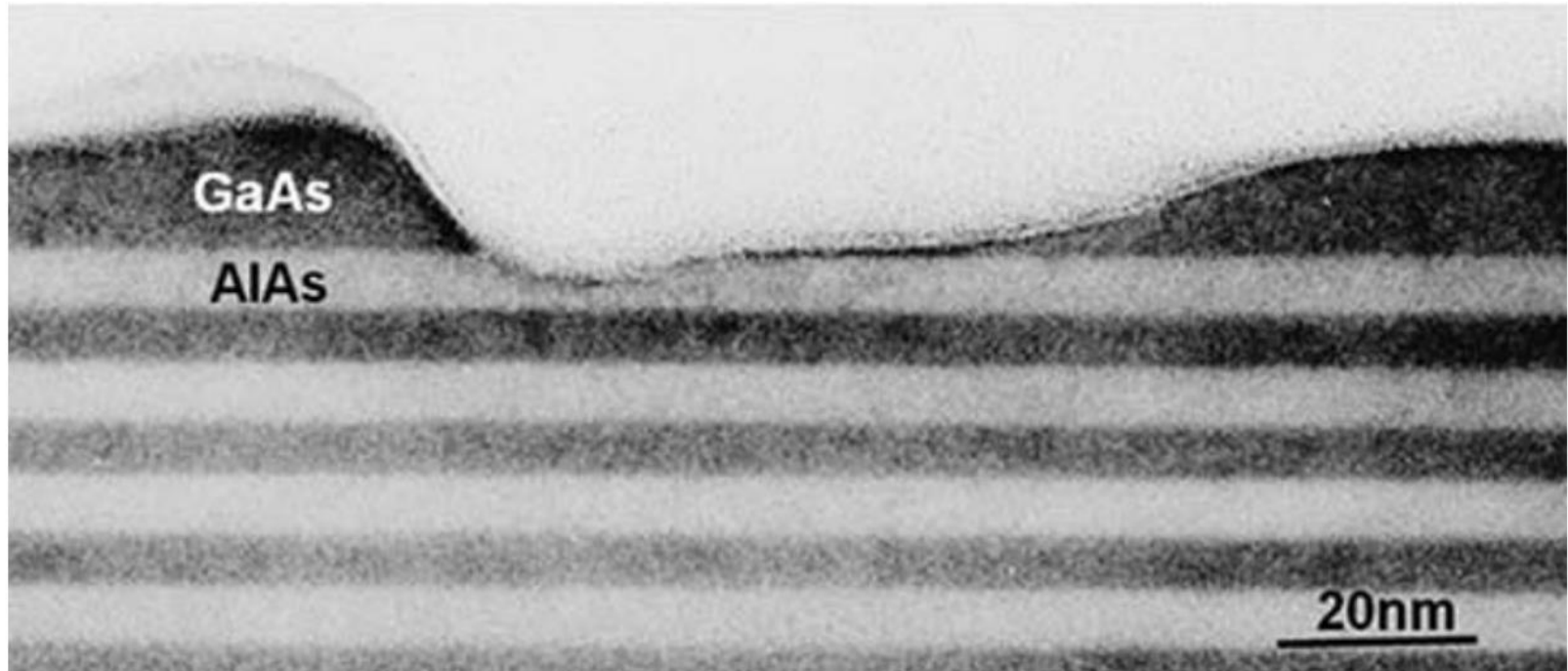
Epitaxy



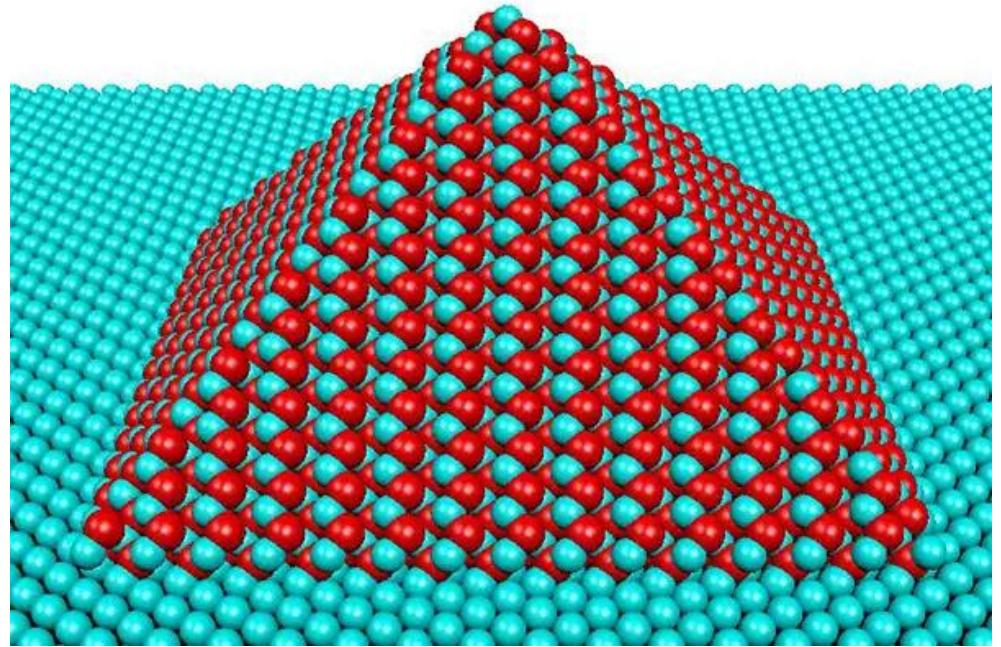
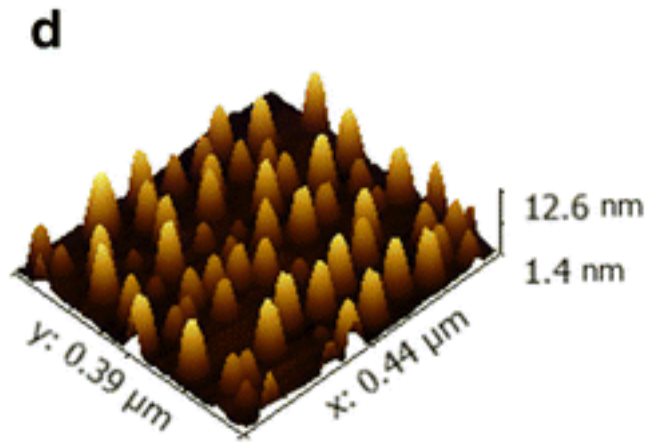
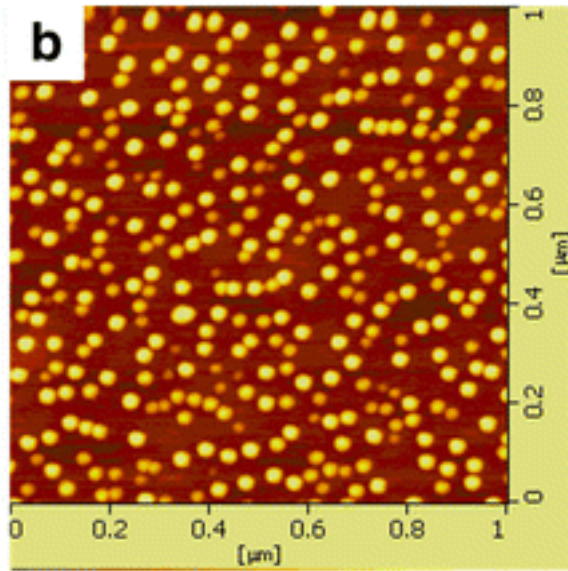
Band gap vs. lattice constant



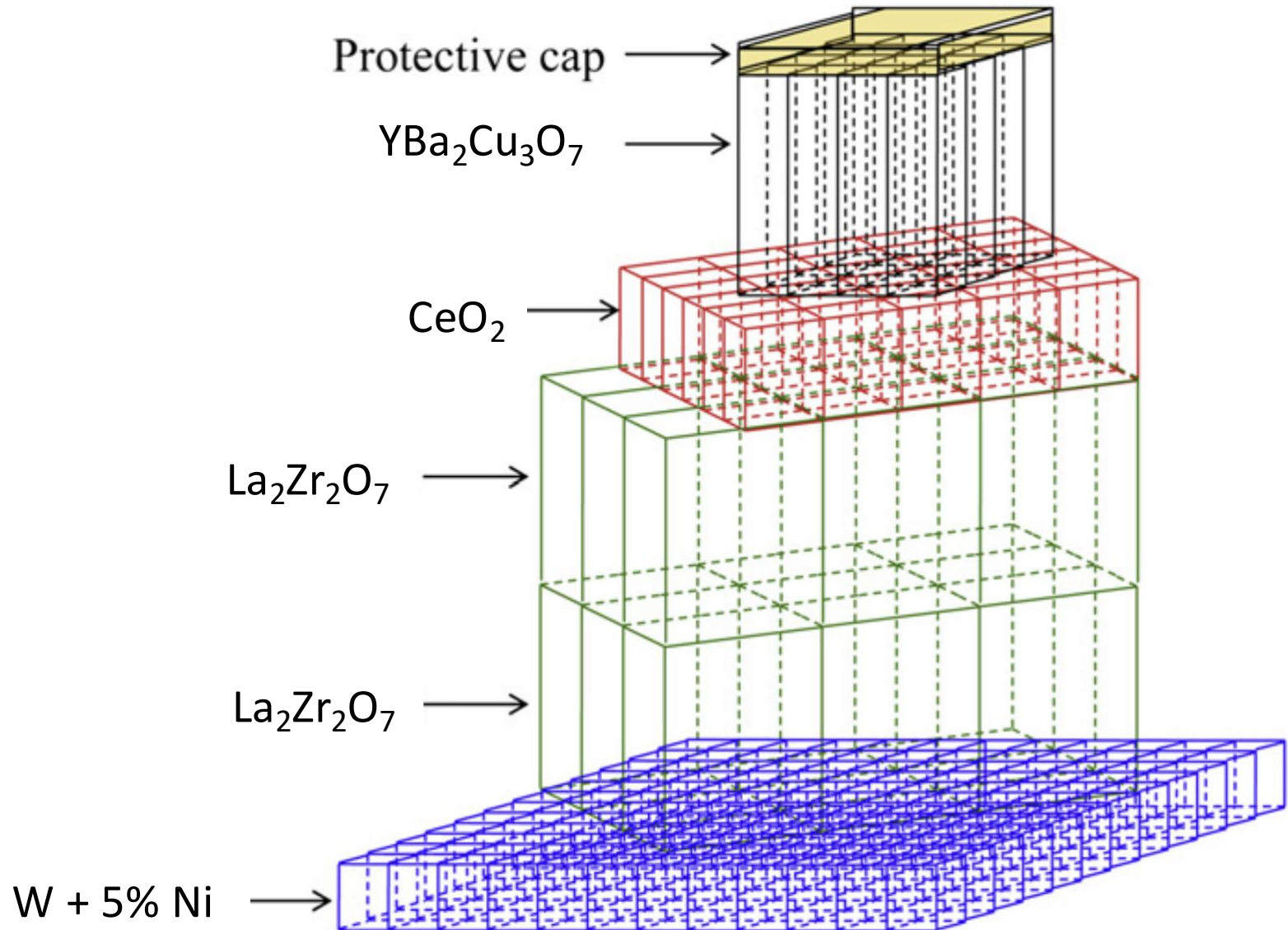
No strain – growth of thin films is possible



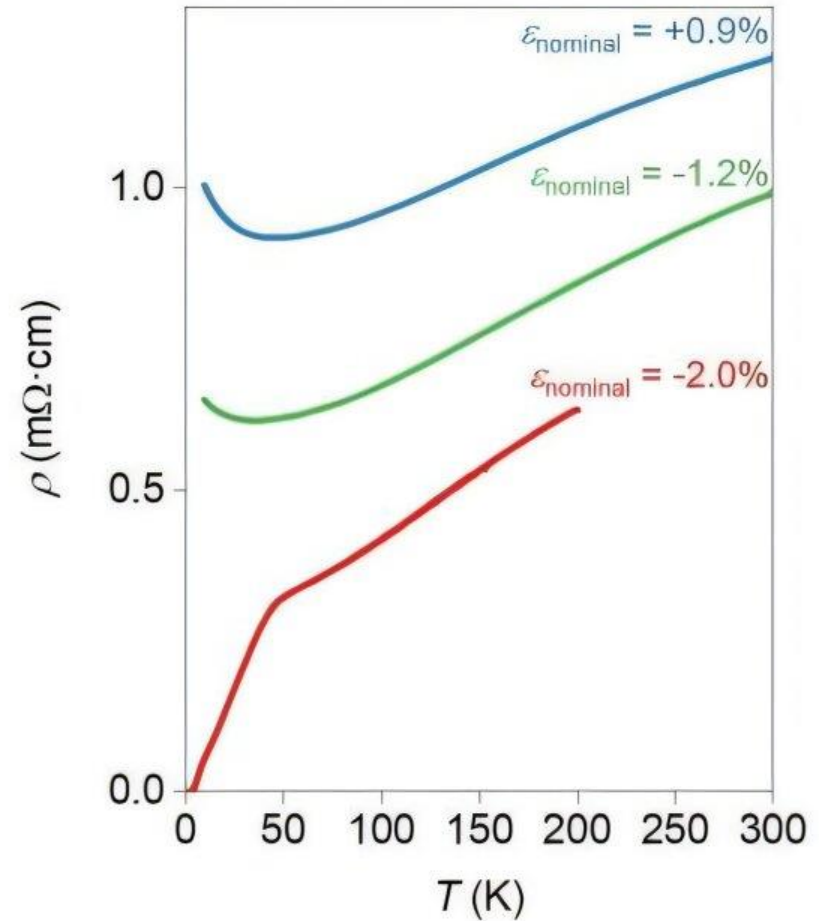
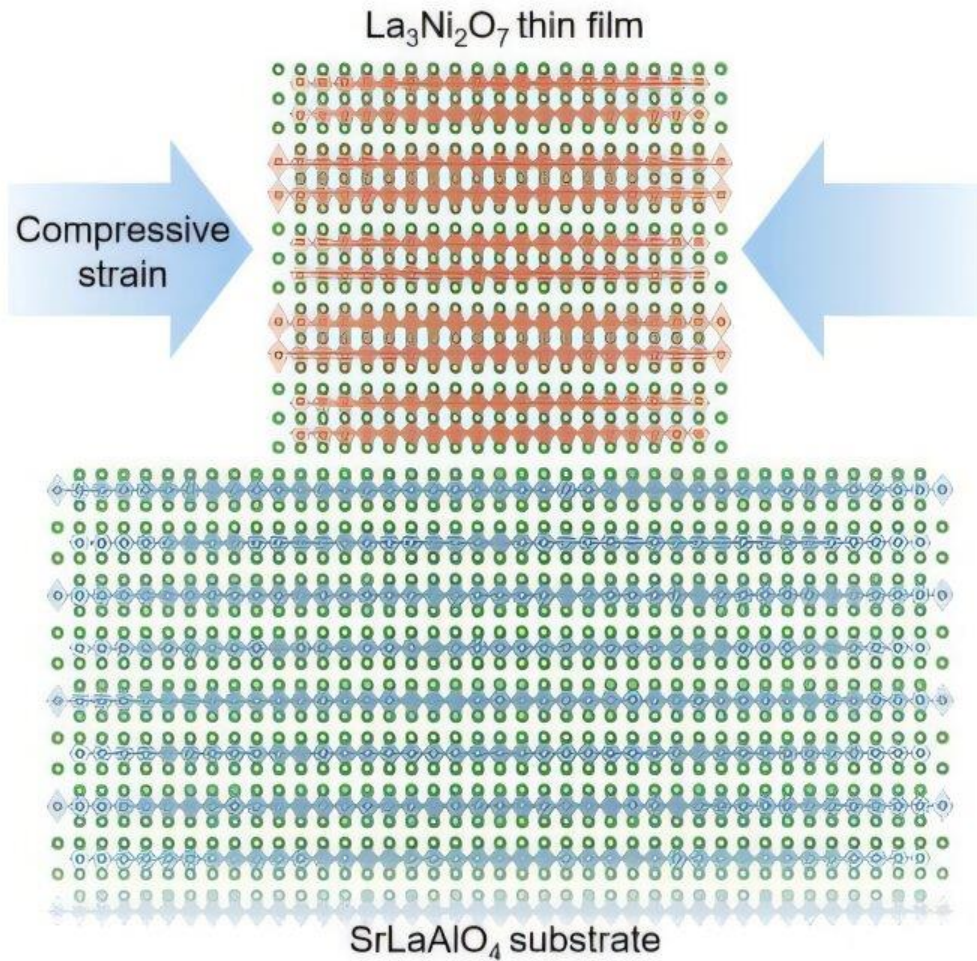
Strain leads to formation of dots



Buffer layer for epitaxial growth

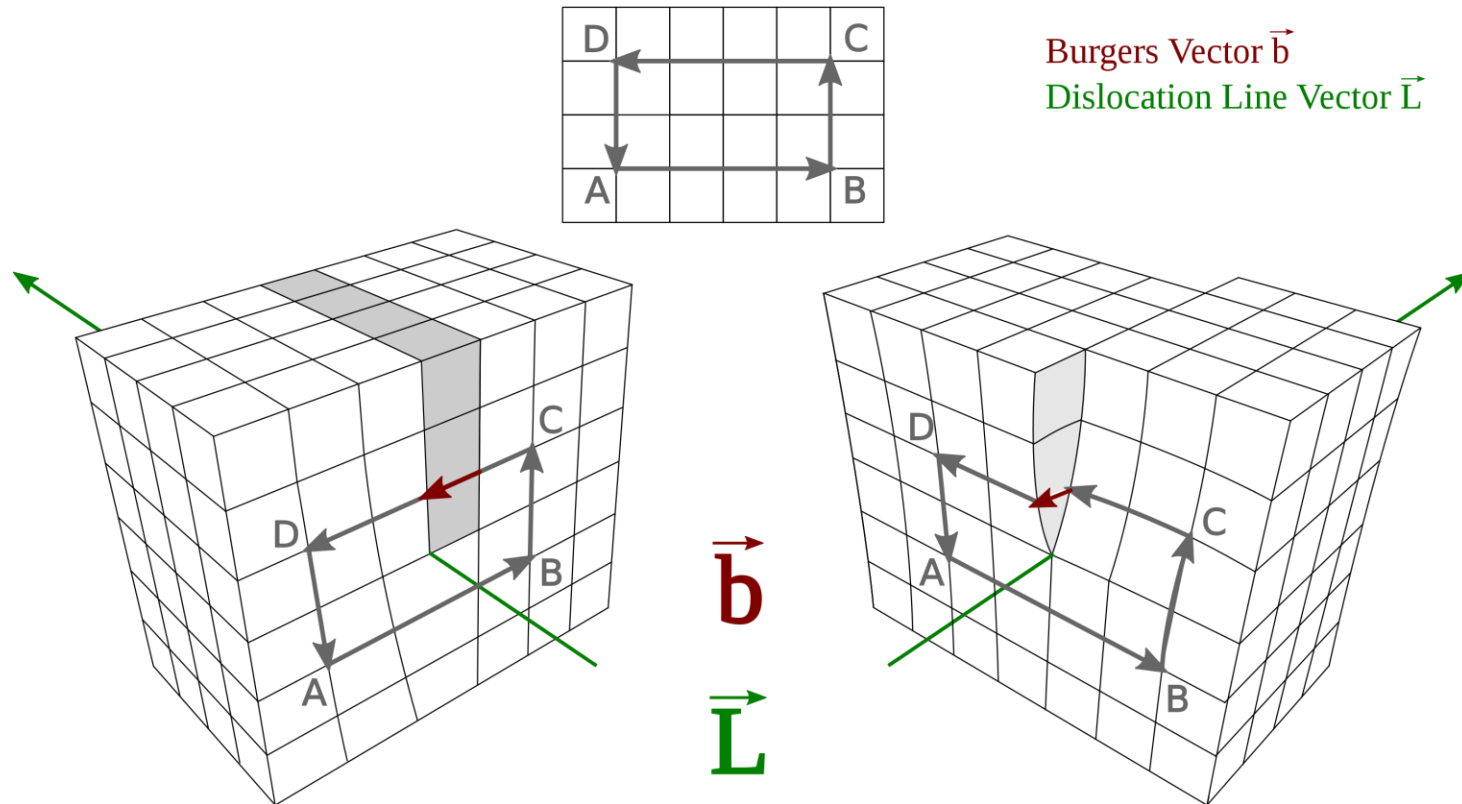


Strain to stabilize superconductivity



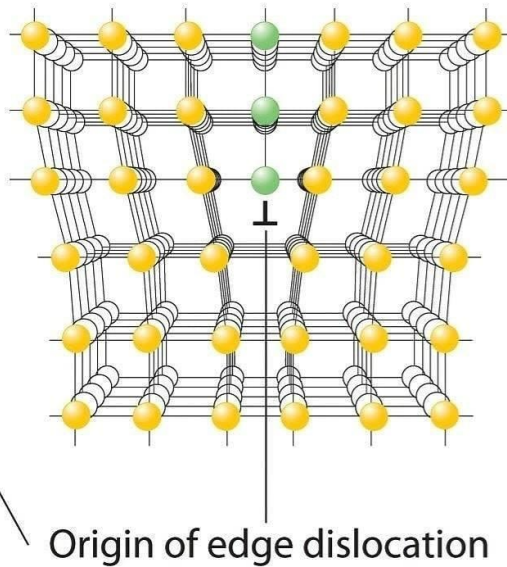
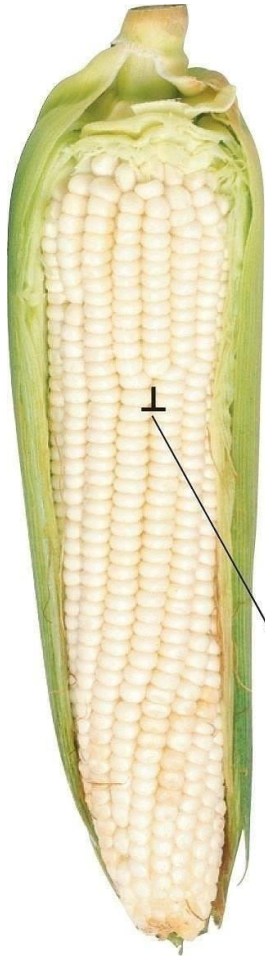
Dislocations

- Dislocations are linear defects which can be described by the Burgers vector



- Burgers vector can be perpendicular to the dislocation line (edge dislocation), parallel to the dislocation line (screw dislocation), or make an arbitrary angle with the dislocation line (mixed dislocation)

Dislocations



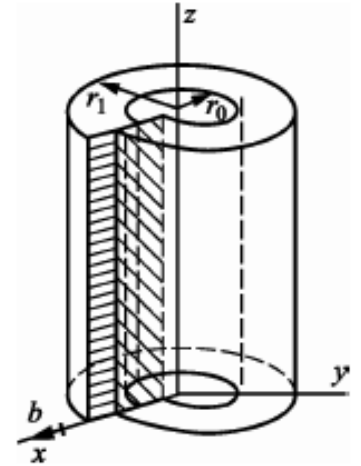
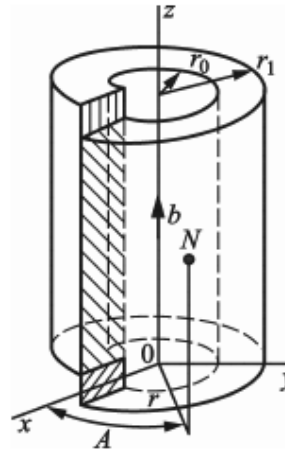
Dislocations

- By definition, the displacement field $\vec{u}(\vec{r})$ satisfies the following condition for any closed contour around the dislocation line

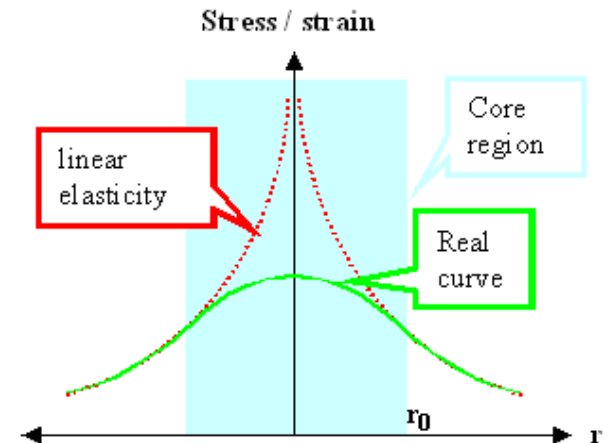
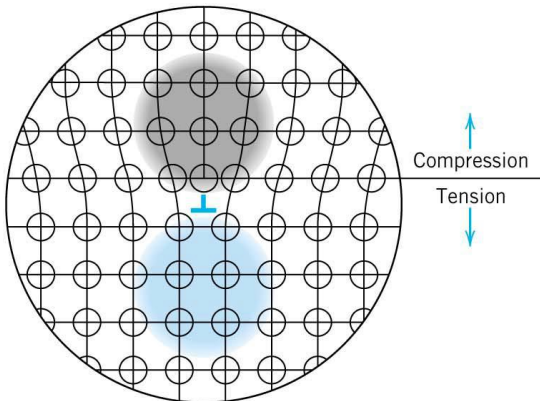
$$\oint d\vec{u}(\vec{r}) = \vec{b}$$

$$u(r) \cdot 2\pi r \sim b$$

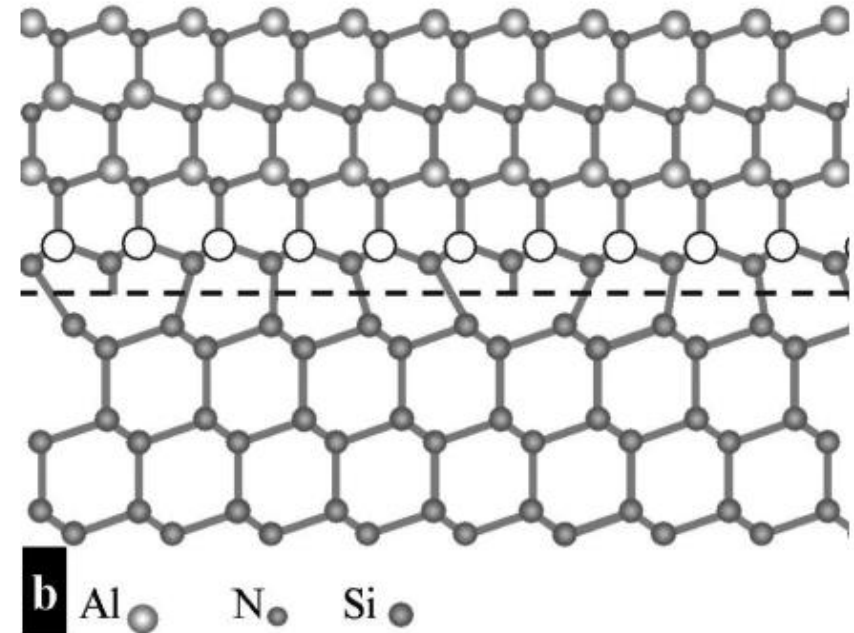
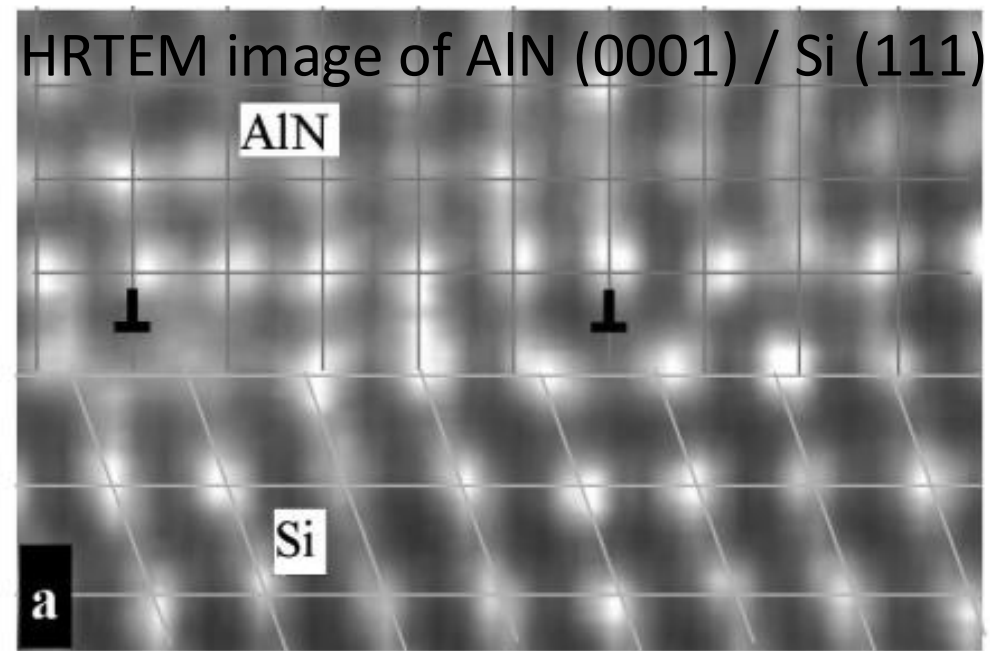
$$u(r) \sim \frac{b}{2\pi r}$$



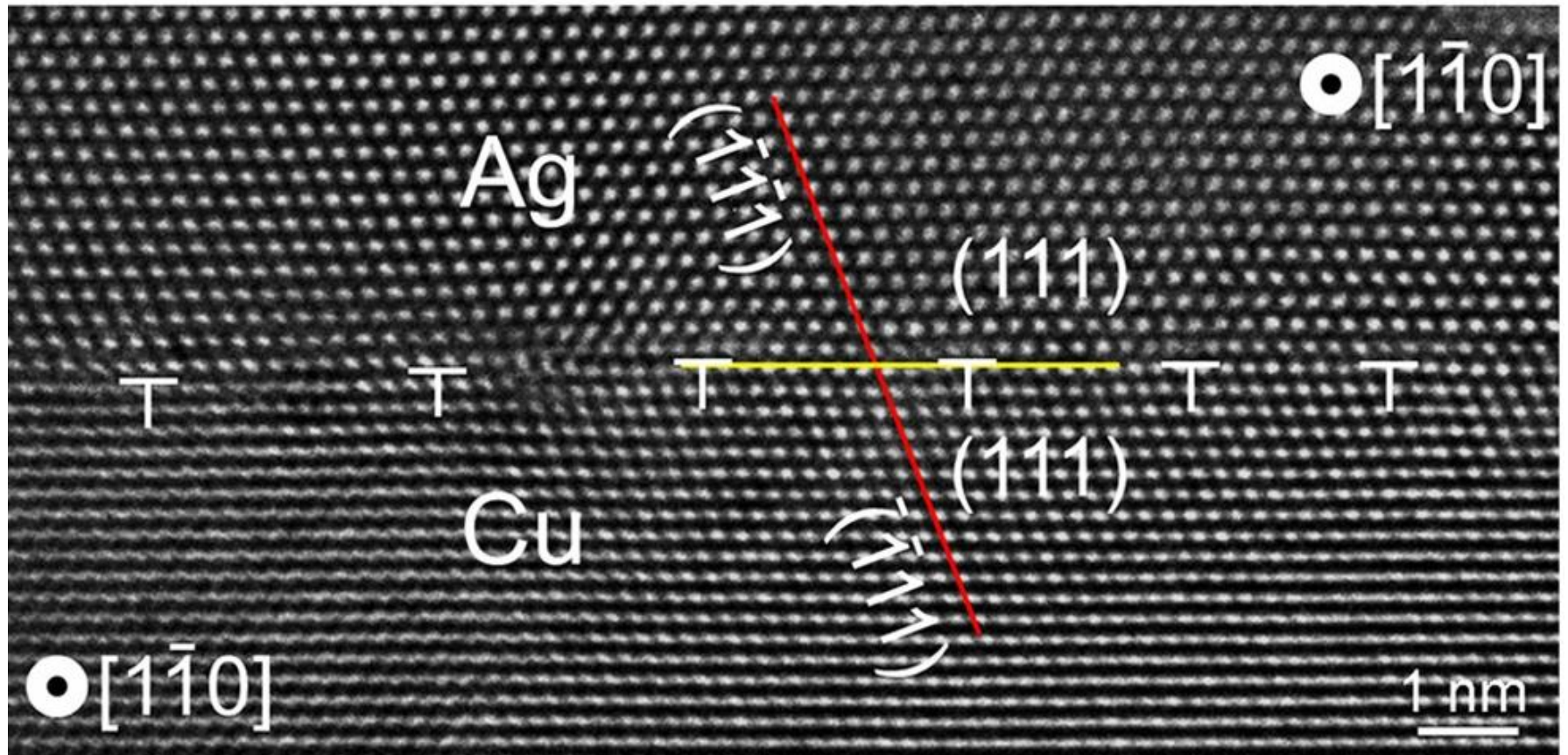
- Dislocations create slowly decaying displacement field $u \sim 1/r$, so dislocations significantly distort the crystal lattice



Misfit dislocations



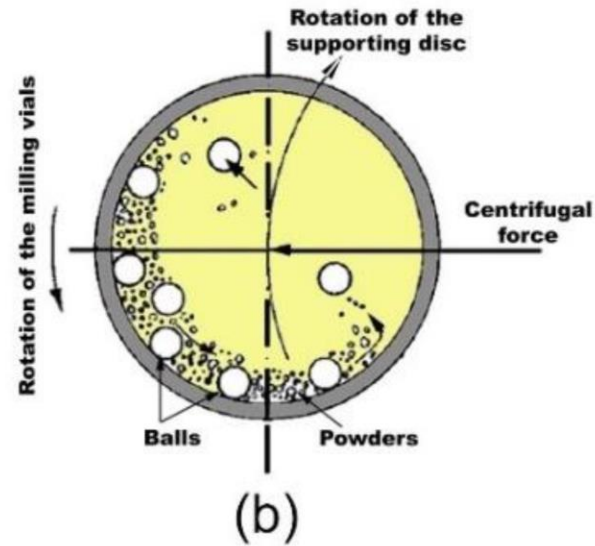
Misfit dislocations



Mechanical milling



(a)



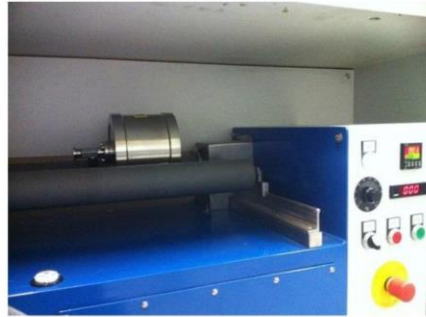
(b)

Planetary-type high energy ball mill

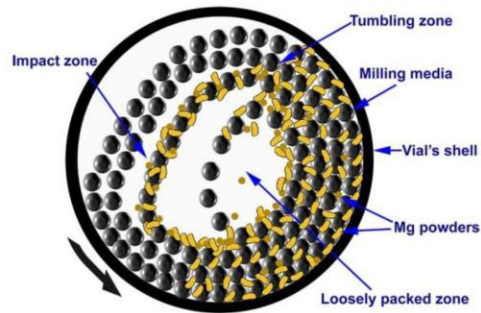
Mechanical milling



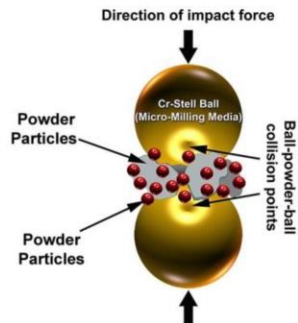
(a)



(b)



(c)

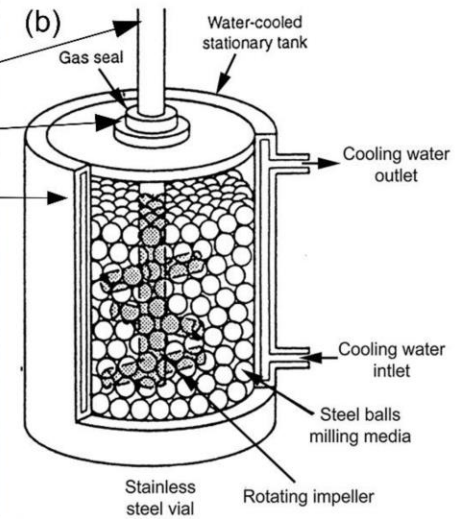


(d)

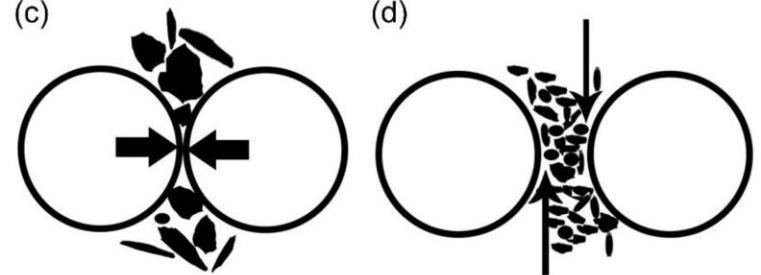
Tumbler Ball Mill



(c)

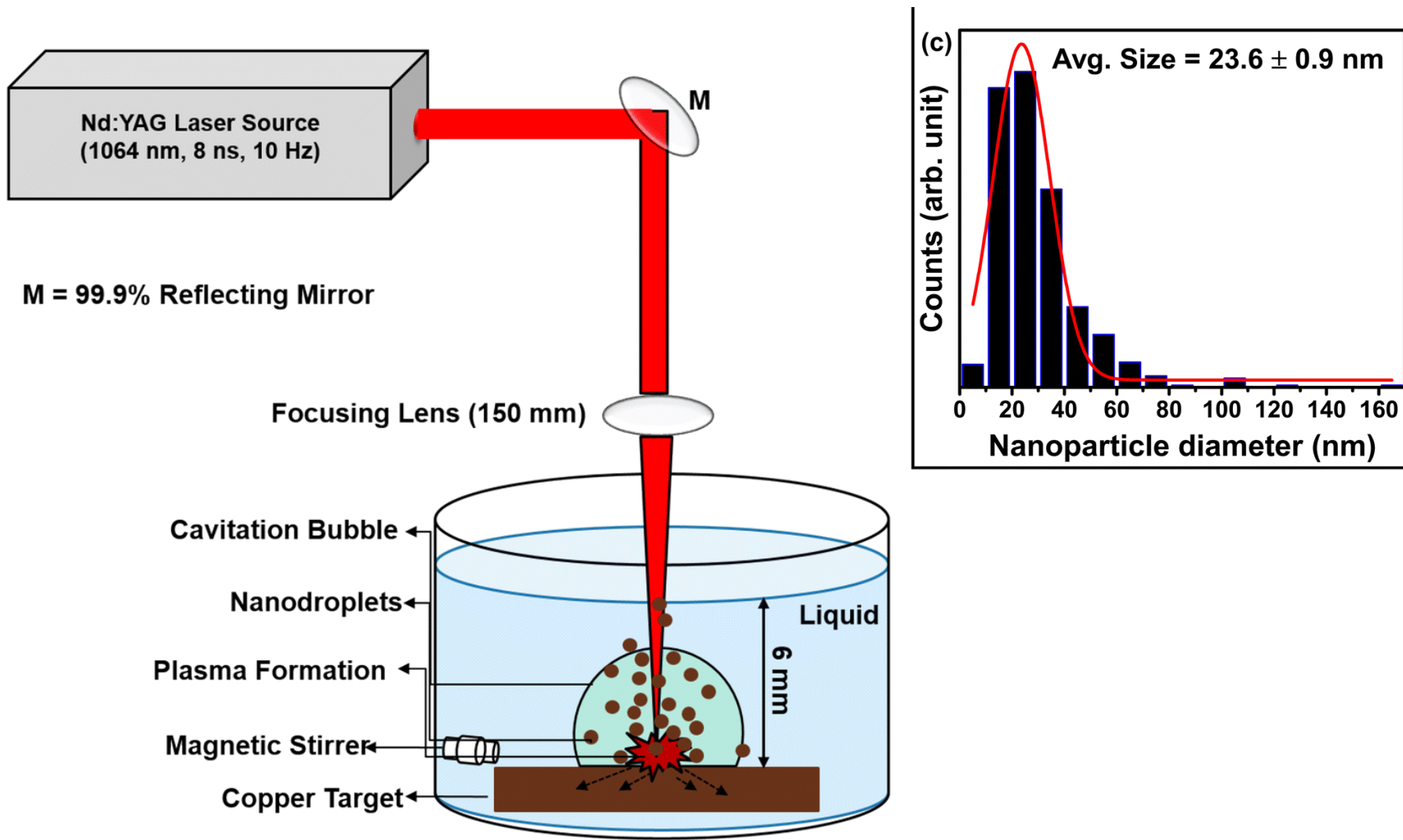


(d)



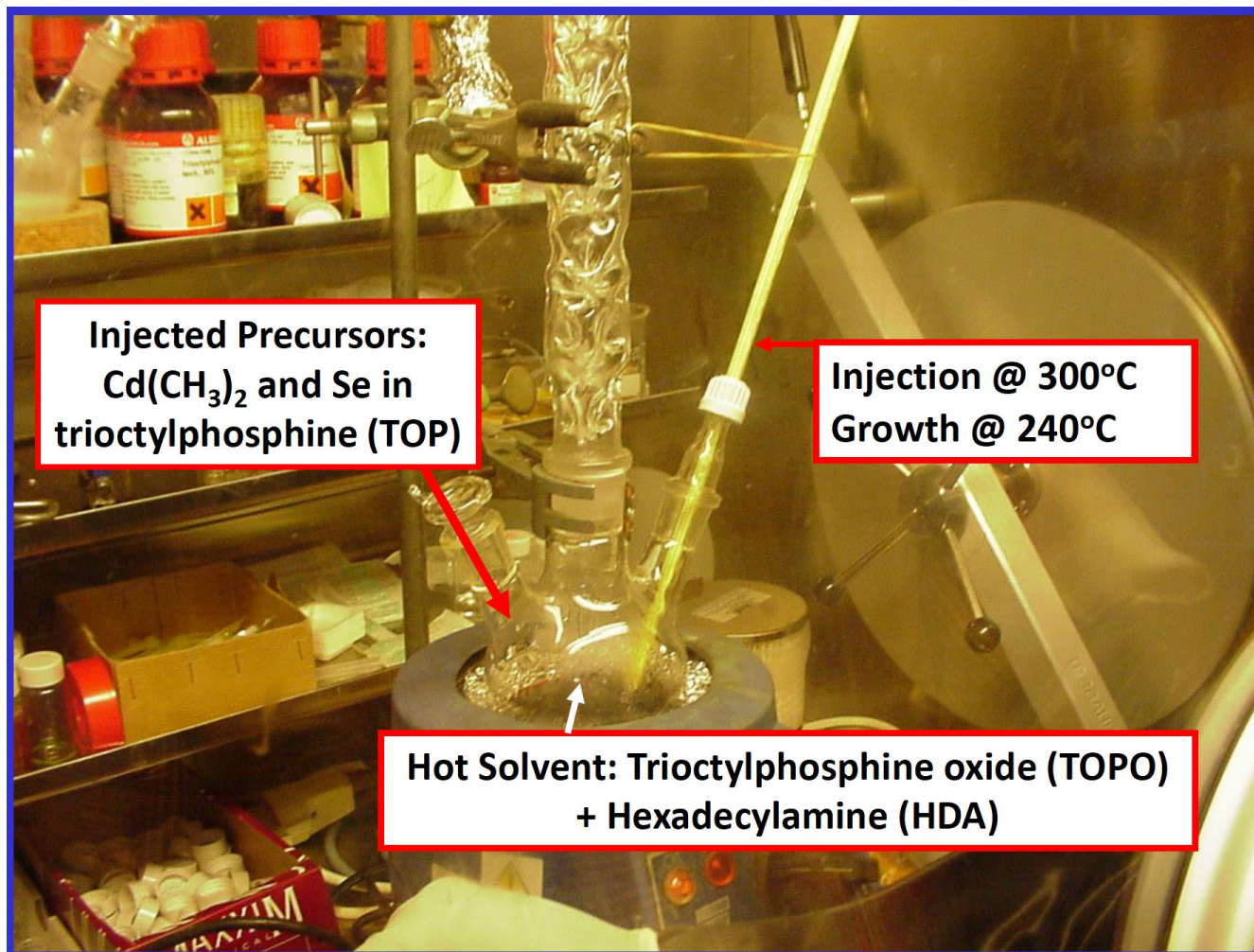
Attritor ball mill

Laser ablation



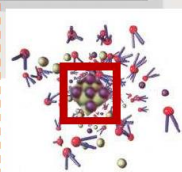
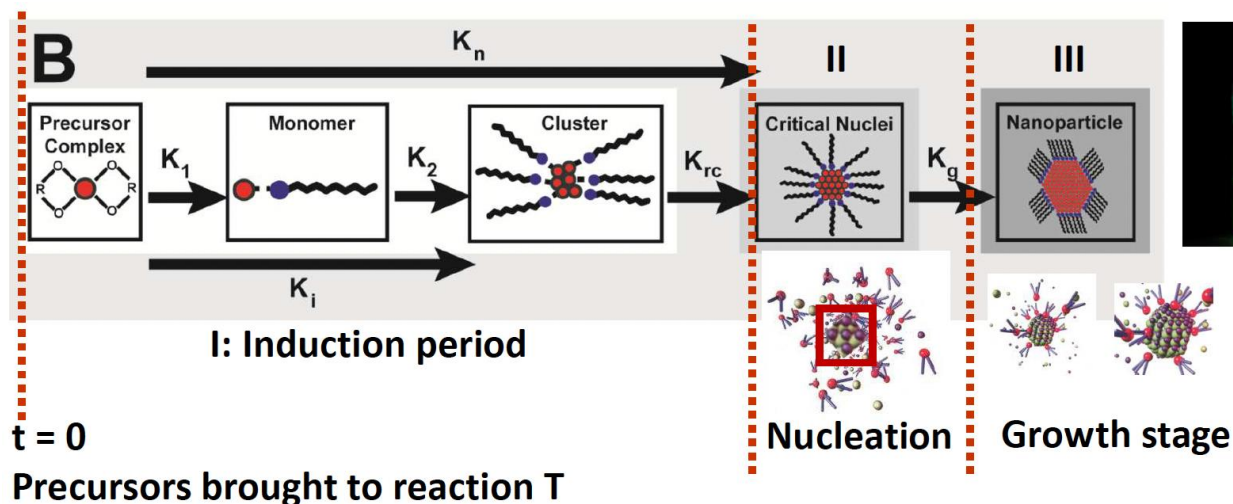
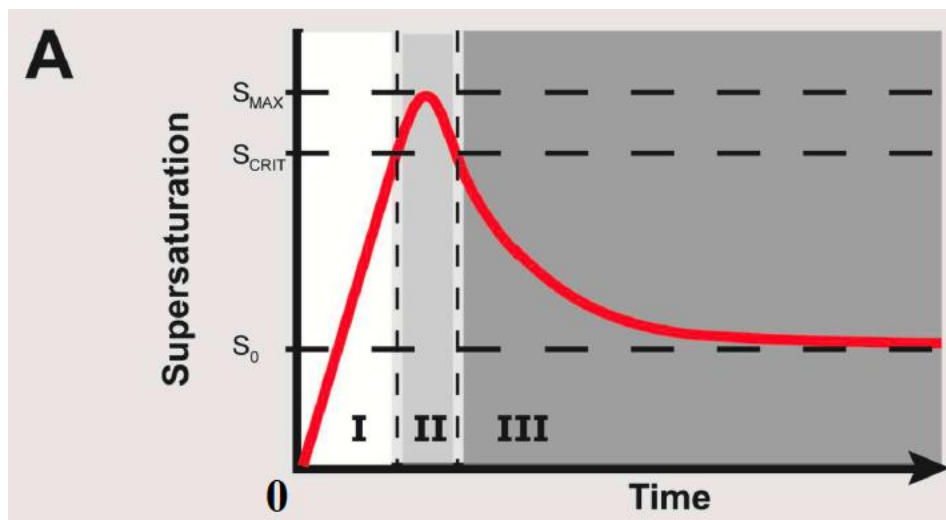
Synthesis in solution

Hot-injection Synthesis of CdSe QDs



Courtesy of Celso de Mello Donega

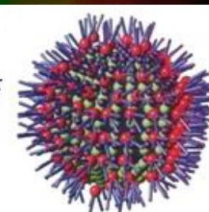
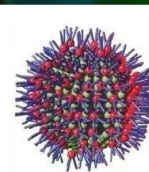
Synthesis in solution



Nucleation



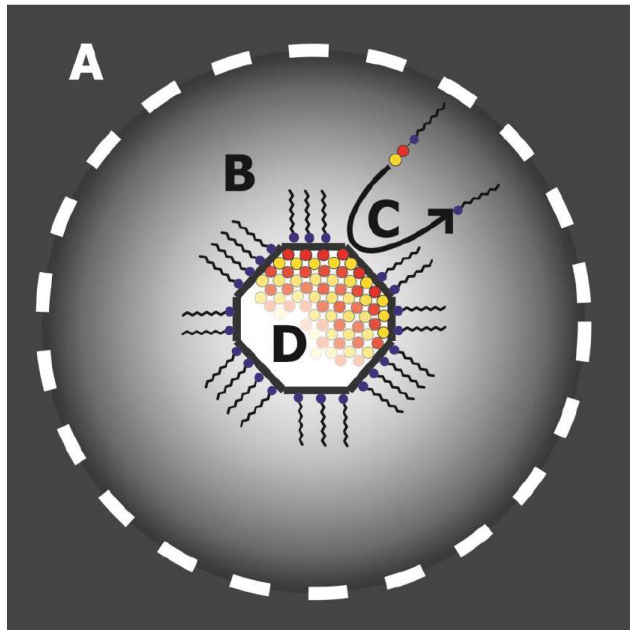
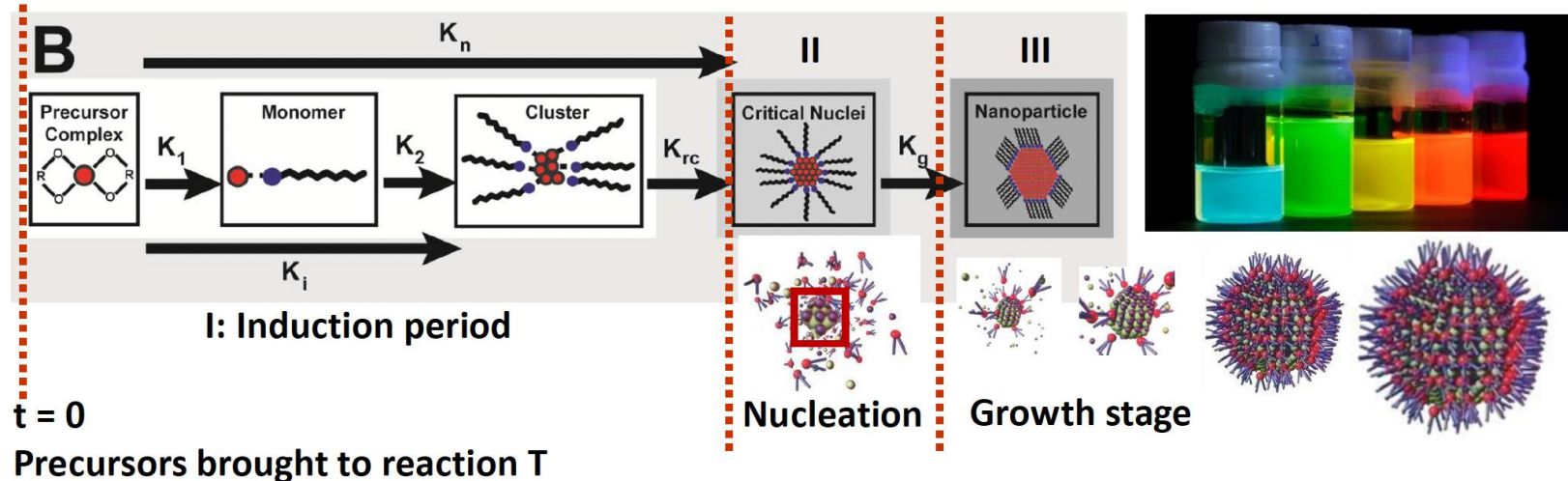
Growth stage



R. Viswanatha and D.D. Sarma in "Nanomaterials Chemistry: Recent Developments and New Directions" (Wiley-VCH, 2007)

Donega, Chem. Soc. Rev., 40 (2011) 1512

Synthesis in solution



Two kinetic growth regimes:

- 1. Diffusion-controlled growth:**
rate limiting step is the diffusion towards NC surface
- 2. Reaction-controlled growth:**
rate limiting step is the incorporation in the nanocrystal

Self-assembly of nanodots

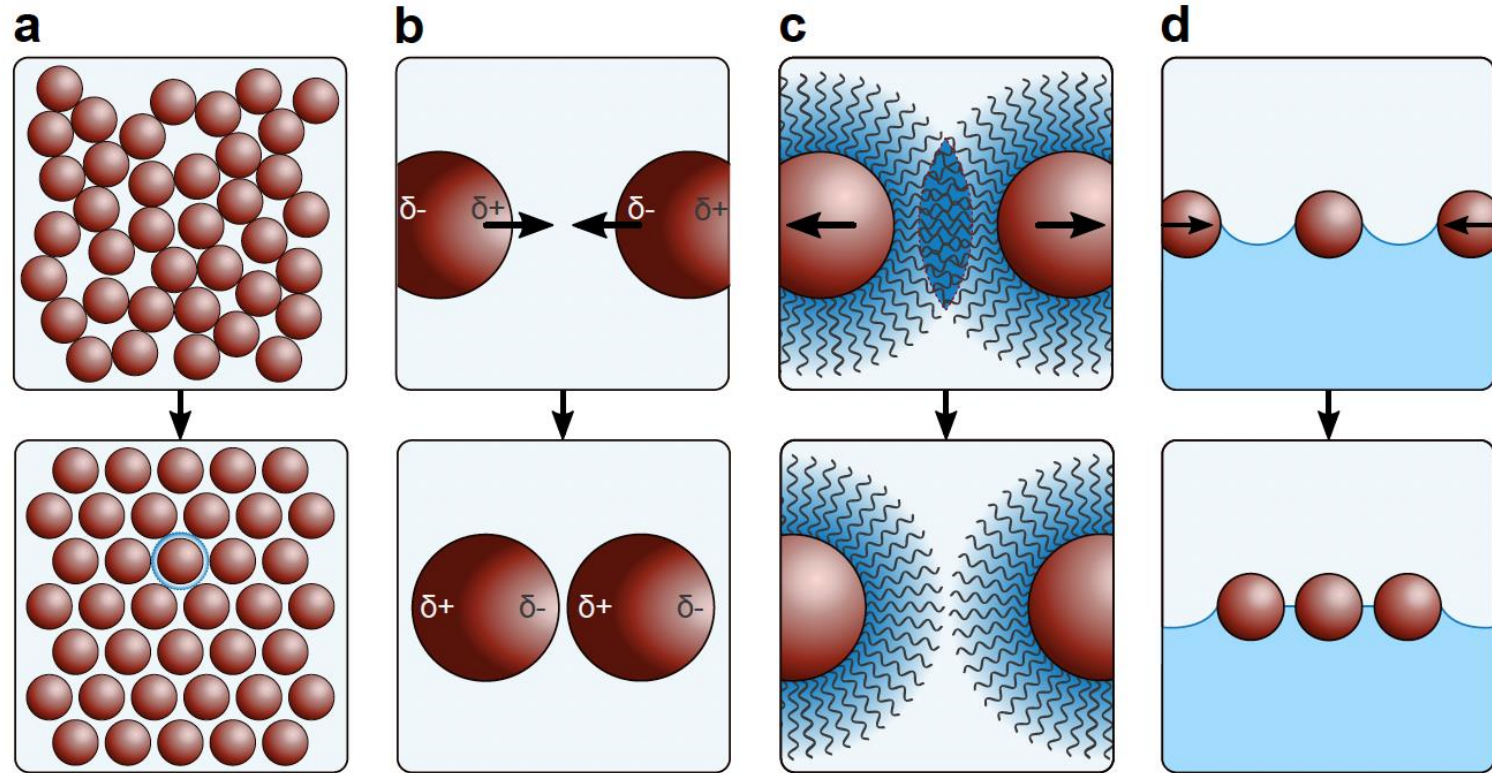


Figure 2.4: Schematic illustration of different interparticle interactions. (a) Entropic forces: For hard spheres the entropy is increased ($\Delta S > 0$) by assembling into crystalline state, caused by a gain in accessible free volume, indicated for one sphere in blue. (b) Van der Waals attraction: Charge density fluctuations create induced dipoles (δ^+ and δ^-), resulting in an attractive force of NCR cores. (c) Steric repulsion: Overlapping ligand shells of adjacent NCRs result in a repulsive force, preventing NCR agglomeration. (d) Capillary forces: Adsorption of NCRs at an interface cause distortion, resulting in an induced attractive force.

Self-assembly of nanodots

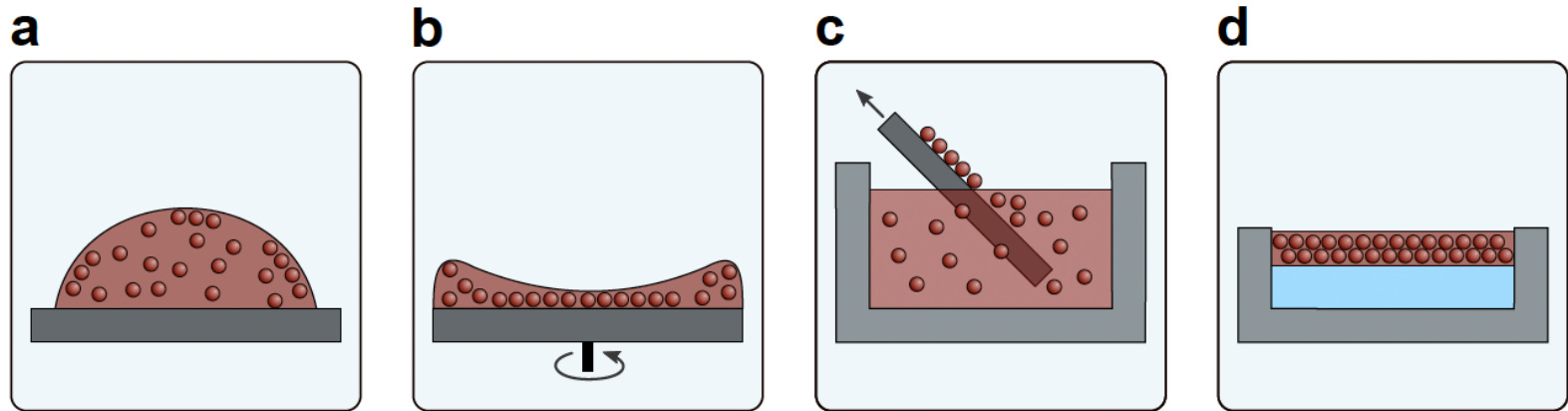
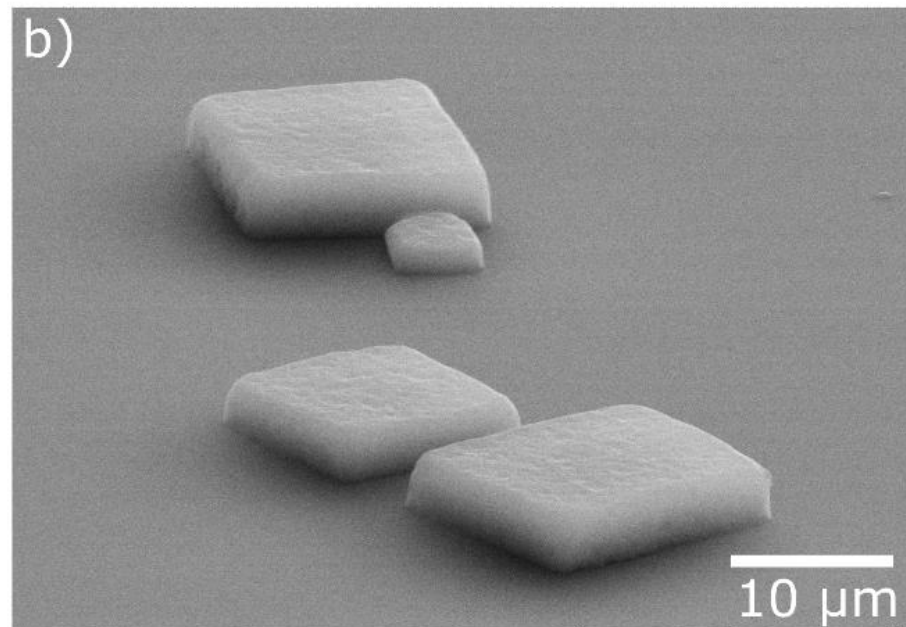
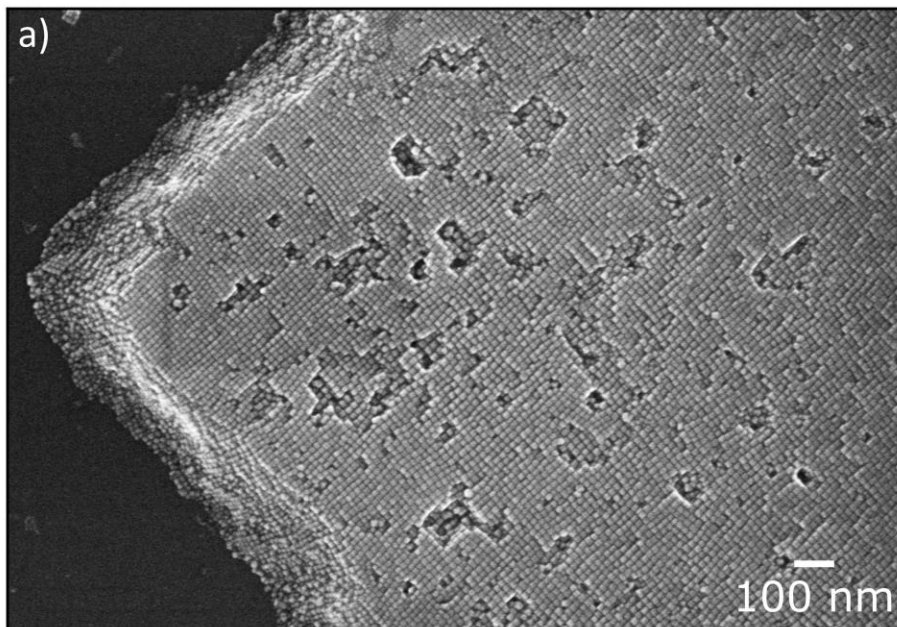
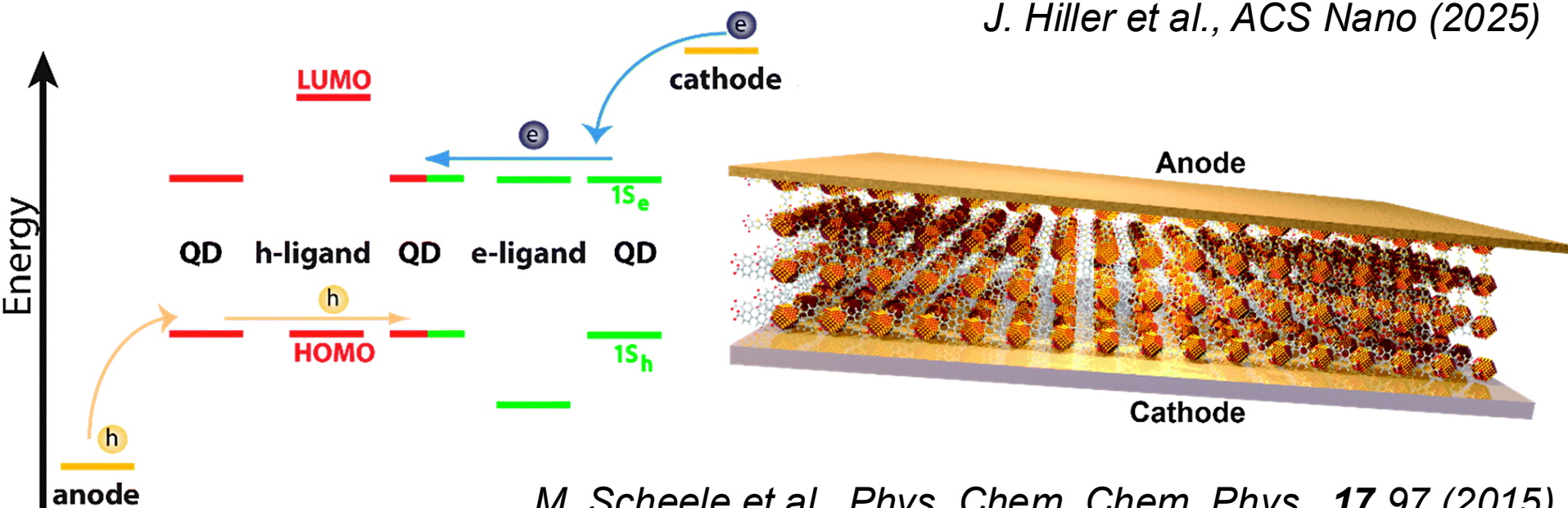


Figure 2.6: Schematic illustration of evaporation-based self-assembly methods: (a) drop-casting, (b) spin-coating, (c) dip-coating, (d) self-assembly at the liquid/air interface.

Self-assembly of nanodots

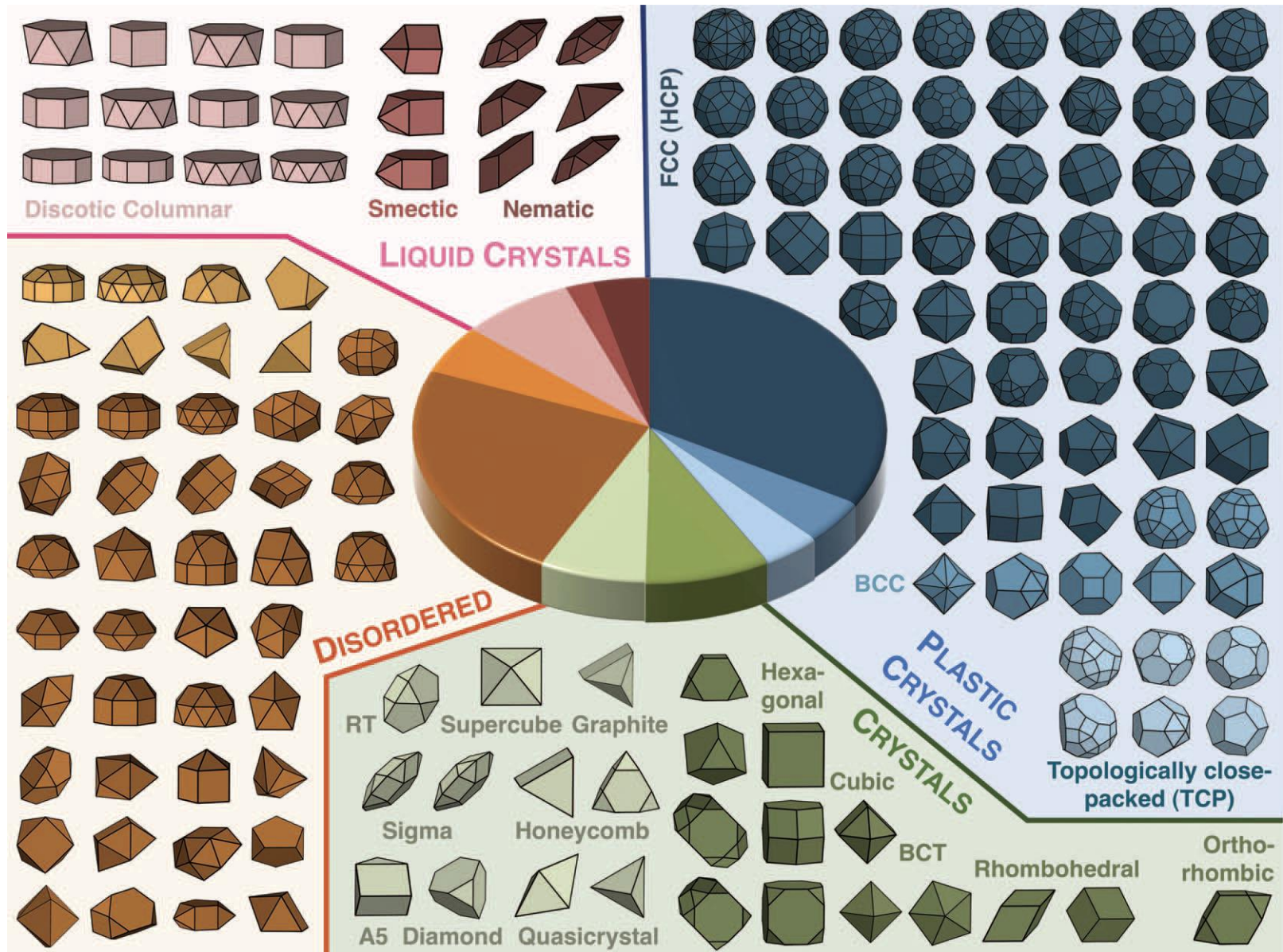


J. Hiller et al., ACS Nano (2025)

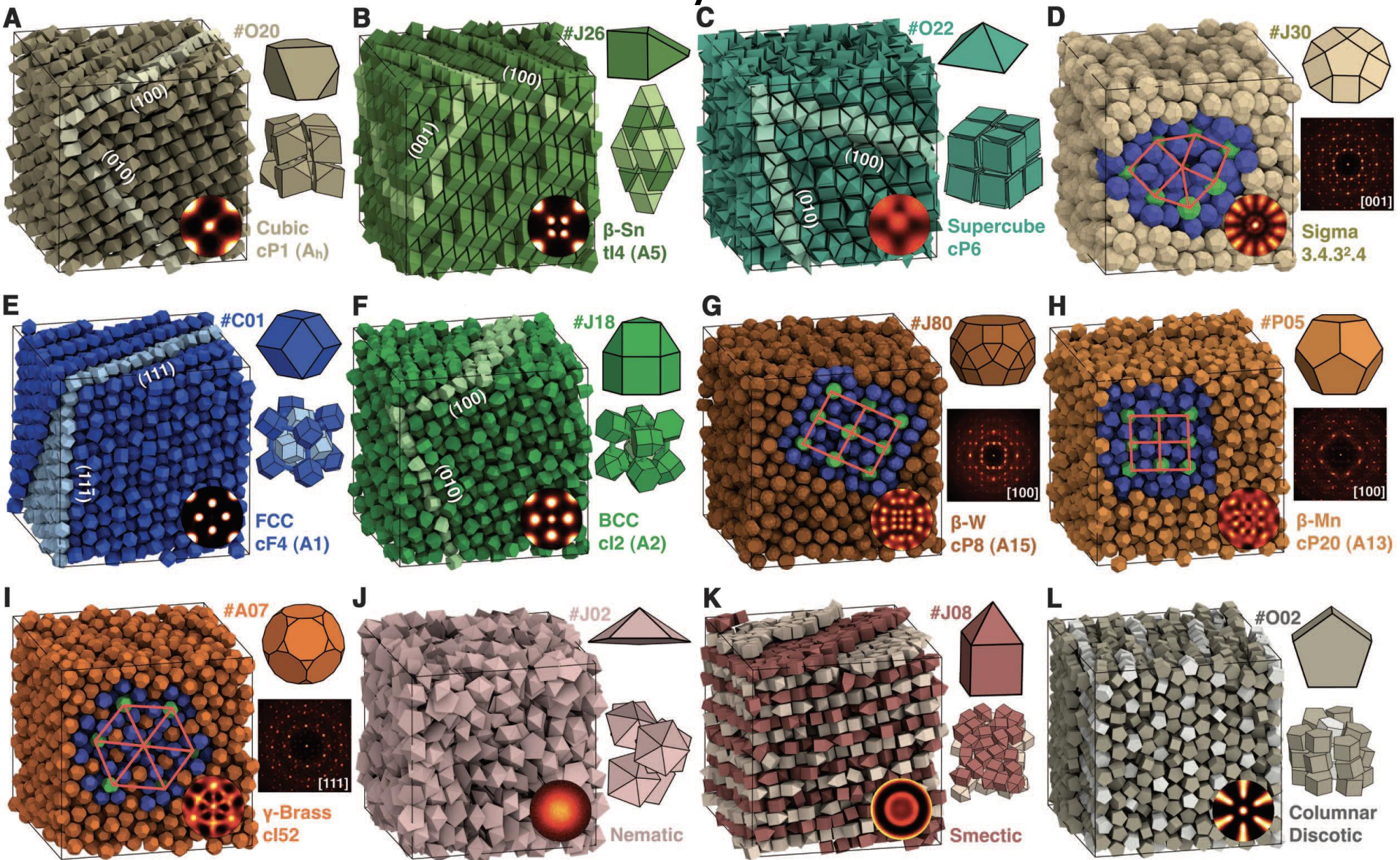


M. Scheele et al., Phys. Chem. Chem. Phys., 17 97 (2015)

Self-assembly of nanodots



Self-assembly of nanodots



Liquid phase epitaxy

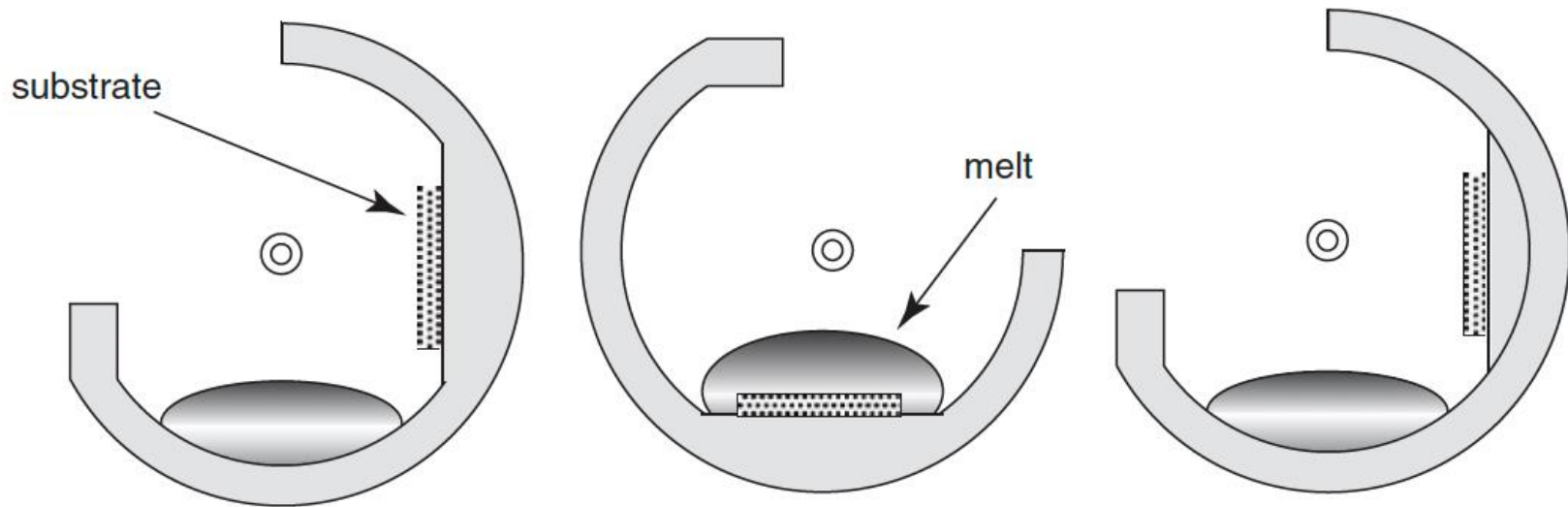
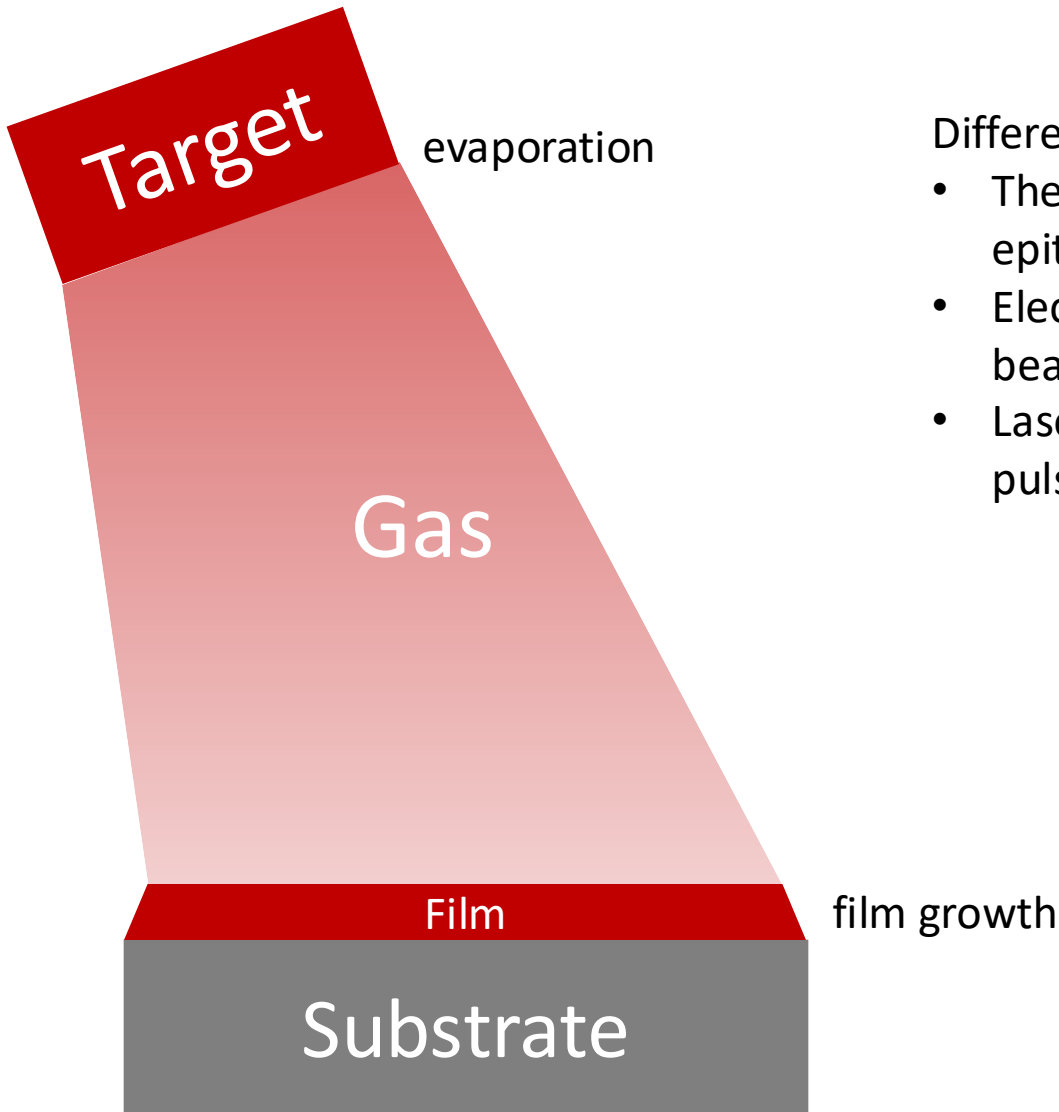


Figure 4.6 Rotating/tipping LPE apparatus. (After Arch *et al.*, 1992)

One of the schemes for liquid phase epitaxy (many other are possible)

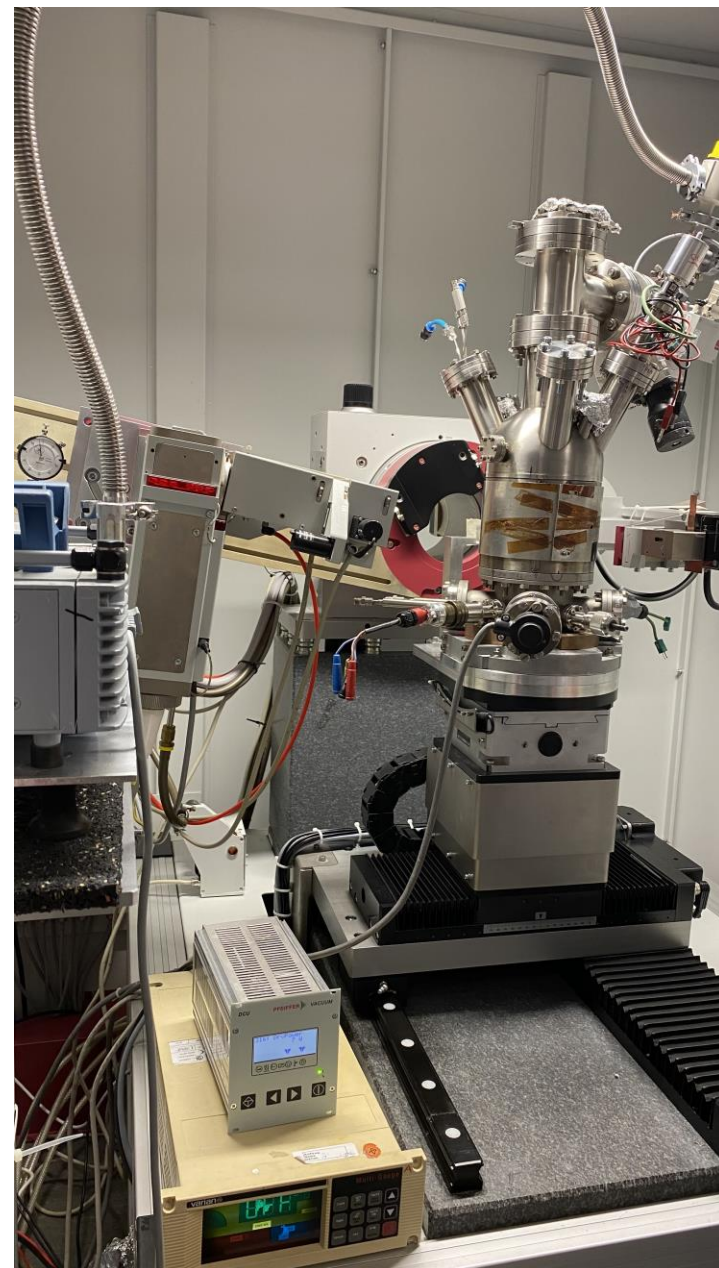
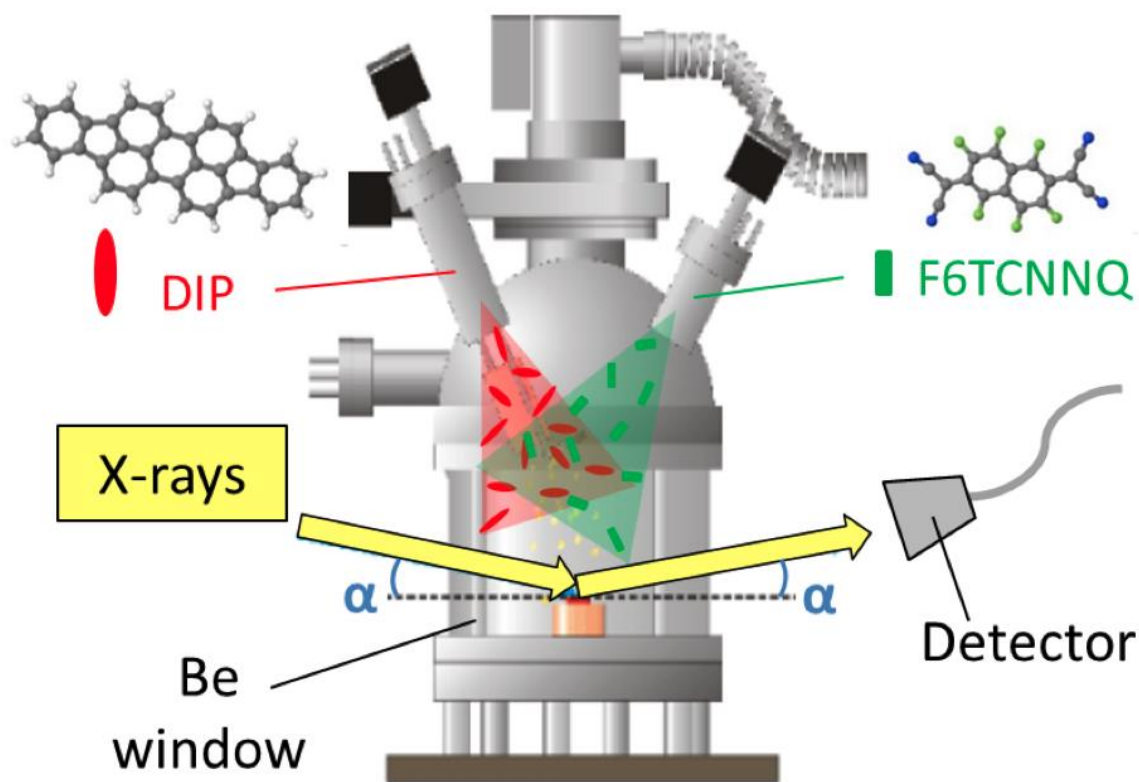
Physical vapor deposition (PVD)



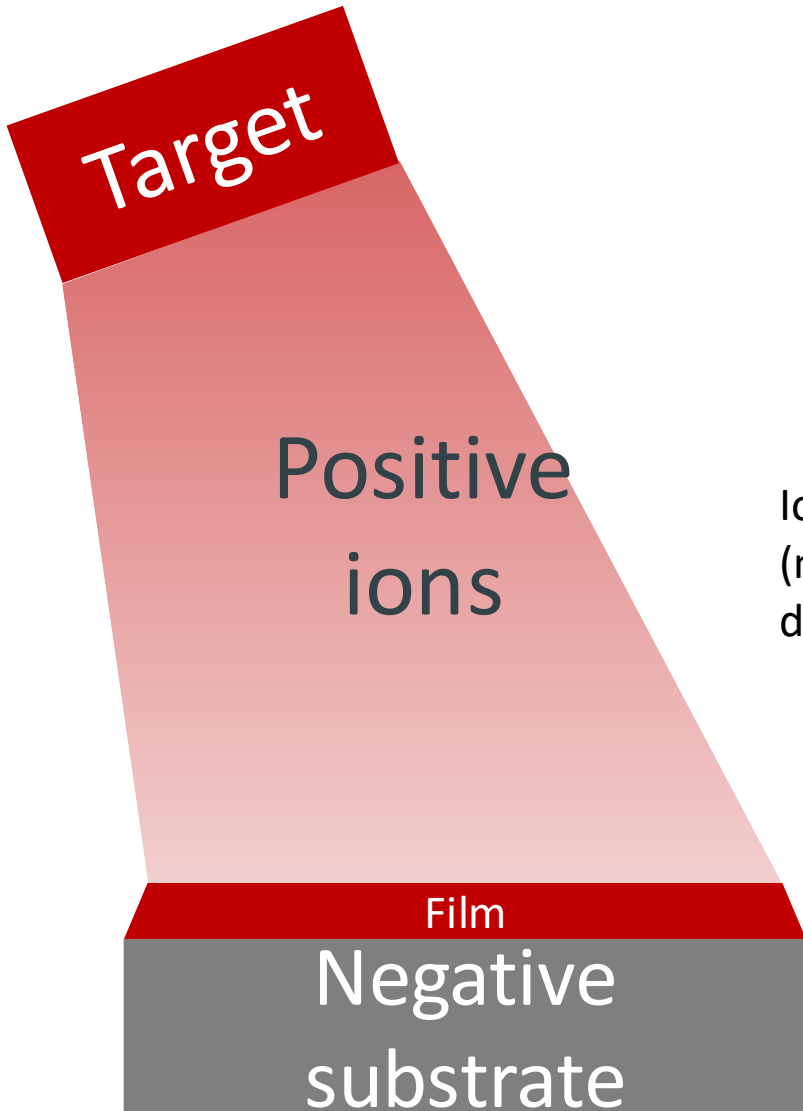
Different methods to create gas phase:

- Thermal evaporation (molecular beam epitaxy)
- Electron beam evaporation (electron beam epitaxy)
- Laser evaporation (thermal laser epitaxy, pulsed laser deposition)

Organic molecular beam deposition



Deposition from plasma

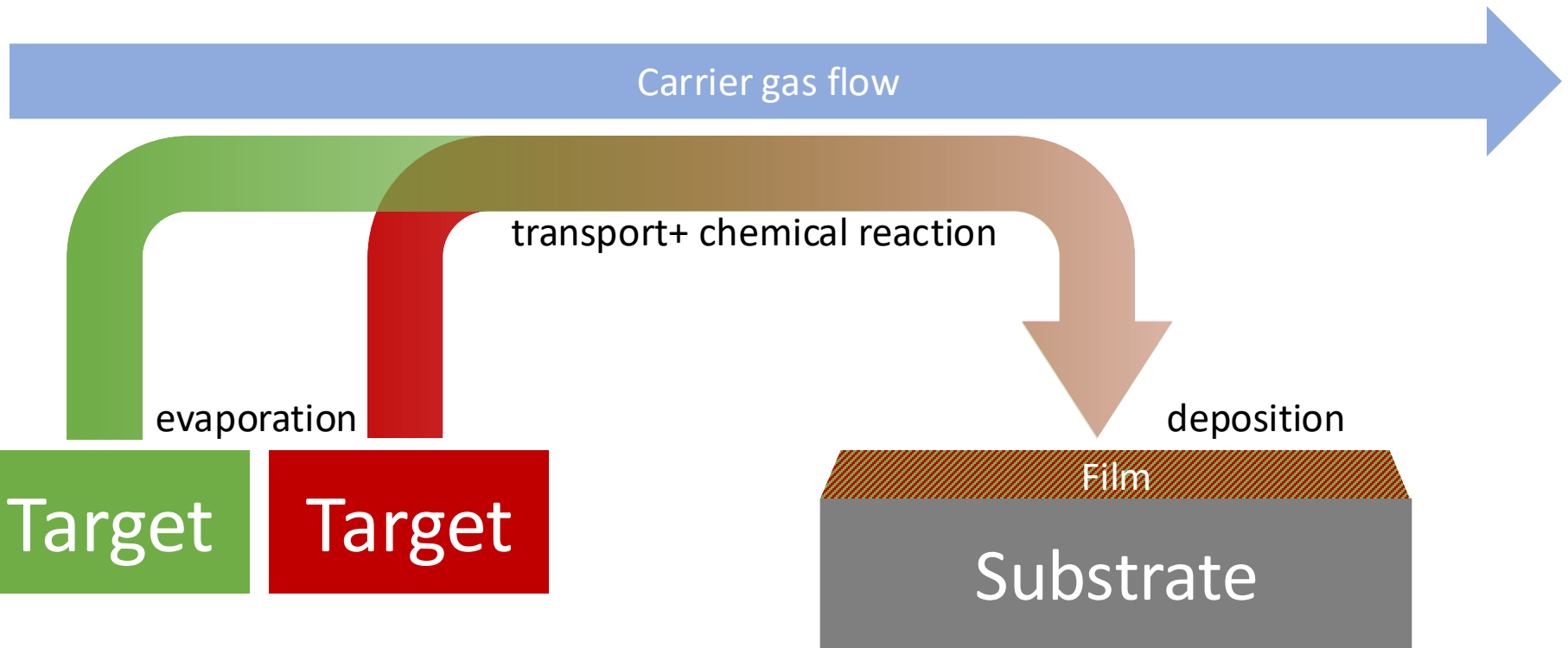


Different methods to create plasma

- Sputtering
- Cathode arc
- Laser pulses

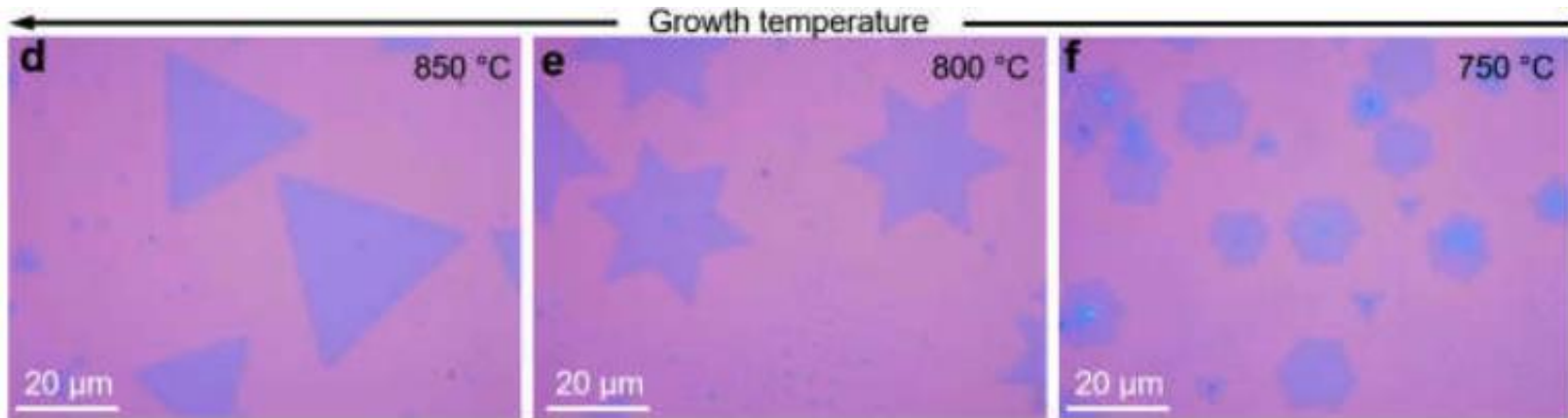
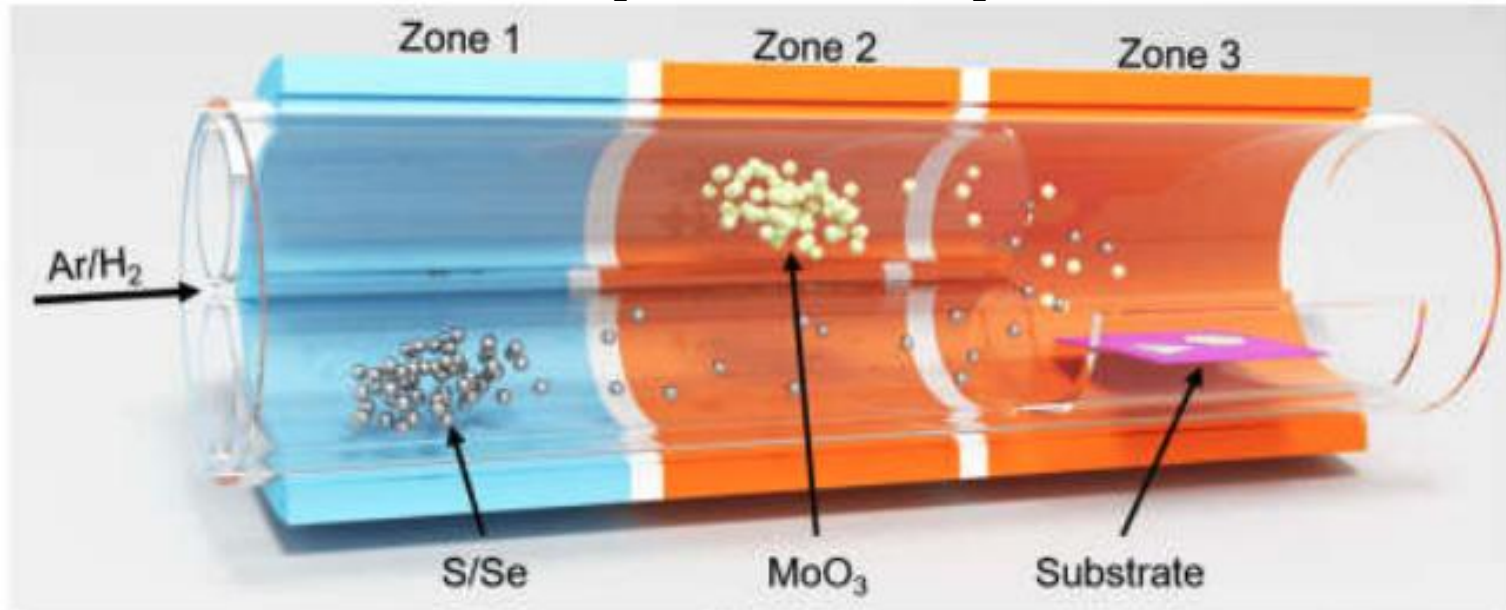
Ions can be guided by magnetic field (magnetron sputtering) or by gas flow (vapor deposition)

Chemical vapor deposition (CVD)

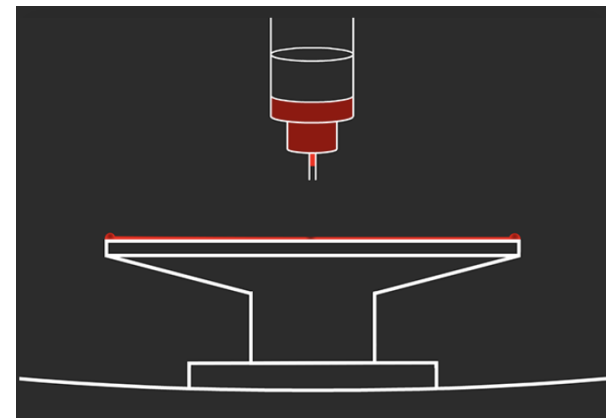
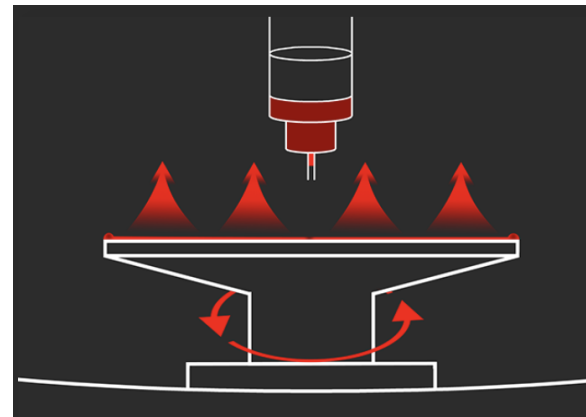
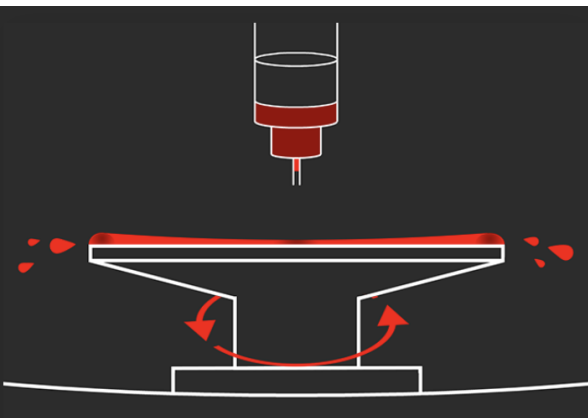
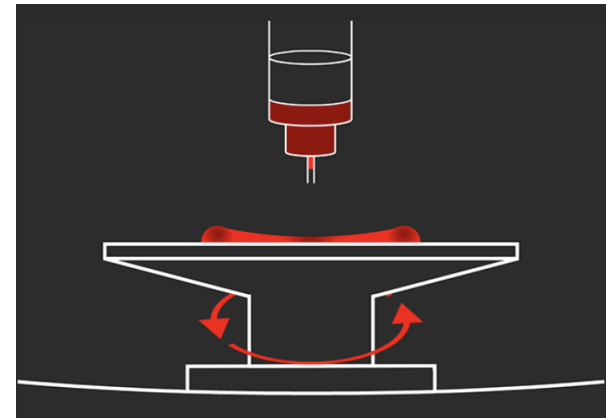
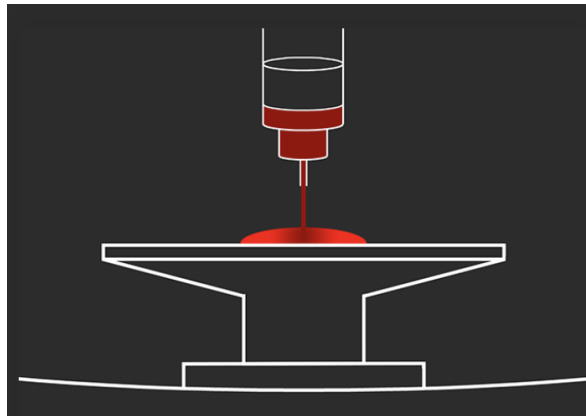
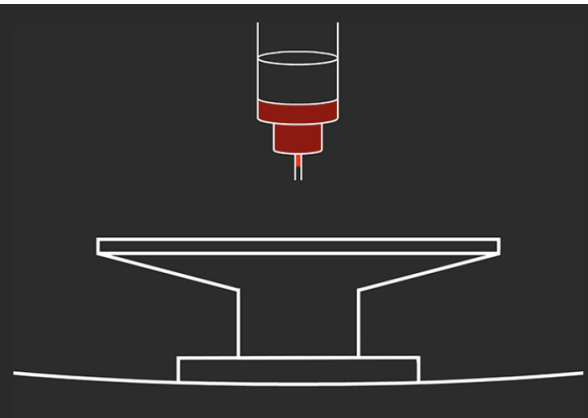


Chemical vapor deposition (CVD)

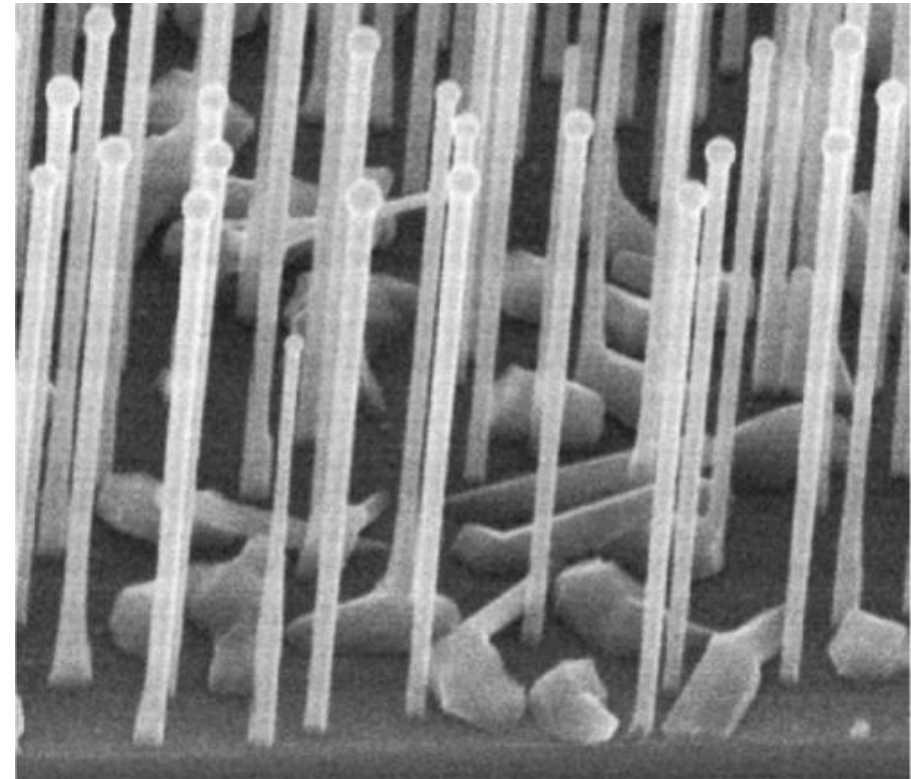
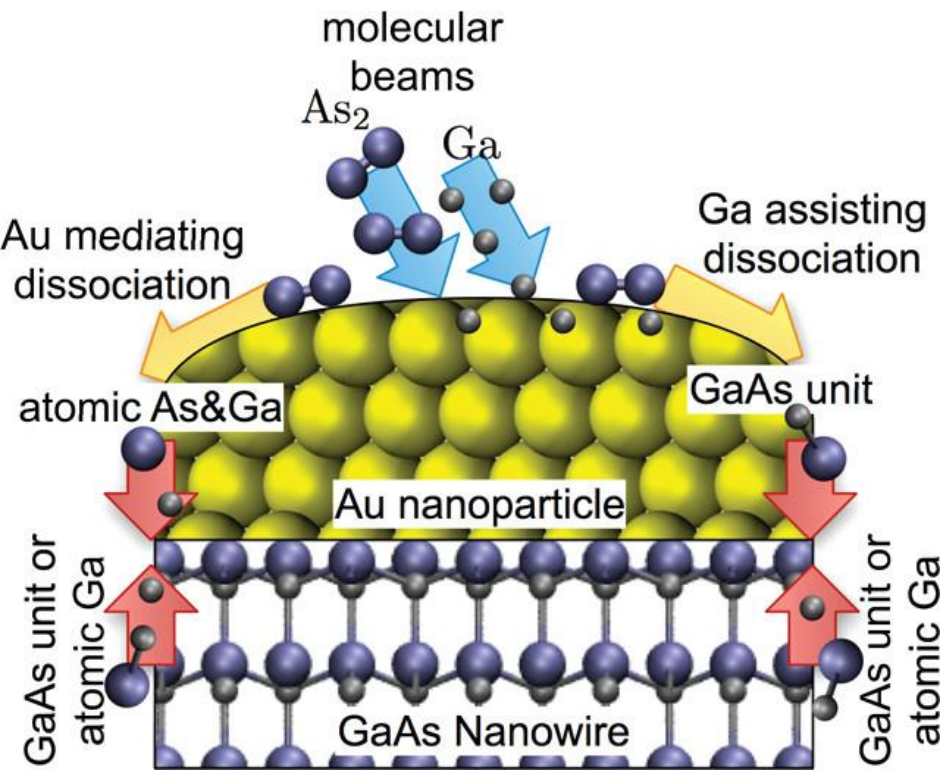
Growth of MoS_2 nanolayer on SiO_2 substrate



Spin coating

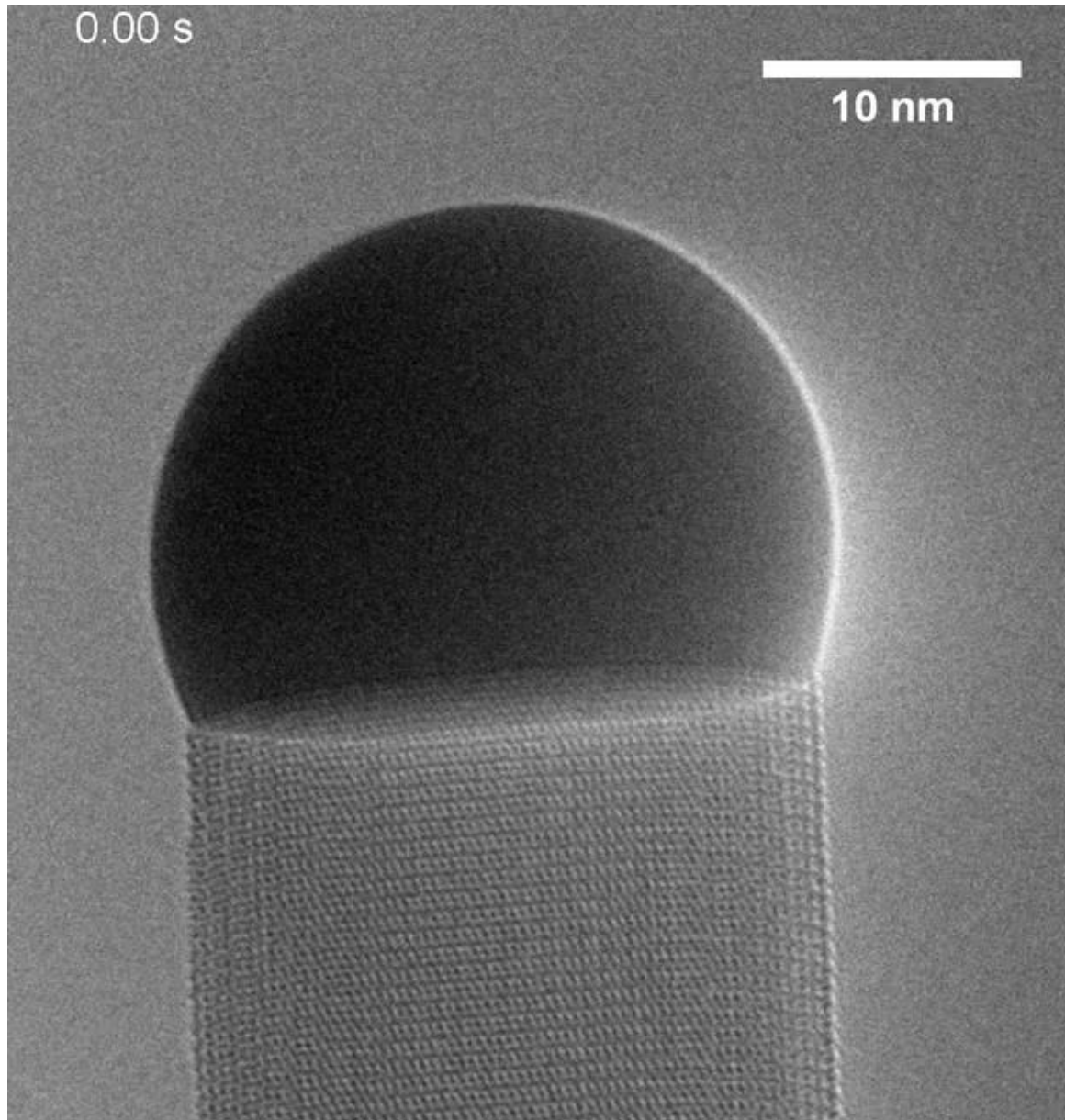


GaAs Nanowires



— 2 μm —

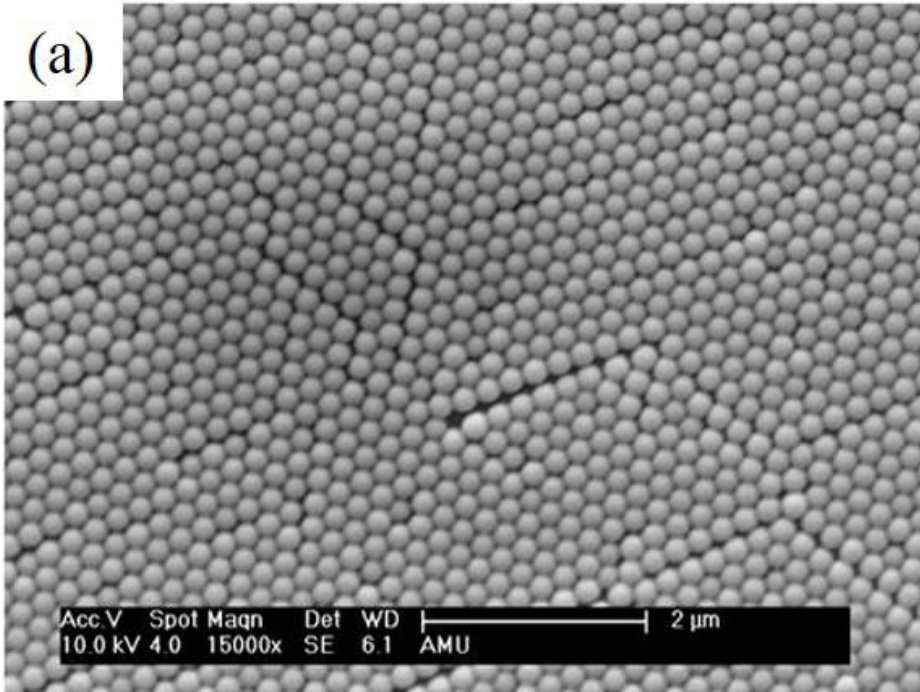
GaAs Nanowires



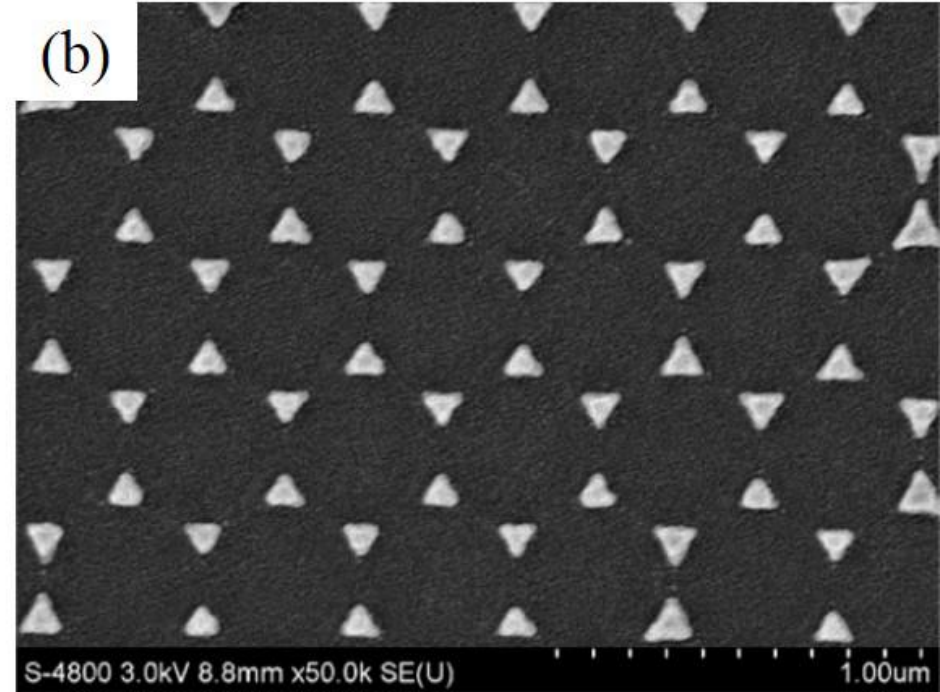
*J.-C. Harmand et al., Phys. Rev. Lett. **121** 166101 (2018)*

Nanosphere lithography

(a)



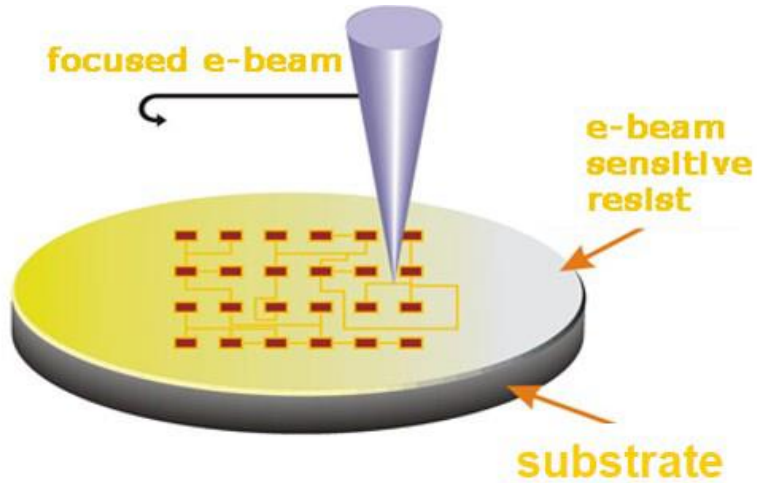
(b)



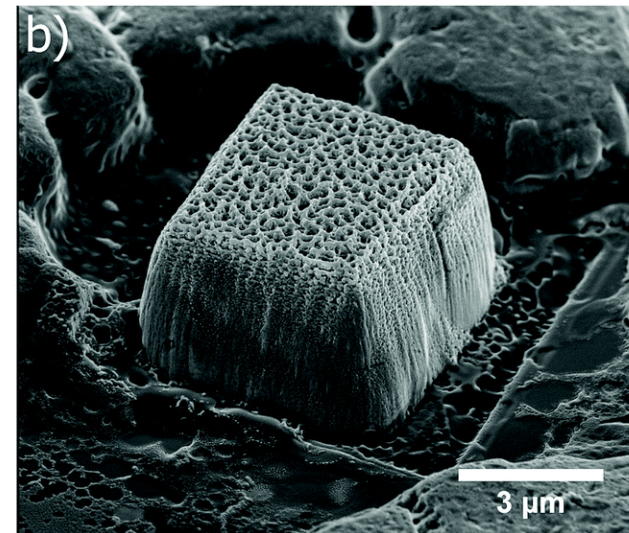
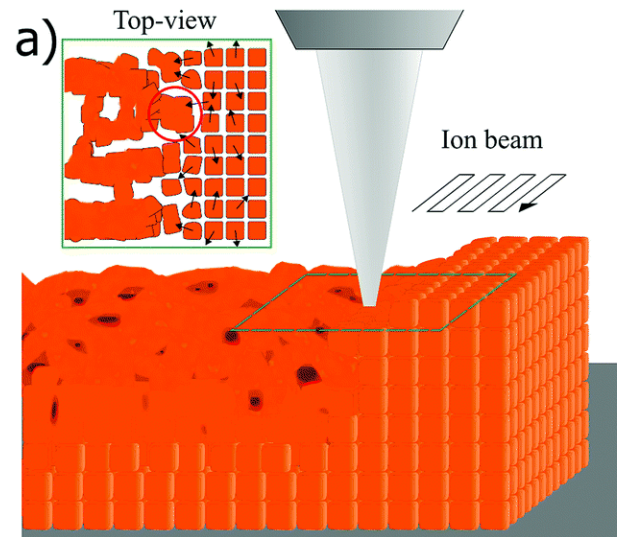
Courtesy of M. Fleischer

Nanopatterning

electron beam lithography



Focused Ion Beam (FIB) milling



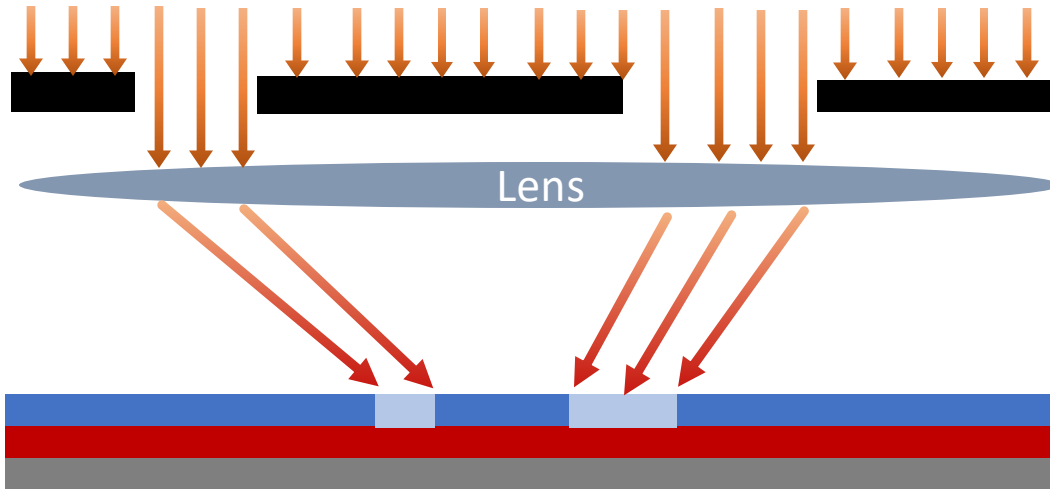
Lithography



Substrate with a thin film



Coating with a (positive) resist via spin coating, dip coating, spray coating, etc.

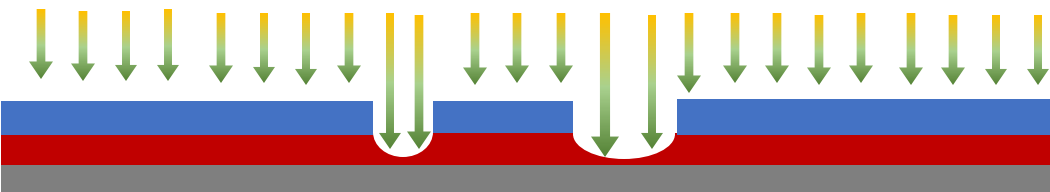


Creating modified resist structure by

- Exposing resist **through mask** to light, electrons, ions, etc.
- Using focused ion / electron beam
- Pressing resist with a stamp



Removal of the modified resist structure (developing)



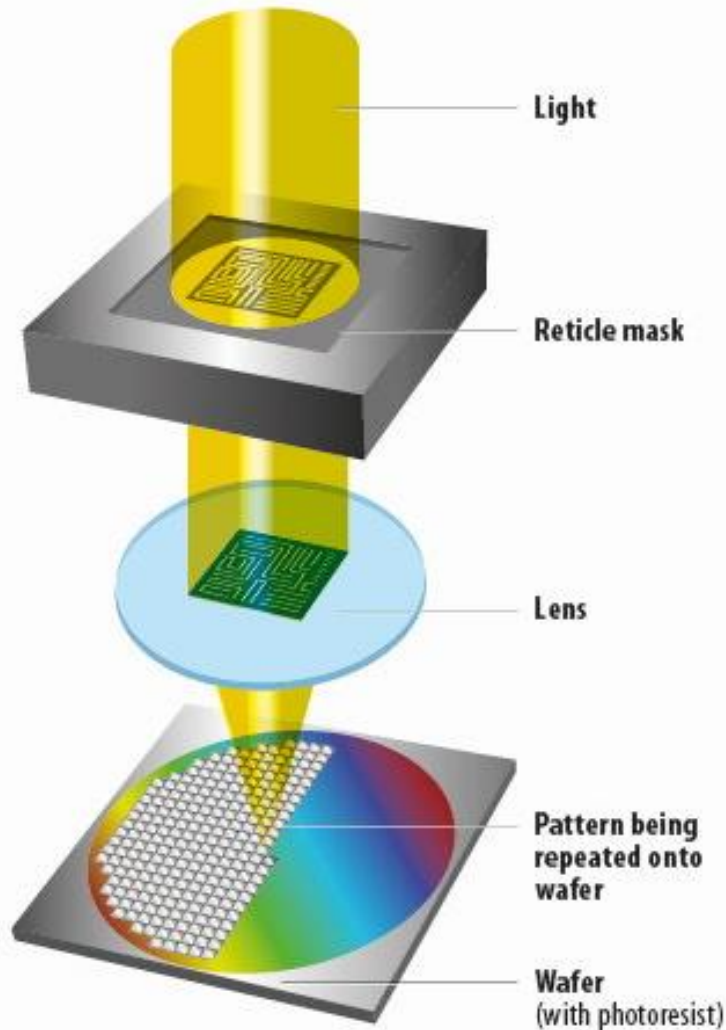
Etching or doping of the film not protected by the resist (chemical etching, dry etching, etc.)



Removal of remaining unmodified resist

Extreme Ultra Violet Photolithography

Rayleigh criterion equation (resolution): $\Delta = k_1 \cdot \frac{\lambda}{NA}$



k_1 -factor, depends on the imaging and resist system (physical limit 0.25)

$NA = n \cdot \sin \theta$ - numerical aperture of the lens (by using ~1m large optical systems, $NA=1.35$)

λ - wavelength of light
(currently $\lambda = 13.5 \text{ nm}$)

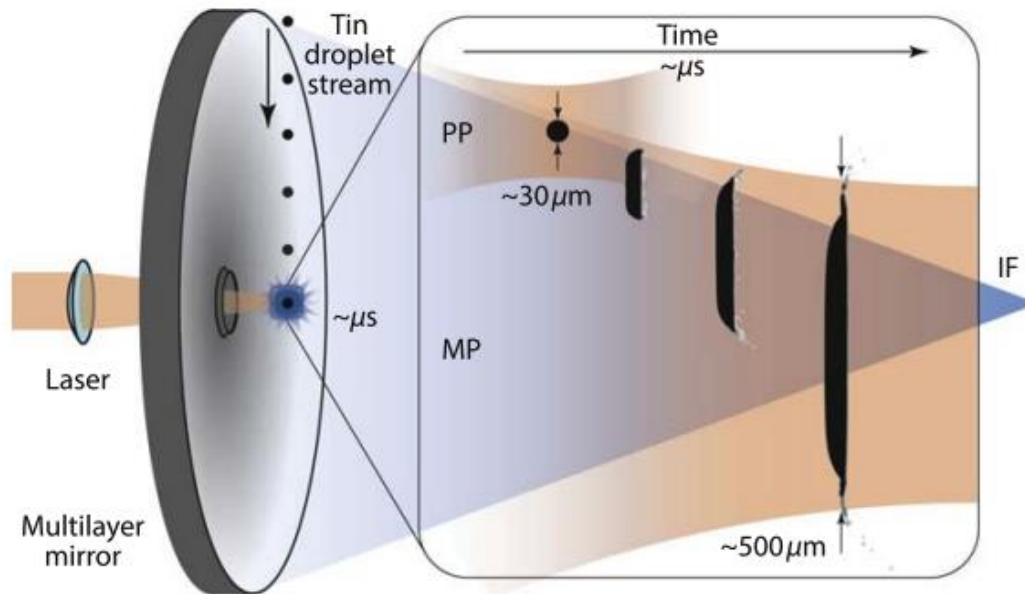
Generating EUV

Source of EUV light :

$4p^6 4d^m - 4p5 4d^{m+1} + 4d^{m-1} 4f$ in electron transitions in $\text{Sn}^{8+}-\text{Sn}^{14+}$

Prepulse forms flat disk of tin

Main pulse creates tin plasma



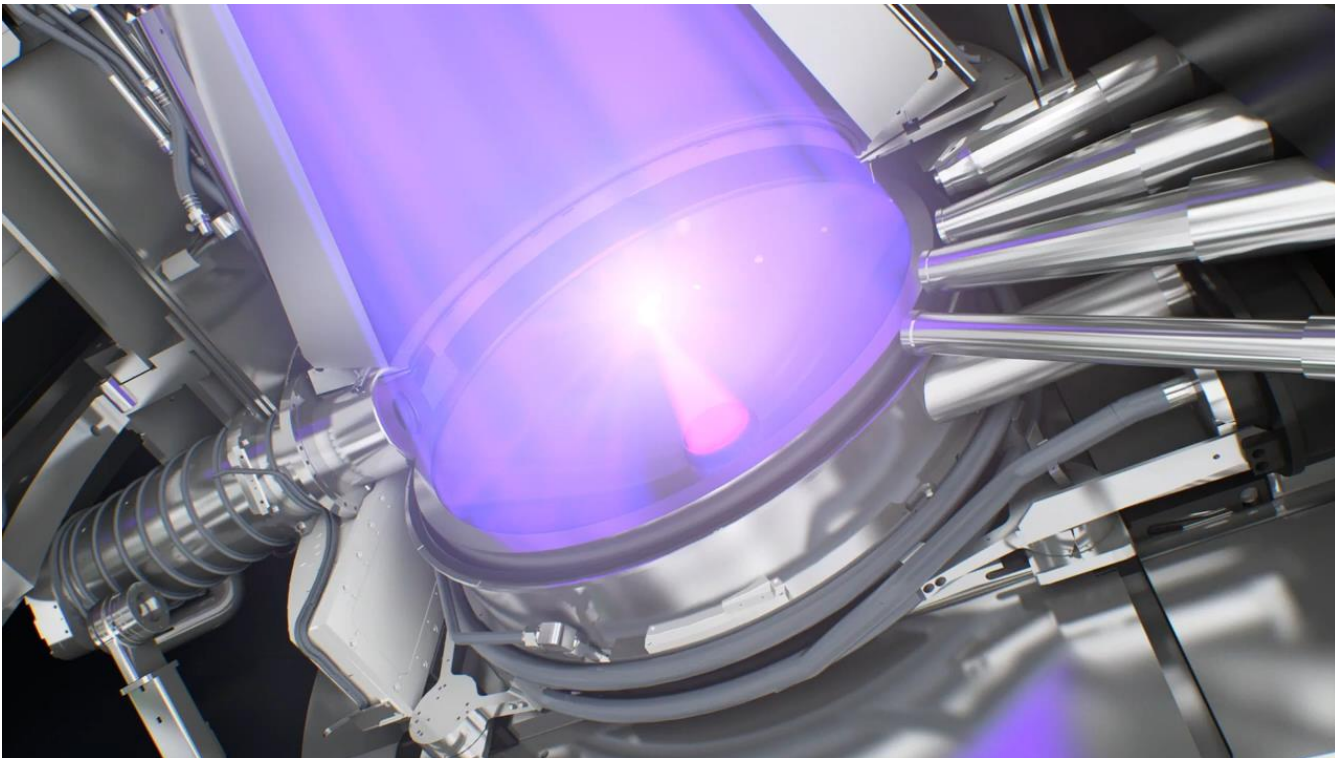
Generating EUV

Source of EUV light :

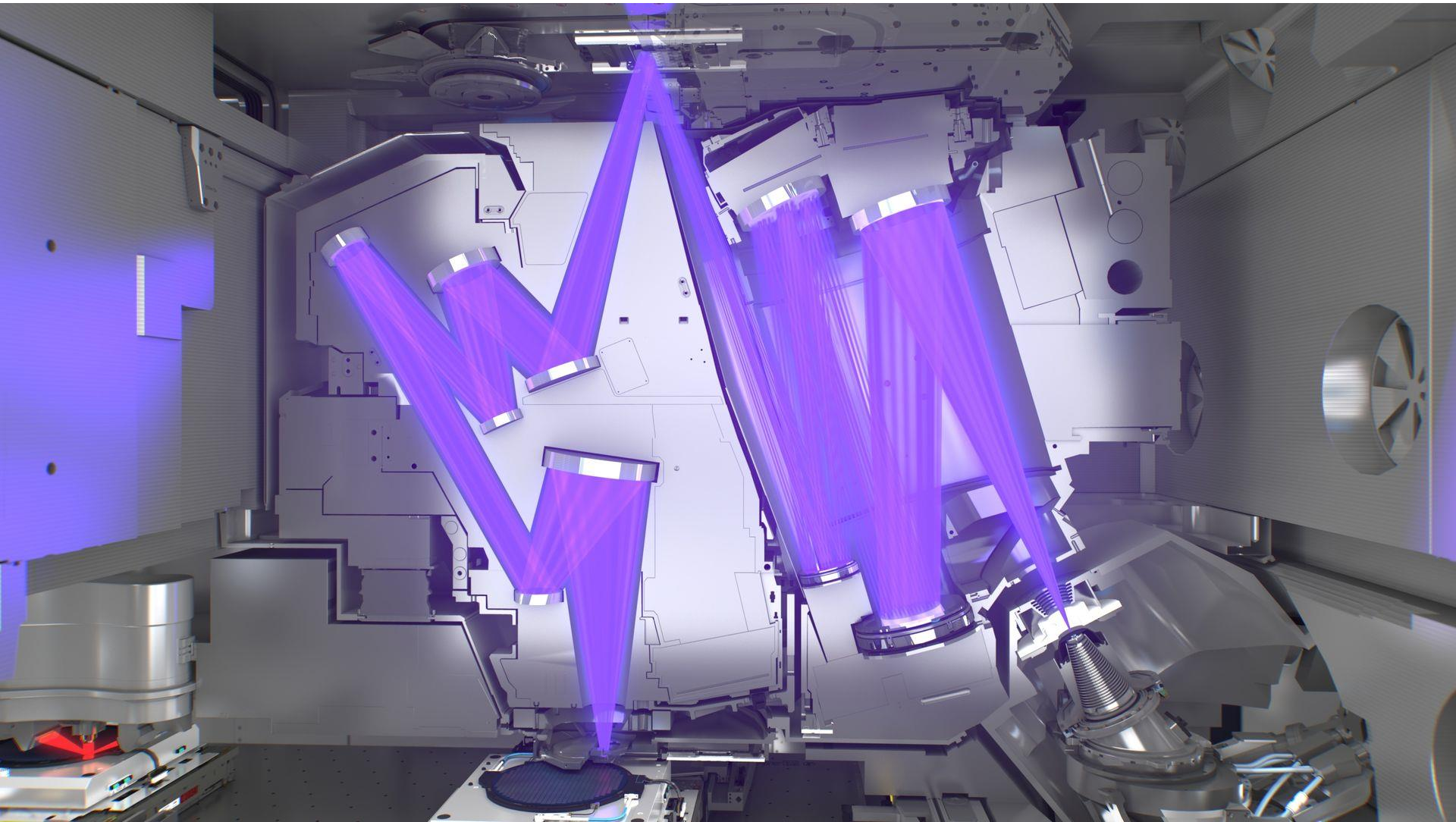
$4p^6 4d^m - 4p5 4d^{m+1} + 4d^{m-1} 4f$ in electron transitions in $\text{Sn}^{8+}-\text{Sn}^{14+}$

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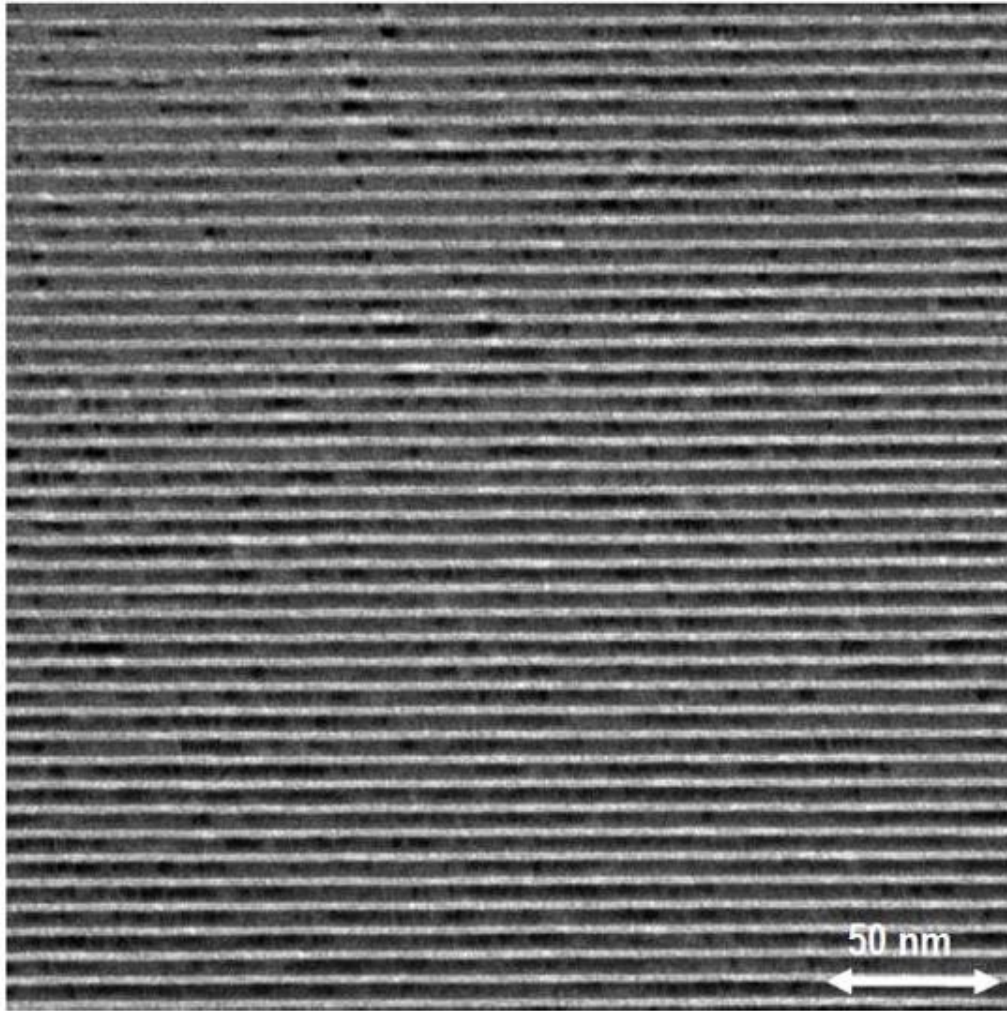
Handling EUV



Source: Zeiss & ASML

Multilayer mirrors for EUV

TEM of a Mo/Si EUV mirror, $N=50$, $\Lambda=6.8$ nm



Deposited by magnetron sputtering

HR TEM reveals interfacial MoSi_2

